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Single-photon emission from InGaAs quantum dots grown on (111) GaAs

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In this letter, we demonstrate that self-organized InGaAs quantum dots (QDs) grown on GaAs (111) substrate using droplet epitaxy have great potential for the generation of entangled photon pairs. The QDs show spectrally sharp luminescence lines and low spatial density. A second order correlation value of $g^{(2)}(0) < 0.3$ proves single-photon emission. By comparing the power dependence of the luminescence from a number of QDs we identify a typical luminescence fingerprint. In polarization dependent microphotoluminescence studies a fine-structure splitting ranging $\leq 40 \mu\text{eV}$ down to the determination limit of our setup ($10 \mu\text{eV}$) was observed. © 2010 American Institute of Physics. [doi:10.1063/1.3337097]

A network providing secure communication based on fundamental physical laws is possible if it uses quantum cryptography. The fundamental building block in such a quantum cryptography network is a robust and easy-to-handle source of entangled photon pairs.¹

Self-organized quantum dots (QDs) (Ref. 2) are highly efficient electrically-driven single-photon sources.^{3–5} As proposed by Benson *et al.*⁶ a single QD will be an excellent source for entangled photon pairs if the fine-structure splitting (FSS) of the bright exciton states is reduced to less than the homogeneous line width.

For In(Ga)As QDs on (001) GaAs the FSS is nonzero,^{7–9} even for perfectly symmetric QDs, due to the asymmetric piezoelectric potential along the QDs base. For the first experimental demonstration of entangled photon pairs from QDs, postgrowth manipulation such as spectral filtering¹⁰ and magnetic fields¹¹ were used. Other attempts to reduce the FSS include thermal annealing,¹² external stress¹³ or electric fields.¹⁴ These techniques are complex and have to be adjusted for every single QD. Only if by accident the asymmetric potential becomes very small or is compensated by the QD elongation, the FSS will be small enough for the generation of entangled photons.^{15,16}

In contrast to QDs on (001) GaAs, the piezoelectric field for QDs grown on (111) GaAs substrates is directed along the growth direction¹⁷ and does not lower the symmetry below C_{3v} along the QDs base. It has been proposed recently that in this case the FSS is expected to be zero.^{5,18,19} Therefore triangular shaped InAs QDs grown on (111) GaAs surface might be ready-to-go sources for entangled photon pairs.²⁰

In previous attempts to grow InAs QDs on (111) GaAs substrates tetrahedral shaped recesses were used.²¹ This leads to the formation of rather large dots with an homogeneous line width of a few millielectron volts. Another attempt to grow QDs on (111) substrates²² resulted in a high density of very small QDs with a small localization energy, which are electronically close to quantum wells.

In this letter, we present the realization of self-organized InGaAs QDs on (111) GaAs substrates with low spatial density, sharp luminescence lines and a FSS close to or below our spectral resolution. These QDs are promising candidates for sources of entangled photon pairs.

The samples were grown by molecular beam epitaxy on GaAs (111) substrates having a miscut of 2° in the $[2\bar{1}\bar{1}]$ direction, using a Riber32P system. On a 500 nm buffer layer a 50 nm $\text{Al}_{0.6}\text{Ga}_{0.4}\text{As}$ layer followed by 65 nm GaAs were deposited at 580°C . The QDs were grown using the droplet technique.²³ The substrate temperature was reduced to 510°C with closed As shutter. Nominally 1.5 ML of Ga was deposited to saturate the excess surface As atoms and ~ 2 ML of In was deposited without As flux, forming In droplets. The formation of InGaAs QDs took place under As flux with closed In shutter and the rotation of the sample was switched off. The QD layer was capped with 65 nm GaAs followed by a second $\text{Al}_{0.6}\text{Ga}_{0.4}$ layer and a final 10 nm GaAs layer. The two AlGaAs layers prevent carrier diffusion from the QD layer to the bulk material. Atomic force images of uncapped QD layers show triangular shaped structures with a base length on the order of 100 nm, while QD-sized structures could not be observed in these measurements.

The samples were excited with a frequency doubled Nd:YVO (yttrium vanadium oxide) laser (532 nm) in a microphotoluminescence setup at 15 K, resulting in a spot diameter of $\sim 1 \mu\text{m}$. The spectra were recorded with a monochromator and a Si-charge coupled device with a spectral resolution of $20 \mu\text{eV}$. Using fitting procedures the energetic position of the luminescence can be determined within $10 \mu\text{eV}$ precision. Polarization dependent measurements were performed using a fixed linear polarizer in front of the monochromator and a $\lambda/2$ plate to rotate the polarization, making the polarization anisotropy of the monochromator negligible. The second-order correlation-function $g^{(2)}(\Delta t)$ was measured using a Hanbury–Brown–Twiss (HBT) setup, with a time resolution of 700 ps and a spectral resolution of 0.3 meV.

Since the wafer rotation was switched off during growth, the luminescence of the samples varies significantly on different areas of the wafers. Some parts emit spectral broad

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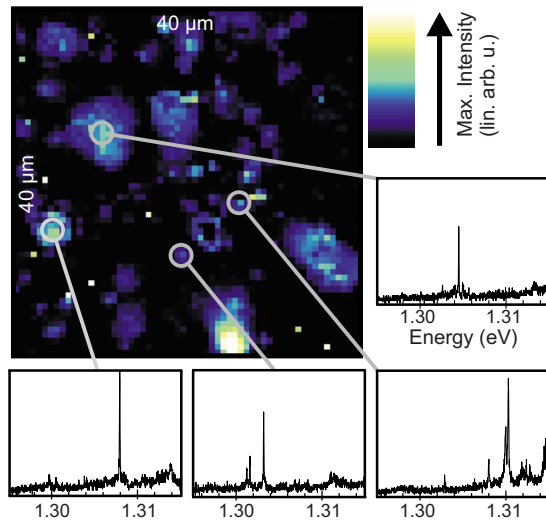


FIG. 1. (Color online) Spatial distribution of the QDs. The color scale indicates the value of the maximum intensity in the spectral range between 1.295 and 1.315 eV. The distance between dots emitting in this spectral range is on the order of micrometer. For four local spectra are plotted with the same scaling of the intensity.

luminescence, having little relation to QDs, whereas a small area showed spatially well separated luminescence centers, with spectral sharp luminescence lines. Figure 1 displays the spatial distribution of the luminescence in a spectral range from 1.295 to 1.315 eV. The color scale gives the value of the maximum intensity in this spectral range. On an area of $40 \times 40 \mu\text{m}^2$ several luminescence centers are visible. The spectra of four of them are plotted with identical intensity scales around the frame. Sharp luminescence lines which are typical for the discrete energy levels of QDs are visible. The line width of most of them is limited by the spectral resolution of our setup and the intensity is comparable to the luminescence from a single QD grown on (001) substrate. The distance between two QDs can exceed a few micrometers and the spatial QD density is significantly less than 10^9cm^{-2} . Thus single-QD spectroscopy without additional spatial selection is possible.

In order to check for the emission of single QDs we performed second-order correlation-function $g^{(2)}(\Delta t)$ measurements with the HBT. Figure 2(a) shows the normalized $g^{(2)}(\Delta t)$ function of one single luminescence line. The increased $g^{(2)}$ at $\pm 5 \text{ ns}$ is due to the so-called cross-talk.²⁴ The measured value of $g^{(2)}(0)=0.3$ is mainly limited by the time resolution of the HBT. A simulation (black line), taking into account the time resolution of the setup,⁵ leads to a $g^{(2)}(0)=0.1$. Hence the emission of single photons and the existence of QDs is proven.

Next we want to investigate the electronic origin of the emission lines. Besides their shape the electronic structure of QDs grown on (111) GaAs is expected to differ substantially from that of (001) QDs. Consequently we will be facing “novel” emission patterns. We start our analysis by finding recurring luminescence line patterns in power-dependent measurements as characteristic fingerprints and follow up with the investigation of FSS in polarization-dependent measurements.

The upper part of Fig. 3 shows the excitation power dependence of several photoluminescence lines, originating from one QD. With increasing laser power three lumines-

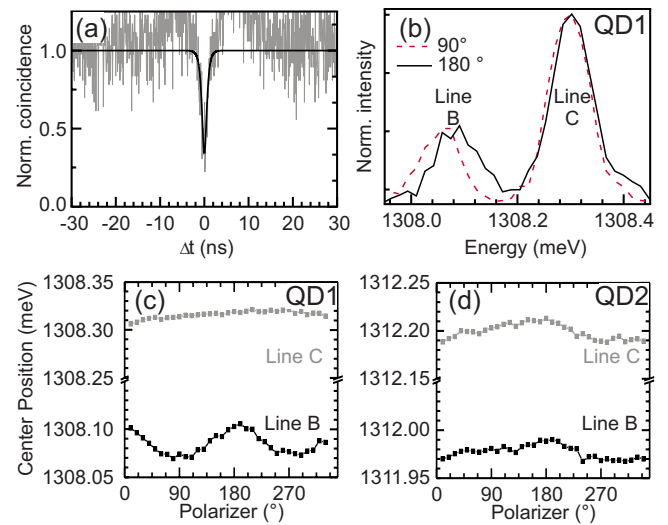


FIG. 2. (Color online) (a) Second-order correlation measurement (gray) and simulation (black) resulting in a $g^{(2)}(0)$ value of 0.1. (b) Spectra of QD1 with a polarizer at 90° (red dashed) and 180° (black solid), demonstrating the splitting of $40 \mu\text{eV}$. (c) Spectral center position of the luminescence line B and C from QD1 obtained from a Gaussian fit in dependence of the polarization. (d) the same for QD2: no periodical dependence is visible indicating a vanishing FSS.

cence lines appear simultaneously. One intense line at 1.306 eV (in the following labeled as line A) and two lines at 1.308 eV separated by $300 \mu\text{eV}$ (lines B and C). The intensity ratio of these lines is 4:1:2 (A:B:C) and their intensity depends almost linear on the excitation power. When the luminescence of these three lines saturates (at $\approx 100 \text{ nW}$ excitation power) several other lines appear and upon further increase in excitation power the lines A, B, and C fade. This scenery is typical for excitonic emission from a single QD. We therefore attribute the lines A, B, and C to single excitonic complexes of a single InAs QD grown on (111) GaAs substrate. For five more QDs we found a similar pattern of three lines with almost the same energetic distance and intensity ratio, exhibiting the same power dependence. In the following we will label all lines with this typical fingerprint by A, B, and C.

The crucial parameter for the generation of entangled photon pairs is the FSS of the bright exciton state. In general,

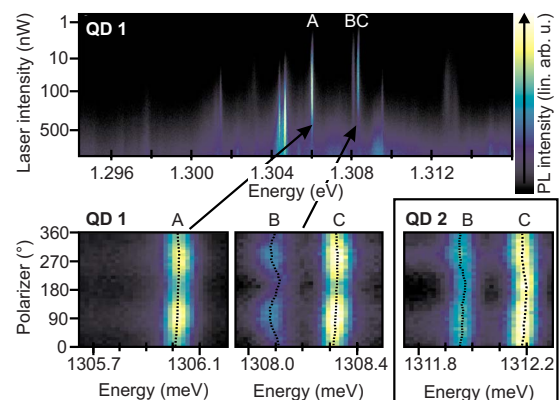


FIG. 3. (Color online) Top: excitation power dependence of the photoluminescence. The luminescence lines from single excitonic complexes are denoted by A, B, and C. Bottom: polarization dependence of the lines A, B, and C from QD1 and of lines B and C from QD2. The dotted lines show the center energy of the lines as in Fig. 2(c) and 2(d).

several excitonic complexes can be observed in the luminescence of a single QD. The uncharged exciton (X) consisting of a single electron-hole pair, the negative/positive trion (two electrons/holes with a single hole/electron, X^- , X^+) or even multiply charged excitons (X^{++} , X^- , ...).²⁵ For nonzero FSS the bright exciton emission line will split into a linearly polarized doubled as will one of the emission lines of the X^{++} .²⁶ Therefore the FSS can be measured in the luminescence of the uncharged and of the doubly charged exciton.

In order to determine a possible FSS for QDs grown on (111) GaAs substrate we performed polarization dependent measurements on lines A, B, and C (bottom of Fig. 3). The intensities of lines A and C are clearly polarization dependent. For an angle of 90° the intensity is 1.5 times larger than that for 180° , while the energetic position does not change. For QD1 line B on the other hand shows almost the same intensity for all polarizations but the energetic position (dashed line in Fig. 3) shifts. This is more visible in Fig. 2(b), which shows the spectra of QD1 for 90° (red) and 180° (black). The center positions of lines B and C obtained from a Gaussian fit for all polarization directions are plotted in Fig. 2(c). For line B a clear energetic change of about $40 \mu\text{eV}$ with a periodicity of 180° is clearly visible, whereas line C shows no systematic energetic shift. Therefore the luminescence of line B gives access to the FSS and originates either from the X or X^{++} . Line A and C are probably from trions. Spectroscopy on single QDs grown on (111) GaAs substrate is a new field and a complete assignment of the observed luminescence lines to transitions from well identified few-particle states requires further investigation like microphotoluminescence excitation spectroscopy²⁷ and has to be addressed in future work.

For QD2 we found the same luminescence fingerprint of three lines. The polarization dependence of the lines B and C is shown in the right bottom part of Fig. 3 as a color plot and the energetic positions in Fig. 2(d). For QD2 line B exhibits only a very small energetic shift on the order of the determination limit of our setup ($10 \mu\text{eV}$). In contrast to QD1 the spectral shift is not periodically and not reproducible. The FSS of QD1 was the largest we found and on five other QDs (not shown) it is ranging from $30 \mu\text{eV}$ down or below the determination limit.

We attribute the nonzero FSS to two effects mainly. The QDs were grown on a (111) substrate with a miscut of 2° . This may lead to the growth of elongated QDs along the steps on the (111) surface. Second, even the growth of GaAs bulk material on this substrate orientation is challenging. The samples exhibit a high density of dislocations, which may influence the built-in piezoelectric field and cause strain in one preferential direction. This may result in an asymmetric confinement potential, similar to that of QDs on (001) substrate. By optimizing the growth process these effects can be eliminated.

In conclusion we demonstrate the realization of self-organized InGaAs QDs on (111) GaAs substrates. The QDs have a very low spatial density and exhibit spectra with sharp and intense lines. The luminescence of the QDs demonstrates clear antibunching with an $g^{(2)}(0) < 0.3$ which

proves single photon emission. On the basis of excitation power-dependent measurements on a number of QDs we were able to identify a typical luminescence fingerprint of QDs grown on (111) GaAs. One of these lines allows the measurement of the FSS. The observed FSS ranges from $40 \mu\text{eV}$ down to the determination limit of our setup ($10 \mu\text{eV}$). Our results present an important step toward the realization of a compact source of entangled photon pairs based on semiconductor technology.

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