

Arsenty Kaganskiya, Tobias Heuser, Ronny Schmidt, Sven Rodt,  
Stephan Reitzenstein

# CSAR 62 as negative-tone resist for high-contrast e-beam lithography at temperatures between 4 K and room temperature

Journal article | Published version

This version is available at <https://doi.org/10.14279/depositonce-11268>



© 2016 American Vacuum Society. This article may be downloaded for personal use only. Any other use requires prior permission of the author and AIP Publishing. This article appeared in

Kaganskiy, A., Heuser, T., Schmidt, R., Rodt, S. & Reitzenstein, S. (2016). CSAR 62 as negative-tone resist for high-contrast e-beam lithography at temperatures between 4 K and room temperature. *Journal of Vacuum Science & Technology B, Nanotechnology and Microelectronics: Materials, Processing, Measurement, and Phenomena*, 34(6), 061603. <https://doi.org/10.1116/1.4965883>

and may be found at <https://doi.org/10.1116/1.4965883>.

## Terms of Use

Copyright applies. A non-exclusive, non-transferable and limited right to use is granted. This document is intended solely for personal, non-commercial use.

WISSEN IM ZENTRUM  
UNIVERSITÄTSBIBLIOTHEK

Technische  
Universität  
Berlin

# CSAR 62 as negative-tone resist for high-contrast e-beam lithography at temperatures between 4 K and room temperature

Cite as: J. Vac. Sci. Technol. B **34**, 061603 (2016); <https://doi.org/10.1116/1.4965883>

Submitted: 22 June 2016 . Accepted: 10 October 2016 . Published Online: 29 November 2016

Arsenty Kaganskiy, Tobias Heuser, Ronny Schmidt, Sven Rodt, and Stephan Reitzenstein



View Online



Export Citation



CrossMark

## ARTICLES YOU MAY BE INTERESTED IN

[Investigation of CSAR 62, a new resist for electron beam lithography](#)

Journal of Vacuum Science & Technology B **32**, 06FJ01 (2014); <https://doi.org/10.1116/1.4899239>

[In situ electron-beam lithography of deterministic single-quantum-dot mesa-structures using low-temperature cathodoluminescence spectroscopy](#)

Applied Physics Letters **102**, 251113 (2013); <https://doi.org/10.1063/1.4812343>

[Resolution and alignment accuracy of low-temperature in situ electron beam lithography for nanophotonic device fabrication](#)

Journal of Vacuum Science & Technology B **33**, 021603 (2015); <https://doi.org/10.1116/1.4914914>

**HIDEN**  
ANALYTICAL

## Instruments for Advanced Science

Contact Hiden Analytical for further details:

W [www.HidenAnalytical.com](http://www.HidenAnalytical.com)  
E [info@hiden.co.uk](mailto:info@hiden.co.uk)

CLICK TO VIEW our product catalogue



### Gas Analysis

- dynamic measurement of reaction gas streams
- catalysis and thermal analysis
- molecular beam studies
- dissolved species probes
- fermentation, environmental and ecological studies



### Surface Science

- UHV/TPO
- SIMS
- end point detection in ion beam etch
- elemental imaging - surface mapping



### Plasma Diagnostics

- plasma source characterization
- etch and deposition process reaction kinetic studies
- analysis of neutral and radical species



### Vacuum Analysis

- partial pressure measurement and control of process gases
- reactive sputter process control
- vacuum diagnostics
- vacuum coating process monitoring



# CSAR 62 as negative-tone resist for high-contrast e-beam lithography at temperatures between 4 K and room temperature

Arsenty Kaganskiy,<sup>a)</sup> Tobias Heuser, Ronny Schmidt, Sven Rodt, and Stephan Reitzenstein  
*Institut für Festkörperphysik, Technische Universität Berlin, Hardenbergstraße 36, D-10623 Berlin, Germany*

(Received 22 June 2016; accepted 10 October 2016; published 24 October 2016)

The temperature dependence of the electron-beam sensitive resist CSAR 62 is investigated in its negative-tone regime. The writing temperatures span a wide range from 4 K to room temperature with the focus on the liquid helium temperature regime. The importance of low temperature studies is motivated by the application of CSAR 62 for deterministic nanophotonic device processing by means of *in situ* electron-beam lithography. At low temperature, CSAR 62 exhibits a high contrast of 10.5 and a resolution of 49 nm. The etch stability is almost temperature independent and it is found that CSAR 62 does not suffer from peeling which limits the low temperature application of the standard electron-beam resist polymethyl methacrylate. As such, CSAR 62 is a very promising negative-tone resist for *in situ* electron-beam lithography of high quality nanostructures at low temperature. © 2016 American Vacuum Society. [<http://dx.doi.org/10.1116/1.4965883>]

## I. INTRODUCTION

The development of functional quantum devices based on single quantum dots is an emerging field in semiconductor nanotechnology.<sup>1,2</sup> One of the big challenges is the precise integration of a single quantum emitter into a nanophotonic device structure.<sup>3</sup> To overcome this issue, deterministic lithography techniques were developed that combine the tasks of spectroscopic identification of a single high quality emitter with instant lithography. This was realized by a combination of optical spectroscopy and optical lithography<sup>4</sup> and by combined cathodoluminescence spectroscopy (CL) and electron-beam lithography (EBL),<sup>5</sup> dubbed cathodoluminescence lithography (CLL). Due to the emission properties of most semiconductor quantum dots, such processing has to be performed at low cryogenic temperatures or even liquid helium (l-He) temperature for sufficient signal-to-noise ratio.<sup>6</sup> Naturally, EBL is the most attractive technique to fabricate very precise or even three dimensional device structures with flexible design.<sup>7–10</sup> This initiates the quest for a detailed understanding of low-temperature-capable electron-beam-sensitive resists.<sup>11</sup> Additionally, a resist with a low sensitivity in the negative-tone regime is needed for CLL: the preceding spectroscopic investigation on the resist-coated sample requires dwell times per pixel of 10–30 ms. The corresponding electron dose must not lead to an effective exposure of the resist. This points to EBL resists that can be operated in both positive-tone (via chain-scission at small doses) and negative-tone (via cross-linking at large doses) regimes, with the effective doses for the negative-tone regime being orders of magnitude larger than for the positive-tone regime. The negative-tone resists also exhibit a higher mechanical stability of written structures due to the branched chain structure that results from the cross-linking process.<sup>12</sup> A prominent example for these classes of resists is polymethyl methacrylate (PMMA). The negative-tone PMMA was already applied for, e.g., the fabrication of quasi-three-dimensional micro/nanomechanical

components<sup>9</sup> and planar cobalt electrodes separated by a sub-10 nm gap.<sup>13</sup> Also in its negative-tone regime, it acts as a high-resolution resist.<sup>14,15</sup> It was found to be suitable for low-temperature CLL,<sup>5,16</sup> especially for 3D patterning.<sup>10,17</sup> A drawback in its use, however, is the necessity of a two-step lithography approach to gain a high device yield.<sup>10</sup> Also, a higher contrast is desirable for the fabrication of filigree structures with steep facets like ring resonators<sup>18</sup> and photonic crystals.<sup>19</sup> Besides PMMA and ZEP (see below), there are some more positive-tone/negative-tone resists like metal-carbonyl organometallic polymers,<sup>20</sup> poly(methylglutarimide),<sup>21</sup> phenyl-bridged polysilsesquioxane,<sup>22</sup> and triphenylene derivatives.<sup>23</sup>

Here, we present investigations on the EBL resist AR-P 6200 (CSAR 62) from Allresist GmbH in its negative-tone regime and in the full range between l-He temperature and room temperature. It is a poly- $\alpha$ -methylstyrene-*co*-methylchloroacrylate based resist together with an acid generator dissolved in anisole that was originally designed to be used as a positive-tone resist. However, it is similar to the positive-tone resist ZEP that was shown to be suitable for negative-tone lithography also.<sup>24,25</sup> We demonstrate that CSAR is very suitable for the aforementioned requirements of deterministic low-temperature CLL without the need for a two-step exposure.

## II. METHODS

EBL at various temperatures was performed in a setup based on a Jeol JSM 840 scanning electron microscope (SEM) which was extended by a custom-made low-temperature lithography attachment including a liquid helium flow cryostat. It also features an extension for cathodoluminescence spectroscopy.<sup>26</sup> The commercial resist under investigation is AR-P 6200 (CSAR) in anisole solvent (4% and 9% solid content for resist thicknesses of 78 and 210 nm, respectively). Prior to coating, the used underlying GaAs substrate was cleaned for 5 min in acetone and isopropanol (IPA) at 70 °C, respectively. Two resist thicknesses of 78 and 210 nm were fabricated by spin coating with a rotation frequency of 6000 and 3000

<sup>a)</sup>Electronic mail: [arsenty.kaganskiy@tu-berlin.de](mailto:arsenty.kaganskiy@tu-berlin.de)

rotations per minute, respectively, as an additional process parameter. After that, the samples were postbaked for 1 min at 175 °C. The resist thickness was determined by mechanical removing of the resist and subsequent profilometry across the area with and without the resist. The uncertainty of the measurement was  $\pm 3$  nm. Motivated by the upcoming low-temperature lithography techniques, we studied the properties of CSAR 62 in the temperature range between 4 and 295 K. EBL was done with an acceleration voltage of 20 kV and beam currents of 0.5 and 1 nA. In order to study the onset-dose and the contrast of CSAR 62 in the negative-tone regime, contrast curves with a linear dose gradient were written. Each dose gradient is made up of a 75  $\mu\text{m}$  long and 15  $\mu\text{m}$  broad stripe, which consists of equally spaced lines with a 10 nm step width (corresponding to  $\approx 6.7 \mu\text{C}/\text{cm}^2$  dose step width) written with a linearly increased nominal dose from 0 to 50  $\text{mC}/\text{cm}^2$ . The developed and etched structures were analyzed with the profilometer Ambios XP2 (160 nm tip step width). By using marker lines above and under the gradient stripe, the scan length measured by the profilometer was converted into the written dose [Fig. 1(a)].

To adapt the development process to the low-temperature writing, different development recipes were applied (see Sec. III). Best results were obtained after dip developing for 45 s with the commercial developer AR 600-546 by Allresist. Then, the samples were dipped for 30 s in isopropanol (Isopropanol Micropur by Technic France, VLSI purification grade) followed by a 30 s dip in deionized water to stop the process. In order to analyze the etch selectivity of the inverted resist, the samples were dry etched using inductively coupled-plasma (ICP) reactive-ion etching. As process gases for the plasma 1.3 sccm  $\text{Cl}_2$ , 4.3 sccm  $\text{BCl}_3$ , and 1.1 sccm Ar were used. The etching was done with a pressure of 0.08 Pa, conductively coupled plasma power of 24.5 W, ICP power of 100 W, and bias voltage of  $-215$  V.

Arrays of lines and dots with varying widths and diameters were written and processed to determine the resolution of the resist. After the resist development as well as resist etching, they were evaluated with a Digital Instruments Multimode AFM (MMAFM-2) in a constant force contact mode and with a Zeiss Ultra 55 SEM using a low acceleration voltage of 0.8 kV to avoid charging artifacts and achieve a charge balance regime.<sup>27</sup>

### III. RESULTS AND DISCUSSION

First we optimized the development process of the inverted low-temperature resist by comparing results for different developers (AR 600-546, AR 600-548, and AR 600-56 by Allresist, IPA:water 7:3 solution and pure IPA) in combination with various development times. All samples (except of those which were developed in IPA or in IPA:water solution) were dipped after the development for 30 s in IPA and swiveled for 30 s in deionized water.

At first, we used the developer AR 600-546 (based on amyl acetate) and developed a sample for 45 s. This time span was enough to complete the development process and it caused no resist peeling or structural damage as observed,

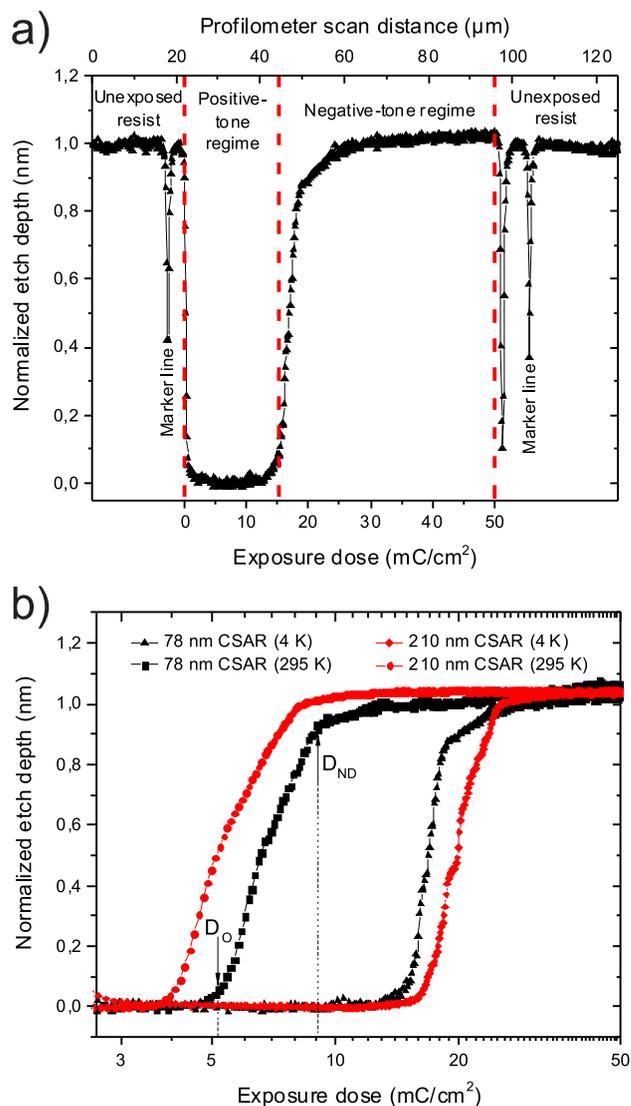


Fig. 1. (Color online) (a) Etched contrast curve of 78 nm thick CSAR 62 written at 4 K. The curve is plotted on a linear scale and schematically divided into unexposed, positive-tone and negative-tone regions. The marker lines used for the accurate assignment of electron dose to lateral position are shown. (b) Etched contrast curves for 78 nm (triangles and rectangles) and 210 nm (diamonds and circles) thick CSAR 62 written at 4 and 295 K, respectively. The curves are plotted on a half logarithmic scale. All curves are restricted to the negative-tone regime, where the resist starts to cross-link. The slopes of the curves refer to the contrast, which is evaluated by Eq. (1) and given in Table I. Doses  $D_O$  and  $D_{ND}$  are shown exemplarily for a 78 nm thick CSAR written at 295 K.

e.g., for low-temperature-exposed PMMA.<sup>10</sup> Due to the fact that the developer AR 600-548 (main component diethyl ketone/diethyl malonate) is more aggressive than AR 600-546 and causes dark erosion, we started from a 5 s development, which was not sufficient to develop tight trenches in the resist. Another 5 s of development lead to a peeling of the resist on the entire sample surface. So, we skipped further tests on that developer. Next, we developed a sample for 5 s with AR 600-56 (on the basis of methyl isobutyl ketone) resulting in a complete development, being comparable to the development for 45 s in AR 600-546. Finally, we tested IPA and IPA:water solution. Neither positive-tone nor negative-tone CSAR 62 were completely developed after developing for 10 s in pure IPA and for 20 s in a 7:3 solution of IPA:water (77% clearance

in the positive-tone regime). Another drawback is a reduced onset dose ( $D_O$ ) of  $8.5 \text{ mC/cm}^2$  ( $3.1 \text{ mC/cm}^2$ ) for the 78 nm thick resist developed in pure IPA (IPA–water solution) in contrast to  $15 \text{ mC/cm}^2$  for the sample developed with AR 600-546. *In situ* spectroscopic tasks via CL are limited by the time window before cross-linking starts at  $D_O$ . Both onset doses hinder such investigations. Together with a reduced sensitivity in the positive regime, it confirms previous results.<sup>25,28</sup> In conclusion, we decided to develop the investigated samples for 45 s with the developer AR 600-546.

To accurately evaluate the contrast curves and to obtain the contrast and the onset dose, etching into the underlying GaAs was performed for two reasons: First, the resist gets scratched by the needle of the profilometer which distorts the results and, second, we directly obtain the etched profiles which is of large interest for the envisaged transfer of resist patterns into the semiconductor. The only direct profilometry on the resist was performed to get the resist thickness of fully inverted CSAR 62 (cf. Fig. 3). Figure 1(a) presents an example of an etched contrast curve that was enclosed into regions of unexposed resist. As mentioned before, at small exposure doses, the resist can be completely removed from the sample surface during the development (positive-tone regime), while at higher doses, its components cross-link and the resist becomes insoluble for the developer (negative-tone regime). Figure 1(b) displays etched dose gradients on a logarithmic scale zoomed into the negative-tone regime. They were written at different temperatures into resist layers with a thickness of 78 and 210 nm, respectively. The etching was stopped at the point when the unexposed resist was completely removed. The etch time was 115 s (250 s) for the 78 nm (210 nm) thick resist. The etch depth is normalized with respect to the regions of unexposed resist. The black (red) curves correspond to an initial resist thickness of 78 nm (210 nm) and writing temperatures of 295 and 4 K were exploited.

In order to calculate the contrast  $\gamma$  of the resist in the negative-tone regime, we used the common equation

$$\gamma = -\frac{1}{\log(D_O/D_{ND})}. \quad (1)$$

$D_O$  is the onset dose for the cross linking and marks the beginning of the curves' upward slopes.  $D_{ND}$  gives the dose for which the resist is fully cross-linked and the etch depth starts to be constant. Consequently, the contrast is directly expressed by the slope of the curve between  $D_O$  and  $D_{ND}$ . Table I summarizes the contrast values as extracted from Fig. 1.

TABLE I. Contrast of CSAR 62 for two different temperatures and resist thicknesses. The highest value of 10.5 corresponds to a resist thickness of 78 nm and a writing temperature of 4 K.

Temperature (K)	Resist thickness	
	78 nm	210 nm
4	10.5	5.7
295	2.97	2.84

The highest contrast of 10.5 is observed for the resist thickness of 78 nm and 4 K temperature. A higher resist thickness results in a reduced contrast as does a higher temperature. It is worth to mention that the contrast value of 5.7 measured for a 210 nm thick CSAR 62 written at 4 K is almost five times higher than the contrast of a similar thick PMMA written at the same temperature as reported by us previously.<sup>10</sup>

Due to its higher contrast and resolution and hence its wider potential application spectrum, the 78 nm thick resist was subsequently investigated in detail by evaluating etched contrast curves in the full temperature range between 4 and 295 K. Figure 2 (black circles) shows the onset dose of the negative-tone regime as a function of temperature. Clearly, the onset dose decreases monotonously with increasing temperature. The processes occurring during the transition from the positive- to negative-tone regime in a similar ZEP resist were studied by Oyama *et al.*<sup>29</sup> They showed that a driving force leading to a main chain scission under e-beam exposure is a dissociation of the chlorine atom of the  $\alpha$ -chloromethacrylate unit of the chain via dissociative electron attachment, which leads to the formation of temporal double bonds. Thereby a number of Cl bonds and amounts of Cl decrease rapidly for an increasing exposure dose. The cross-linking occurs during a further exposure at higher doses, at which free radicals are activated at the cost of a reduced number of double bonds and an increasing number of C-H bonds.<sup>24,30</sup> Due to a higher mobility of activated and free radicals at higher writing temperatures, the cross-linking starts at smaller onset doses (Fig. 2).

A similar trend is observed for the contrast (red squares in Fig. 2). To assess this behavior, it is necessary to evaluate both doses that are considered in Eq. (1). As the cross linking starts from the onset dose, the dose difference  $D_{ND}-D_O$  indicates the electron dose that has to be additionally deposited to fully invert the resist. As it can be seen in the inset of Fig. 2,  $D_{ND}-D_O$  is almost temperature independent, which is also

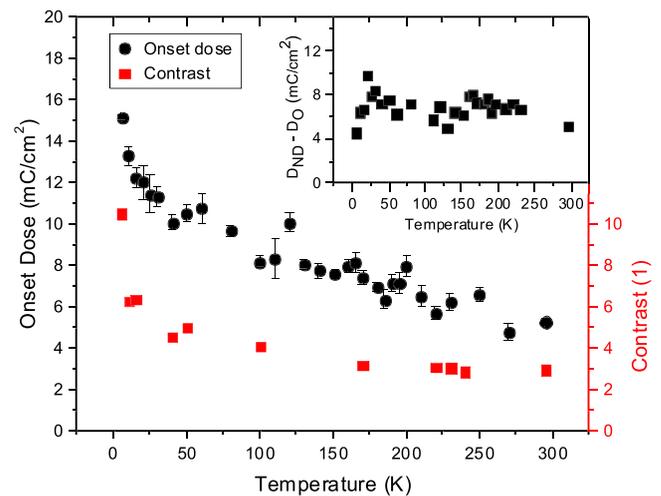


Fig. 2. (Color online) Onset dose (circles) and contrast (rectangles) of 78 nm CSAR 62 as a function of the writing temperature. The onset-dose decreases with increasing temperature due to accelerated cross-linking of free radicals. The inset shows the temperature dependence of the additional electron dose that is required to finish the cross-linking process when it has started at  $D_O$ .

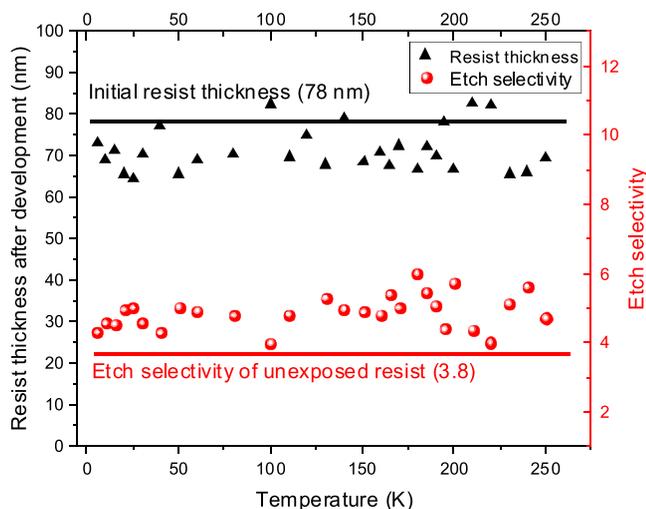


FIG. 3. (Color online) Remaining resist thickness of formerly 78 nm thick CSAR 62 after the development (triangles). The thickness reduction of 8% corresponds to a moderate outgassing of HCl and other compounds. Etch selectivity of negative-tone CSAR 62 (circles). In all cases the applied electron dose was 40 mC/cm<sup>2</sup>. The thickness and etch selectivity are almost constant in the whole temperature range.

reflected in the identical slopes of linearly plotted contrast curves (not shown here). Together with the trend of  $D_O$  and the logarithmic definition of the contrast value [Eq. (1)], this results in the observed temperature dependence of the contrast. A similar behavior of decreasing contrast with increasing temperature was observed by Sidorkin *et al.*<sup>31</sup> for hydrogen silsesquioxane; however, no explanation could be given.

The thickness of fully inverted resist after its development was also investigated as a function of temperature (black triangles in Fig. 3). It is independent on temperature and has decreased on average by about 6.3 nm (8%) as compared to the initial resist thickness of 78 nm. This is attributed to the elimination of Cl from the resist and subsequent outgassing of HCl and other scission products and the buildup of a more dense cross-linked compound.<sup>29</sup> Additionally and consequently, also the etch selectivity against GaAs is almost constant over the whole temperature range with an increase of  $\approx 30\%$  relative to the unexposed resist. This is an ideal situation for deterministic nanoprocessing as it enables constant etching and processing conditions over the entire range of writing temperatures. Additionally, the etch selectivity of  $\approx 4.9$  in combination with an increased etch stability of CSAR 62 as compared to PMMA ( $\approx$  by a factor of 2) allows for direct etching without the need for intermediate hard masks.

These results for the resist thickness and etch selectivity are in contrast to our experimental findings on PMMA. The low-temperature negative-tone PMMA was found to suffer from an extensive swelling due to trapped gaseous scission products around 70 K that also reduced the etch selectivity in this temperature regime in a drastic way.<sup>11</sup> Low-temperature writing of PMMA also requires a two-step lithography technique to prevent it from peeling when trapped gases burst out.<sup>10,32</sup> In the case of CSAR 62, we did not observe such structural damage for structures written at low-temperatures, which makes it much more suitable for low temperature EBL. It is noteworthy that the reproducibility is very good

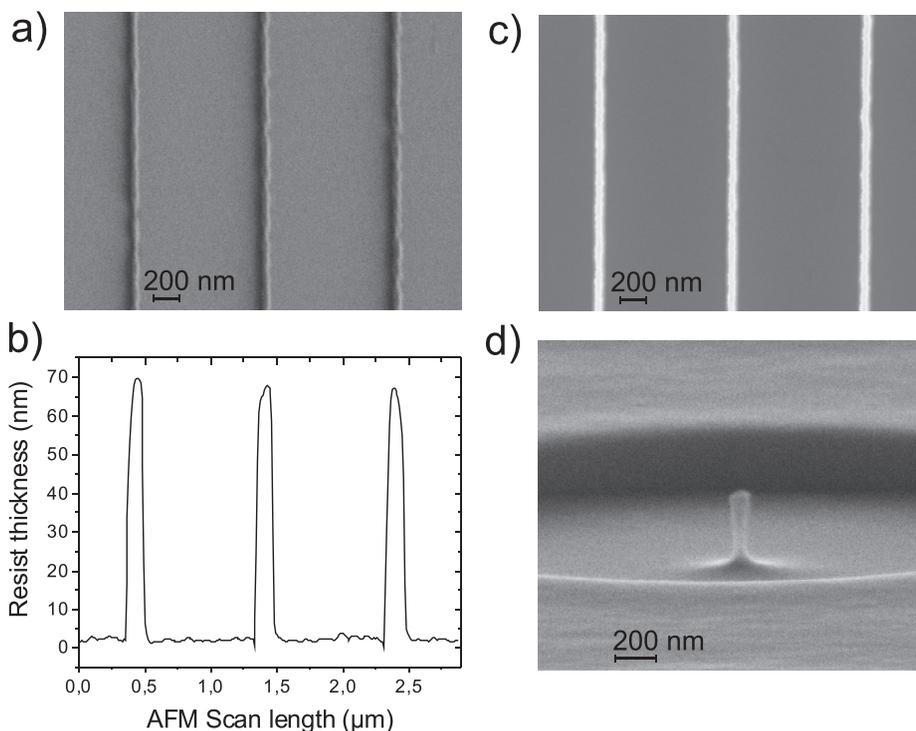


FIG. 4. SEM images of developed (a) and etched (c) line arrays. The width of the lines is 49 nm. (b) AFM profile of a group of narrow lines after the development. (d) Etched pillar structure with a diameter of 49 nm. All structures were written into 78 nm thick CSAR 62 in its negative-tone regime and at a temperature of 4 K.

by means of characteristic doses and quality of the written structures.

In order to determine the resolution capability of low-temperature CSAR 62, we performed lithography of line arrays and dots at 4 K written into a 78 nm thick resist layer. The lines were written with a pitch of 1  $\mu\text{m}$ . They consist of aligned equidistant dots while their width was varied by means of an increasing exposure dose. We could observe continuous lines starting from a width of 49 nm as it can be seen in the SEM image in Fig. 4(a). The lines are well resolved, structurally intact, and exhibit low line width and edge roughness. In Fig. 4(b), an AFM scan of such narrow lines is shown. Due to the aforementioned high contrast of CSAR 62, one can see pronounced steep edges of the lines. A slight broadening of the lines is caused by convolution with the AFM tip. SEM image of a group of 49 nm wide lines after etching is shown in Fig. 4(c). It can be clearly seen that the high resolution and edge steepness of the resist could be completely transferred in the etched GaAs. The aforementioned high resolution of the resist is confirmed by a number of pillars written in the positive-tone as well as in the negative-tone regime which exhibit a smallest diameter of the developed pillars of 48 and 49 nm, respectively. An etched pillar with a diameter of 49 nm written in the negative-tone regime is exemplarily shown in Fig. 4(d). In conclusion, we obtain a resolution that is even better than that of the low-temperature negative-tone PMMA, which exhibited at best a resolution of 65 nm.<sup>26</sup>

From AFM measurements, we evaluated the average surface roughness and root mean square roughness of 0.43 and 0.58 nm, respectively, measured over an area of  $5 \times 5 \mu\text{m}^2$  of the negative-tone (40 mC/cm<sup>2</sup>) 78 nm thick resist at 4 K after the development.

We also want to point out our finding that negative-tone CSAR 62 can be easily removed in a moderate oxygen plasma, while the surface of the semiconductor sample is not damaged. This feature is crucial for the residue-free processing of nanophotonic devices.

#### IV. SUMMARY AND CONCLUSION

The EBL resist CSAR 62 has been investigated in the temperature range between 4 K and room temperature in its negative-tone regime. At low temperatures, it exhibits high contrasts of up to 10.5 that are linked to the temperature dependence of the onset dose. The resist thickness and etch selectivity are almost independent on temperature and no detrimental swelling or peeling of low-temperature written structures was observed. The resolution at 1-He temperature was better than 50 nm. This makes CSAR 62 a very attractive resist for novel low-temperature applications in the field of deterministic semiconductor device technology.

#### ACKNOWLEDGMENTS

The research leading to these results has received funding from the European Research Council under the European

Union's Seventh Framework ERC Grant Agreement No. 615613, and from the Deutsche Forschungsgemeinschaft (DFG) through SFB 787 "Semiconductor Nanophotonics: Materials, Models, Devices."

- <sup>1</sup>S. Buckley, K. Rivoire, and J. Vuckovic, *Rep. Prog. Phys.* **75**, 126503 (2012).
- <sup>2</sup>P. Lodahl and S. Stobbe, *Nanophotonics* **2**, 39 (2013).
- <sup>3</sup>S. Reitzenstein, *IEEE J. Sel. Top. Quantum Electron.* **18**, 1733 (2012).
- <sup>4</sup>A. Dousse *et al.*, *Phys. Rev. Lett.* **101**, 267404 (2008).
- <sup>5</sup>M. Gschrey, F. Gericke, A. Schüßler, R. Schmidt, J.-H. Schulze, T. Heindel, S. Rodt, A. Strittmatter, and S. Reitzenstein, *Appl. Phys. Lett.* **102**, 251113 (2013).
- <sup>6</sup>M. Bayer and A. Forchel, *Phys. Rev. B* **65**, 041308 (2002).
- <sup>7</sup>Y. Chen, *Microelectron. Eng.* **135**, 57 (2015).
- <sup>8</sup>V. Kudryashov, X.-C. Yuan, W.-C. Cheong, and K. Radhakrishnan, *Microelectron. Eng.* **67**, 306 (2003).
- <sup>9</sup>W. H. Teh and C. G. Smith, *J. Vac. Sci. Technol., B* **21**, 3007 (2003).
- <sup>10</sup>P. Schnauber, R. Schmidt, A. Kaganskiy, T. Heuser, M. Gschrey, S. Rodt, and S. Reitzenstein, *Nanotechnology* **27**, 195301 (2016).
- <sup>11</sup>M. Gschrey, R. Schmidt, A. Kaganskiy, S. Rodt, and S. Reitzenstein, *J. Vac. Sci. Technol., B* **32**, 061601 (2014).
- <sup>12</sup>W.-M. Yeh, D. E. Noga, R. A. Lawson, L. M. Tolbert, and C. L. Henderson, *J. Vac. Sci. Technol., B* **28**, C6S6 (2010).
- <sup>13</sup>L. Ressier, J. Grisolia, C. Martin, J. P. Peyrade, B. Viallet, and C. Vieu, *Ultramicroscopy* **107**, 985 (2007).
- <sup>14</sup>I. Zailer, J. E. F. Frost, V. Chabasseur-Molyneux, C. J. B. Ford, and M. Pepper, *Semicond. Sci. Technol.* **11**, 1235 (1996).
- <sup>15</sup>A. C. F. Hoole, M. E. Welland, and A. N. Broers, *Semicond. Sci. Technol.* **12**, 1166 (1997).
- <sup>16</sup>A. Kaganskiy, M. Gschrey, A. Schlehahn, R. Schmidt, J.-H. Schulze, T. Heindel, A. Strittmatter, S. Rodt, and S. Reitzenstein, *Rev. Sci. Instrum.* **86**, 073903 (2015).
- <sup>17</sup>M. Gschrey *et al.*, *Nat. Commun.* **6**, 7662 (2015).
- <sup>18</sup>M. Davanco, M. T. Rakher, D. Schuh, A. Badolato, and K. Srinivasan, *Appl. Phys. Lett.* **99**, 041102 (2011).
- <sup>19</sup>T. D. Happ, I. I. Tartakovskii, V. D. Kulakovskii, J.-P. Reithmaier, M. Kamp, and A. Forchel, *Phys. Rev. B* **66**, 041303 (2002).
- <sup>20</sup>J. Zhang, K. Cao, X. S. Wang, and B. Cui, *Chem. Commun.* **51**, 17592 (2015).
- <sup>21</sup>G. Karbasian, P. J. Fay, H. Xing, D. Jena, A. O. Orlov, and G. L. Snider, *J. Vac. Sci. Technol., B* **30**, 06F101 (2012).
- <sup>22</sup>L. Brigo, V. Auzelyte, K. A. Lister, J. Brugger, and G. Brusatin, *Nanotechnology* **23**, 325302 (2012).
- <sup>23</sup>A. P. G. Robinson, R. E. Palmer, T. Tada, T. Kanayama, M. T. Allen, J. A. Preece, and K. D. M. Harris, *J. Phys. D: Appl. Phys.* **32**, L75 (1999).
- <sup>24</sup>T. G. Oyama, A. Oshima, H. Yamamoto, S. Tagawa, and M. Washio, *Appl. Phys. Express* **4**, 076501 (2011).
- <sup>25</sup>M. A. Mohammad, K. Koshelev, T. Fito, D. A. Z. Zheng, M. Stepanova, and S. Dew, *Jpn. J. Appl. Phys., Part 1* **51**, 06FC05 (2012).
- <sup>26</sup>M. Gschrey, R. Schmidt, J.-H. Schulze, A. Strittmatter, S. Rodt, and S. Reitzenstein, *J. Vac. Sci. Technol., B* **33**, 021603 (2015).
- <sup>27</sup>R. Wührer and K. Moran, *IOP Conf. Ser.: Mater. Sci. Eng.* **109**, 012019 (2016).
- <sup>28</sup>K. Koshelev, A. M. Mohammad, T. Fito, K. L. Westra, S. K. Dew, and M. Stepanova, *J. Vac. Sci. Technol., B* **29**, 06F306 (2011).
- <sup>29</sup>T. G. Oyama, K. Enomoto, Y. Hosaka, A. Oshima, M. Washio, and S. Tagawa, *Appl. Phys. Express* **5**, 036501 (2012).
- <sup>30</sup>T. G. Oyama, H. Nakamura, A. Oshima, M. Washio, and S. Tagawa, *Appl. Phys. Express* **7**, 036501 (2014).
- <sup>31</sup>V. Sidorkin, E. van der Drift, and H. Salemkink, *J. Vac. Sci. Technol., B* **26**, 2049 (2008).
- <sup>32</sup>S. Unai, N. Puttaraksa, N. Pussadee, K. Singkarat, M. W. Rhodes, H. J. Whitlow, and S. Singkarat, *Microelectron. Eng.* **102**, 18 (2013).