

Article

Modelling, Analysis, and Optimization of the Effects of Pulsed Electrophoretic Deposition Parameters on TiO₂ Films Properties Using Desirability Optimization Methodology

Nesrine Barbana ^{1,*}, Adel Ben Youssef ², Mohamed Ali Rezgui ², Latifa Bousselmi ³ and Mohammad Al-Addous ⁴

- ¹ Department of Environmental Process Engineering, Institute of Environmental Technology, Technische Universität Berlin, Secr. KF 2, Straße des 17. Juni 135, 10623 Berlin, Germany
- ² Higher National Engineering School of Tunis, University of Tunis, Tunis 1008, Tunisia; adel.benyoussef@esstt.rnu.tn (A.B.Y.); mohamedali.rezgui@gmail.com (M.A.R.)
- ³ Wastewaters and Environment Laboratory, Center for Water Research and Technologies, Technopark of Borj-Cedria, Soliman 8020, Tunisia; latifa.bousselmi.certe@gmail.com
- ⁴ Department of Energy Engineering, School of Natural Resources Engineering and Management, German-Jordanian University, Amman 11180, Jordan; mohammad.addous@gju.edu.jo
- * Correspondence: nesrine@campus.tu-berlin.de

Received: 6 October 2020; Accepted: 12 November 2020; Published: 16 November 2020



Abstract: Titanium dioxide thin films immobilized over treated stainless steel were prepared using the pulsed electrophoretic deposition technique. The effects of process parameters (deposition time, applied voltage, initial concentration, and duty cycle) on photocatalytic efficiency and adhesion properties were investigated. To optimize the multiple properties of the thin film, a response surface methodology was combined with a desirability optimization methodology. Additionally, a quadratic model was established based on response surface analysis. The precision of the models was defined based on the analysis of variance (ANOVA), R^2 , and the normal plot of residuals. Then, a desirability function was used to optimize the multiple responses of the TiO₂ thin film. The optimum values of applied voltage, catalyst concentration, duty cycle, and deposition time were 4 V, 16.34 g/L, 90% *DC*, and 150 s, respectively. Under these conditions, the decolorization efficiency of tested dye solution reached 82.75%. The values of critical charges L_{C1} , L_{C2} , and L_{C3} were 5.9 N, 12.5 N, and 16.7 N, respectively.

Keywords: pulsed electrophoretic deposition; photocatalysis; scratch test; conversion layer; RSM; desirability optimization methodology

1. Introduction

Stainless steel (SS-316L) has been used extensively in a variety of applications in industries along with everyday life, due to its low corrosion rate and excellent mechanical properties [1]. It has been employed as a substrate for TiO_2 films due to its electrical properties [2–4].

In recent decades, various coating methods have been employed to deposit titanium dioxides such as sputtering [5], spin coating [6], dip coating [7], anodization [8–10], chemical vapor deposition (CVD) [11], plasma-enhanced CVD (PECVD) [12], atomic layer deposition (ALD) [13], physical vapor deposition (PVD) [14], and pulsed laser deposition [15]. The Electrophoretic Deposition (EPD) technique, well-matched with varied applications, has recently attracted great attention to process innovative materials such as ceramics, metals, living cells, biological materials, and polymers [16,17] because of its low cost, high productivity, short processing time, basic equipment, and simple set up.



During the EPD process, the charged particles in the suspension are moved under a direct current electric field to be deposited over the oppositely charged electrode [18]. Up to now, few studies have considered EPD of TiO_2 by organic suspensions [19,20] or mixtures of organic solvents and water [21,22]. An aqueous medium can cause the formation of bubbles in the deposit due to the electrochemical reactions at the surface of the electrodes. To overcome this problem, pulsed voltage could be used instead of direct voltage. Besra et al. [23] have confirmed based on their study on aqueous EPD of alumina suspension that using pulsed direct current as an alternative of continuous direct current is a suitable and effective approach to control and suppress the bubble formation in the deposit. At proper pulse widths and/or duty cycle, bubble-free deposits were attained. Naturally, adhesion between TiO_2 and bare SS is poor due to incompatibility between these two [24,25]. To overcome this adhesion problem, the surface needs to be etched chemically. Among the conventional applied techniques is the creation of a conversion layer (CL) before the coating [26].

In this present work, pulsed EPD of TiO_2 coating on an SS substrate with a CL has been studied. Pulsed EPD is an interesting process to deposit a controlled coating by adjustment of the deposition parameters. Limited works have been prepared with the aim to see the effect of EPD factors on the photocatalytic degradation of pollutants [27,28]. The impact of these parameters on decolorization efficiency and adhesion of the film must be studied to find the optimal conditions for the deposition of the TiO_2 layer.

The majority of researchers have defined their experimental results according to the influence of one parameter at a time [29]. This research aims to evaluate the effects of pulsed EPD parameters: applied voltage, deposition time, initial concentration, and duty cycle.

Among the methodologies to achieve the optimum results is the Response Surface Methodology (RSM) [28,30,31]. An experimental design methodology must be economical to extract the maximum amount of complex information, a significant reduction in experimental time, thus saving material as well as personnel cost. This is the first reported study dealing with the production of TiO₂ nanoparticles and the optimization of pulsed EPD parameters to accomplish maximum decolorization efficiency and critical charge values based on a set of pulsed EPD parameters.

In a previous experiment [26], the conversion layer has led to improving the film adhesion and resulted in a degradation percentage of 60%. The key objective of this work is to develop a practical model that predicts maximum decolorization efficiency and critical charge values based on a set of pulsed EPD parameters. The optimization method was based on RSM followed by a desirability optimization methodology as a practical technique when optimizing multiple responses [32,33]. The current technique is an interesting process to deposit a low-cost active photocatalyst that showed no adhesion difficulty.

2. Materials and Methods

2.1. Substrate Preparation

A raw sheet of stainless steel (316L) was cut into discs of 3 mm thickness and 15 mm diameter. Substrates were first polished by 400, 600, 800, and 1200 sandpaper grits and cleaned using deionized water. Then, they were dried in the air. The result was a smooth surface. Afterward, the substrates were ultrasonically cleaned in acetone for 10 min. Next, they were immersed in a conversion bath [26] for 35 min at 60 to 70 °C to produce a passive layer. The conversion bath is composed of 5 mL of sulfuric acid (H₂SO₄) mixed with 0.126 g of hydrated sodium thiosulfate (Na₂S₂O₃·5H₂O) and 0.6 mL of propargylic alcohol (C₃H₄O) in a 100 mL volumetric flask. Then, the samples were rinsed in demineralized water and dried at 120 °C for 60 min. To stabilize the coating, the samples were heat-treated at 450 °C for 2 h [26]. The conversion layer was used to simultaneously increase the adhesion of the TiO₂ film and the efficiency of the photocatalytic treatment.

2.2. Pulsed Electrophoretic Deposition

Pulsed electrophoretic deposition (pulsed EPD) was performed to depose the TiO₂ layer onto the prepared substrate. The setup is described in Figure 1. The counter electrode was made of platinum. The cathode was a SS substrate. The two electrodes were 20 mm apart in a 40 mL glass beaker. The electrolyte was prepared by mixing the TiO₂ powder with 40 mL of deionized water. The powder was prepared by controlled hydrolysis of TiCl₄ followed by dialysis through a cellulose membrane [26]. The mixture was stirred vigorously for 30 min before being used to deposit TiO₂ thin films over pre-functioned stainless steel (316L) with a conversion layer (Fe₂O₃). The deposition was performed at ambient temperature (25 °C). Then, the prepared electrodes were dried in ambient air for 24 h.



Figure 1. Schematic illustration of the pulse circuit generator and working electrophoresis cell [26].

2.3. Characterization of the Film

Scratch tests were carried out for the characterization of adhesion. They were performed at room temperature via the Nanoindenter NHT^2 (CSM Instruments, Peseux, Switzerland) using a Rockwell spherical diamond tip with a radius of 50 μ m. The tests were achieved by progressive load from 30 mN to 15 N. The loading speed was 30 N/min over a distance of 3 mm.

Based on the loads at which cracking and delamination occurred, in addition to the variation in the frictional force, the adhesive force of the coating is determined. The results are confirmed by the optical observations of the scratch track [34,35].

The different adhesion failures can be identified by three critical charges:

 \checkmark *L*_{C1} is the load at which the first cracks occurred (cohesive failure).

 \checkmark L_{C2} is the load at which the film starts to delaminate at the edge level of the scratch track (adhesion failure) [36].

✓ L_{C3} [28] occur when the damage of the film exceeds 50% [37].

2.4. Degradation Experiments

In each experiment, a solution of textile dye: amido black-10B ($C_{22}H_{14}N_6Na_2O_9S_2$; M = 616.5; AB-10B with a concentration of 10 mg/L was used to assess the photocatalytic activity at room temperature. The adsorption equilibrium of AB-10B on the catalysts was reached during 30 min of immersion of the electrode in the dark. Shortly thereafter, the solution was exposed to a UV lamp (HPK 125W, Cathodeon, Cambridge, UK) with λ = 365.5 nm. A noticeable peak at 617 nm and other

small peaks at 226 and 318 nm characterize the absorbance spectra of AB-10B. The concentration of the AB-10B was monitored using a UV-Vis spectrophotometer by examining the main absorbance at 617 nm. The decolorization efficiency (*De*) was calculated as follows (Equation (1)):

$$De(\%) = \left(\frac{Abs_0 - Abs_t}{Abs_0}\right) \times 100\tag{1}$$

where Abs_0 and Abs_t point to the solution absorption at 617 nm at UV irradiation time 0 and t, correspondingly.

2.5. Experimental Design

The multivariate experimental design provides appropriate information about the suitable operating conditions with minimum experimental work. The Central Composite Design (CCD) [38] is a commonly used form of RSM that was employed to investigate the effects of independent variables on the 4 responses (see below).

Four factors were selected as operating parameters in the study. They are the initial TiO_2 concentration, the deposition time, the duty cycle, and the applied voltage referred to as A, B, C, and D, respectively.

The duty cycle is defined by the following equation (Equation (2)):

$$DC = \frac{T_{ON}}{T_{ON} + T_{OFF}}$$
(2)

where T_{ON} is the portion of the cycle for which the voltage is ON and T_{OFF} is the portion of the cycle for which the voltage is null.

By varying T_{ON} and T_{OFF} we managed to adjust the frequency of pulse application at a constant voltage.

The experimental range and levels of these independent variables are listed in Table 1. The experimental domain for each factor was chosen according to prior knowledge of our system [26], the results of other research [39], and the performance of the used photocatalytic reactor [40,41].

Table 1. Experimental range and levels of independent factors used in Central Composite Design (CCD).

| Coded Verichles | Feetere | Linite | Coded Level | | | | | | |
|-----------------|-----------------------|--------|-------------|-----|-----|-----|-----|--|--|
| Coded variables | Factors | Units | $-\alpha$ | -1 | 0 | +1 | +α | | |
| А | Initial Concentration | g/L | 2 | 8 | 14 | 20 | 26 | | |
| В | Time | s | 150 | 300 | 450 | 600 | 750 | | |
| С | Duty Cycle (DC) | % | 10 | 30 | 50 | 70 | 90 | | |
| D | Voltage | V | 4 | 22 | 40 | 58 | 76 | | |

CCD includes three parts: a two-level full (2^k , where k is the number of input factors) or fractional factorial, axial points, and center points [38]. In our case, we used a two-level full factorial design (2^4 points for a four-factor design (k = 4)), 2^{k-1} axial points, and 6 center points. This design allows us to estimate the main effects of input factors and their interactions on output responses [42]. It was adopted to maximize four multiple responses which were the photocatalytic efficiency of the thin film (*De* (%)) and three critical charges L_{C1} (N), L_{C2} (N), and L_{C3} (N).

CCD can have diverse design properties by monitoring alpha " α ": the value of the distance from each axial point to the center of the design. The commonly selected value of α is $\alpha = k^{0.5}$ or $\alpha = (NF)^{0.25}$, with NF being the number of factorial points in a *k* factor design. In our case, α is equal to 2. The factorial points help to estimate the interaction terms. The center points deliver information about the existence of curvature in the method, with which the axial points enable us to estimate the quadratic terms.

Table 2 shows the completely experimental design and real responses of the experiments involved in this study. The 30 manipulations were prepared in random order. We are interested in maximizing all four responses.

| | | Varia | ables | | | Resp | onses | |
|------------------|-----------|------------|-----------|-----------|--------|----------------------------|---------------------|---------------------|
| Stander Order | A | В | С | D | De (%) | <i>L</i> _{C1} (N) | L _{C2} (N) | L _{C3} (N) |
| 1 | -1 | -1 | -1 | -1 | 76.39 | 4.69 | 8.88 | 9.23 |
| 2 | 1 | -1 | -1 | -1 | 83.57 | 3.89 | 7.56 | 9.56 |
| 3 | -1 | 1 | -1 | -1 | 68.77 | 4.67 | 7.32 | 10.74 |
| 4 | 1 | 1 | -1 | -1 | 58.87 | 2.32 | 6.89 | 13.84 |
| 5 | -1 | -1 | 1 | -1 | 57.58 | 4.78 | 9.29 | 11.21 |
| 6 | 1 | -1 | 1 | -1 | 61.96 | 3.24 | 7.52 | 11.47 |
| 7 | -1 | 1 | 1 | -1 | 31.82 | 5.19 | 10.05 | 10.78 |
| 8 | 1 | 1 | 1 | -1 | 29.23 | 2.92 | 6.59 | 12.85 |
| 9 | -1 | -1 | -1 | 1 | 30.96 | 3.61 | 6.53 | 9.64 |
| 10 | 1 | -1 | -1 | 1 | 38.36 | 4.39 | 7.94 | 8.96 |
| 11 | -1 | 1 | -1 | 1 | 43.9 | 2.56 | 5.32 | 10.92 |
| 12 | 1 | 1 | -1 | 1 | 41.68 | 3.25 | 6.36 | 10.45 |
| 13 | -1 | -1 | 1 | 1 | 42.26 | 2.97 | 5.51 | 11.85 |
| 14 | 1 | -1 | 1 | 1 | 50.09 | 3.8 | 6.25 | 9.35 |
| 15 | -1 | 1 | 1 | 1 | 34.42 | 2.94 | 4.48 | 10.3 |
| 16 | 1 | 1 | 1 | 1 | 34.62 | 2.75 | 4.35 | 10.26 |
| 17 | 0 | 0 | 0 | 0 | 52.09 | 3.46 | 6.35 | 9.56 |
| 18 | 0 | 0 | 0 | 0 | 51.89 | 3.56 | 7.34 | 9.17 |
| 19 | 0 | 0 | 0 | 0 | 50.57 | 3.55 | 6.14 | 9.75 |
| 20 | 0 | 0 | 0 | 0 | 53.24 | 3.87 | 6.71 | 9.64 |
| 21 | 0 | 0 | 0 | 0 | 55.09 | 3.52 | 5.98 | 9.43 |
| 22 | 0 | 0 | 0 | 0 | 51.84 | 3.77 | 6.85 | 9.58 |
| 23 | $-\alpha$ | 0 | 0 | 0 | 36.13 | 3.5 | 5.26 | 8.41 |
| 24 | $+\alpha$ | 0 | 0 | 0 | 35.52 | 2.6 | 4.97 | 8.53 |
| 25 | 0 | - α | 0 | 0 | 60.96 | 5.89 | 8.92 | 9.3 |
| 26 | 0 | $+\alpha$ | 0 | 0 | 41.31 | 3.96 | 6.44 | 10.33 |
| 27 | 0 | 0 | $-\alpha$ | 0 | 70.95 | 3.24 | 6.92 | 10.45 |
| 28 | 0 | 0 | +α | 0 | 45.37 | 3.06 | 6.911 | 11.94 |
| 29 | 0 | 0 | 0 | $-\alpha$ | 57.62 | 4.86 | 9.96 | 14.46 |
| 30 | 0 | 0 | 0 | $+\alpha$ | 24.56 | 2.92 | 5.98 | 12.05 |

Table 2. Experimental conditions and actual values of different responses in CCD design.

The Minitab software (version 17.1.0) was applied to carry out the statistical analysis and create the model of the experimental data. The methodology adopted for the experimental modeling involved a sequential study of the experimental results. Initially, the analysis of the observations (De, L_{C1} , L_{C2} , and L_{C3}) was carried out at the end of a complete factorial design 2⁴. Then, the design was improved by 6 center points. Finally, in this study, we present the analysis obtained by a CCD design in which we added the axial points.

The standard error (S), the correlation coefficient (\mathbb{R}^2), and the adjusted correlation coefficient (\mathbb{R}^2 adj) were used to determine the quality of developed models. They show how well the model fits the data. These values can help to choose the most suitable model. The obtained models were approved by ANOVA (analysis of variance), regression analysis, and normal plot of residuals. A *p*-value of less than 0.05 shows that model terms are significant.

The smaller the value, the more important its corresponding coefficient, and the bigger the influence on the response variable [43]. As a result, we can analyze the relationship between the predictor and the response.

When a regression model fails to describe the functional relationship between the response variable and the experimental factors and if important model terms such as quadratic terms or interactions are not considered, we are talking about lack of fit.

The Durbin–Watson statistic (DW-statistic) is used to test the existence of autocorrelation in residuals. It reports a test statistic "d" between 0 and 4. A test statistic equal to 2 signifies no autocorrelation. If it is from 0 to 2, it agrees with positive autocorrelation, and from 2 to 4, we are talking about negative autocorrelation.

The optimization process included evaluating the statistically designed combinations, fitting the experimental data to the predicted responses, calculating the response of the fitted model, and checking the suitability of the model.

3. Results and Discussion

The experimental design was developed to evaluate the effect of the operating parameters (initial concentration, deposition time, *DC*, and applied voltage) on the decolorization efficiency (*De* %) of the TiO₂ film and its three critical charges (L_{C1} , L_{C2} , and L_{C3}).

The statistical data analysis was split into two phases. First, we studied the effect and the interactions of different factors using ANOVA. Second, the results were optimized in such a way to maximize the four responses.

3.1. Analysis of Variance (ANOVA) and Response Surface Analysis

ANOVA analysis of reduced models of the four responses De (%), L_{C1} , L_{C2} , and L_{C3} based on a CCD experimental design is summarized in Table 3. A quadratic model was applied to provide an optimum combination of the pre-treatment parameters.

The experimental design for De (%) in terms of the four independent variables was modeled by the polynomial equations (Equation (3)) with a probability of 5% (p < 0.05).

$$De = 52.549 + 0.461 \text{ A} - 5.715 \text{ B} - 6.320 \text{ C} - 9.084 \text{ D} - 3.756 \text{ A} \times \text{A} + 1.828 \text{ C} \times \text{C} - 2.440 \text{ D} \times \text{D} - 2.581 \text{ A} \times \text{B} - 4.109 \text{ B} \times \text{C} + 5.235 \text{ B} \times \text{D} + 7.094 \text{ C} \times \text{D}$$
(3)

The interactions (AC, AD, and B^2) are removed from the reduced model because they are not relevant. Despite it being irrelevant, factor A was kept in the model for reasons of hierarchy.

The values of R^2 and R^2 (adj) were superior to 97%, which proves the existence of a strong correlation between the experimental results and the reduced model [44].

DW-statistic value of the reduced model was d = 1.76 > dU = 1.64 ($\alpha = 5\%$), thus indicating no serial correlation. The *p*-value of 0.08 indicated that no significant autocorrelation remained in the residuals.

The residual analysis of the response surface design was determined based on the diagnostic plot (Figure 2a).

The plot confirmed that the statistical assumptions fitted the analysis data. Given the normal probability of the residuals, we noticed that the standard deviations between the actual and the predicted response values followed a normal distribution.

The results from Figure 2a correspond to the normal distribution of principal errors. The residuals decreased near to a straight line, highlighting the conformity of the experimental results.

The experimental design for L_{C1} in terms of the four independent variables was modeled by the polynomial equations (Equation (4)) with a probability of 5% (p < 0.05).

$$L_{C1} = 3.6217 - 0.2771 \text{ A} - 0.3596 \text{ B} - 0.0479 \text{ C} - 0.3879 \text{ D} - 0.1647\text{ A} \times \text{A} + 0.3041 \text{ B} \times \text{B} - 0.1397 \text{ C} \times \text{C} + 0.0453 \text{ D} \times \text{D} - 0.2119 \text{ A} \times \text{B} - 0.0931 \text{ A} \times \text{C} + 0.5669 \text{ A} \times \text{D} + 0.1744 \text{ B} \times \text{C} - 0.1106 \text{ B} \times \text{D} - 0.1194 \text{ C} \times \text{D}$$
(4)

| | I | De% | | | | LC | 1 | | | LC2 | | | | L_{C2} | | |
|-----------------------------------|-----------------|-----|-------------|-----------------|-----------------|----|-------------|-----------------|--------------------|-----|-------------|---------|----------------------|----------|-------------|-----------------|
| Source of Variation | Sum of Squares | Df | Mean Square | <i>p</i> -Value | Sum of Squares | Df | Mean Square | <i>p</i> -Value | Sum of Squares | Df | Mean Square | p-Value | Sum of Squares | Df | Mean Square | <i>p</i> -Value |
| А | 5.09682 | 1 | 5.09682 | 0.2039 | 1.8426 | 1 | 1.8426 | 0.0004 | 0.84375 | 1 | 0.84375 | 0.1282 | 0.222337 | 1 | 0.222337 | 0.0664 |
| В | 783.869 | 1 | 783.869 | 0.0000 | 3.1032 | 1 | 3.1032 | 0.0001 | 7.1286 | 1 | 7.1286 | 0.0032 | 4.9777 | 1 | 4.9777 | 0.0001 |
| С | 958.618 | 1 | 958.618 | 0.0000 | 0.0551042 | 1 | 0.0551042 | 0.2039 | 0.321554 | 1 | 0.321554 | 0.3119 | 2.47684 | 1 | 2.47684 | 0.0006 |
| D | 1980.53 | 1 | 1980.53 | 0.0000 | 3.6115 | 1 | 3.6115 | 0.0001 | 26.7126 | 1 | 26.7126 | 0.0002 | 6.7947 | 1 | 6.7947 | 0.0000 |
| $A \times A$ | 395.026 | 1 | 395.026 | 0.0001 | 0.820288 | 1 | 0.820288 | 0.0024 | 3.90084 | 1 | 3.90084 | 0.0112 | 2.02907 | 1 | 2.02907 | 0.0009 |
| $A \times B$ | 106.606 | 1 | 106.606 | 0.0011 | 0.718256 | 1 | 0.718256 | 0.0033 | | | | | 3.28516 | 1 | 3.28516 | 0.0003 |
| A×C | | | | | | | | | 1.7689 | 1 | 1.7689 | 0.0461 | 0.387506 | 1 | 0.387506 | 0.0272 |
| $A \times D$ | | | | | 5.14156 | 1 | 5.14156 | 0.0000 | 6.3001 | 1 | 6.3001 | 0.0042 | 5.58141 | 1 | 5.58141 | 0.0001 |
| $B \times B$ | | | | | 2.47966 | 1 | 2.47966 | 0.0002 | 2.01107 | 1 | 2.01107 | 0.0375 | | | | |
| $B \times C$ | 270.109 | 1 | 270.109 | 0.0001 | 0.486506 | 1 | 0.486506 | 0.0074 | | | | | 4.25391 | 1 | 4.25391 | 0.0002 |
| $B \times D$ | 438.484 | 1 | 438.484 | 0.0000 | 0.195806 | 1 | 0.195806 | 0.0401 | | | | | 1.32826 | 1 | 1.32826 | 0.0023 |
| C×C | 93.5315 | 1 | 93.5315 | 0.0015 | 0.598163 | 1 | 0.598163 | 0.0048 | | | | | 4.75407 | 1 | 4.75407 | 0.0001 |
| C×D | 805.141 | 1 | 805.141 | 0.0000 | 0.228006 | 1 | 0.228006 | 0.0311 | 4.3681 | 1 | 4.3681 | 0.0090 | | | | |
| $D \times D$ | 166.676 | 1 | 166.676 | 0.0004 | | | | | 3.24633 | 1 | 3.24633 | 0.0160 | 24.064 | 1 | 24.064 | 0.0000 |
| Lack-of-fit | 96.7267 | 13 | 7.44052 | 0.1085 | 0.912965 | 12 | 0.0760804 | 0.1206 | 3.17031 | 14 | 0.226451 | 0.6075 | 1.97794 | 12 | 0.164828 | 0.0663 |
| Pure error | 11.9433 | 5 | 2.38867 | | 0.129083 | 5 | 0.0258167 | | 1.27188 | 5 | 0.254377 | | 0.203083 | 5 | 0.0406167 | |
| Total | 6144.9 | 29 | | | 20.8783 | 29 | | | 62.1753 | 29 | | | 63.0796 | 29 | | |
| R ² | 98% | | | | 95% | | | | 92% | | | | 96% | | | |
| R ² (adi) | 97% | | | | 91% | | | | 89.09% | | | | 94% | | | |
| S | 1.54 | | | | 0.16 | | | | 0.504358 | | | | 0.201536 | | | |
| Mean absolute error | 1.58 | | | | 0.15 | | | | 0.297567 | | | | 0.200067 | | | |
| DW-statistic | 1.76 (p = 0.08) | | | | 2.15 (v = 0.43) | | | 2 | 56707 (v = 0.9315) | | | | 2.15913 (v = 0.6006) | | | |
| Lag 1 residual autocorrelation | 0.09 | | | | -0.13 | | | - | -0.297881 | | | | -0.0798249 | | | |

Table 3. Analysis of variance: reduced models of the four responses.

adj: adjusted



Figure 2. Normal probability plot for the four responses: (a) decolorization efficiency De, (b) L_{C1} , (c) L_{C2} , (d) L_{C3} .

The interactions (C, AC, and D^2) were detached from the reduced model because they were not relevant.

The values of R^2 and R^2 (adj) were superior to 91%, thus attesting that there was a strong correlation between the experimental results and the reduced model.

The value of DW-statistic of the reduced model of L_{C1} was d = 2.15 > dU = 1.64 ($\alpha = 5\%$) demonstrating that there is no serial correlation. A *p*-value of 0.43 showed that there was no significant autocorrelation in the residuals.

The diagnostic plot illustrated in Figure 2b shows how the statistical assumptions fit the analysis data. The standard deviations between the actual and the predicted L_{C1} values led to a normal distribution. Moreover, the residuals were near to a straight line, thereby confirming that the experimental results were normal.

The experimental design for L_{C2} , in terms of the four independent variables, was modeled by the polynomial equations (Equation (5)) with a probability of 5% (p < 0.05).

$$L_{C2} = 6.562 - 0.1875 A - 0.5450 B - 0.1158 C - 1.0550 D - 0.3604 A \times A + 0.2808 B \times B + 0.0897 C \times C + 0.3533 D \times D - 0.128 A \times B - 0.332 A \times C + 0.628 A \times D + 0.120 B \times C - 0.207 B \times D - 0.522 C \times D$$
(5)

The interactions (AB, BC, BD, and C^2) were removed from the reduced model because they were not relevant. Although factors A and C were irrelevant, they were saved in the model for causes of hierarchy.

The values of R^2 and R^2 (adj) were superior to 89%, thus demonstrating that a strong correlation was established between the experimental results and the reduced model.

The value of DW-statistic of the reduced model of L_{C2} was d = 2.56 > dU = 1.64 ($\alpha = 5\%$), which suggested no serial correlation. Based on the *p*-value of 0.9315, there was an absence of significant autocorrelation remaining in the residuals. The residual analysis of the response surface design of L_{C2} was determined from the plot illustrated in (Figure 2c).

The standard deviations between the actual and the predicted L_{C2} values produced a normal distribution. Additionally, the residuals were reduced near to a straight line, attesting to the normality of experimental results.

The experimental design for L_{C3} in terms of the four independent variables was modeled by the polynomial equation (Equation (6)) with a probability of 5% (p < 0.05).

$$L_{C3} = 9.522 + 0.0962 A + 0.4554 B + 0.3213 C - 0.5321 D - 0.2580 A \times A + 0.0782 B \times B + 0.4232 C \times C + 0.9382 D \times D + 0.4531 A \times B - 0.1556 A \times C - 0.5906 A \times D - 0.5156 B \times C - 0.2881 B \times D - 0.0719 C \times D$$
(6)

The interactions (CD and B^2) were detached from the reduced model due to irrelevance. Although it is not significant, factor A was presented in the model for causes of hierarchy.

The values of R² and R² (adj) were superior to 94%, thus displaying/attesting to the existence of a strong correlation between the experimental results and the reduced model.

The value of DW-statistic of the reduced model of L_{C3} was d = 2.15 > dU = 1.64 (α = 5%), thus suggesting no serial correlation. Additionally, the *p*-value of 0.6006 proved the absence of significant autocorrelation remaining in the residuals.

The residual analysis of the response surface design of L_{C3} was determined from the plot illustrated in (Figure 2d). The points in the normal probability plots of the residuals were reasonably close to a straight line. This confirmed the normal distribution of the errors and the effect of the input variables on the responses.

In this study, the main effects plots for the fitted values of the four responses are presented in Figure 3. *DC* (%) did not appear in the main effects plots of *De* (%), L_{C1} , and L_{C2} responses because it presented no major effect. Each level of *DC* (%) from 10 to 90% affected the response in the same way, and the average response was the same for all factor levels.



Figure 3. Main effects plots for different responses: (a) degradation, (b) L_{C1} , (c) L_{C2} , and (d) L_{C3} .

In the example of initial concentration, we noticed in Figure 3a that degradation efficiency constantly increased with the increase in the initial concentration of the suspension from 2 to 14 g L⁻¹. This result may be linked to the structural and morphological properties of TiO₂ films, which differ

from one another. Increasing the TiO_2 concentration produced a non-uniform and thick TiO_2 film, thus forming more cracks [45].

The catalyst aggregation caused a smaller number of accessible active sites on the film surface, which explained the decrease in the De (%). The colorant degradation rate De (%) was influenced by the photocatalyst quantity in a photocatalytic process [46].

Increasing the initial concentration led to an increase in the amount of the deposited TiO_2 , and created agglomerates on the substrate's surface. Particles are apt to agglomerate at higher concentrations [47].

We also noticed that the adhesion properties of the film became weaker (Figure 3b–d) when the initial concentration was above 14 g L^{-1} . This was related to the cracks on the surface of the film.

In the case of the applied voltage, we found that in Figure 3a–c, the De (%), L_{C1} , and L_{C2} decreased with the increase in voltage. When the applied voltage was above the water splitting potential, gas evolution was likely to occur at the electrodes. There was formation of H₂-gas on the deposition cathode, which had damaging effects on the coatings. The gas bubbles inhibited good adhesion to the substrate and generated cracks in the deposit [48].

Films with good adhesion and high decolorization efficiency were formed within very short deposition times as seen in Figure 3a–c. Depending on the voltage and the deposition time, the thickness of the film changed [49,50]. As a result, the adhesion of the TiO₂ film to the substrate and the photocatalytic properties were modified. The interaction curves (Figures 4–7) created a matrix of interaction graphs for the factors of each response. This interaction diagram was a diagram of averages for each factor level. However, the level of the second factor was kept constant.



Figure 4. Interaction plots for L_{C1} ((*) refers to multiplication sign (×)).

Furthermore, these figures show that L_{C1} and L_{C2} values increased at low voltages (4–40V), and initial concentrations varied from 8 to 14 g/L. However, starting from about 40V, this phenomenon was inverted. The small decrease in L_C with an increase in applied voltage can be attributed to several factors. To have a regular and homogeneous deposit, the particles must be dispersed. Higher voltage made more cracks on the TiO₂ surface. This suggested that the film adhesion was poor. These mixtures between the TiO₂ concentration and the applied voltages monitor the speed and the height of each pulse throughout the deposition process.

Figure 6 illustrates the interaction plots of different factors on L_{C3} . *DC* (%) was present as the main effect factor for L_{C3} (Figure 3), but it had no interaction with the other factors (Figure 6). This demonstrated that the main effects and interactions were independent of each other. Main effects can exist without interactions.



Figure 5. Interaction plots for L_{C2} ((*) refers to multiplication sign (×)).







Figure 7. Interaction plots for De (%) ((*) refers to multiplication sign (×)).

However, there was a slight interaction between the voltage and the concentration at a constant voltage, and between the time and the concentration at the constant time. At the low voltage and a concentration of 14 g/L, the De (%) was the highest (Figure 7).

There was no interaction between the applied voltage and the concentration ($D \times A$) at constant concentrations. The parallel lines for the interaction plot of $D \times A$ (Figure 7) were high which indicated that the degree of interaction was also high.

The interaction plot for the four responses showed that there was a strong interaction between the initial concentration used for the deposition of the film and the two other relevant factors: time and voltage.

Given the link between the applied voltage (D) and the degradation efficiency (De (%)) of the films, it was clear that at high voltage (58 V), the De (%) was between 22% and 40%, yet it overstepped 50% at low voltage (22 V). The electrophoretic potential affected the nanocrystalline TiO₂ film morphology [48]. The voltage and the initial concentration had the greatest impact on the efficiency of the film.

Electrophoretic deposition produces smooth and homogeneous films. Increasing the voltage yield further produces films that are too thick to produce enough adhesion and contact forces between particles. Consequently, cracks will be formed all over the surface of the film [51]. Additionally, raising the applied voltage reduces the creation of H atom due to a by-production, which cuts off the lattice of the Ti-O-Ti bond. As a result, there is a low quantity of TiO₂ film deposition and a low *De* (%) [52].

In the case of TiO_2 concentration, it was found that the *De* (%) continuously increased with the increase in TiO_2 concentration from 8 to 20 g/L and also at a deposition time between 150 and 450 s. Nevertheless, we noticed a greater degradation when the deposition time was over 450 s and the concentration was as low as 8 to 14 g/L. This effect may be linked to the morphological properties of TiO_2 films.

When the time of deposition increased, De (%) simply decreased. This negative effect may be a result of the relationship between deposition time and the deposited weight of TiO₂ on the substrates.

3.2. Optimization and Verification of the Results

The polynomial models (Equations (3)–(6)) obtained in this study were used for the experimental optimization of the process parameters (initial concentration, time, *DC*, and voltage) to maximize photocatalytic efficiency (*De* (%)) and the adhesion properties (L_{C1} , L_{C2} , and L_{C3}) of the TiO₂ film. The graphical representations of the experimental *De* (%), L_{C1} , L_{C2} , and L_{C3} versus the predicted values are shown in Figure 8.

There was a resemblance between the experimental data and the expected results in the four responses. The desirability method [53] was applied by changing each expected response, Y_i , into a function without units confined by $0 < d_i < 1$.

The d_i values for L_{C1} , L_{C2} , and L_{C3} were equal to 1, which gives more credibility to the responses. However, if $d_i = 0$, this shows an undesired response [53]. Usually, the desirability plot characterizes the optimal region that fulfills the essential predetermined conditions.

The main objective of this work is the deposition of a TiO_2 film on treated SS substrate under optimal process conditions to maximize decolorization efficiency and film adhesion to the substrate. The single desirability for each response d_i was calculated using Equations (3) to (6). The optimization function in the Minitab software was applied to achieve optimal conditions.

The vertical red lines in the graph represent the current factor settings as well as the displayed numbers on the top of the column. The horizontal blue lines and numbers indicate the responses for the current factor level.

The ultimate goal of this optimization was to achieve maximum values of different responses that altogether fulfilled all desirable conditions. This objective was reached based on two steps. First, obtaining the desirability for all the responses (De, L_{C1} , L_{C2} , and L_{C3}), then maximizing the desirability and optimal value identification.



Figure 8. Graphical representations of the experimental values of (**a**) AB-10B decolorization efficiency (%), (**b**) L_{C1} , (**c**) L_{C2} , and (**d**) L_{C3} values plotted against the predicted values derived from the central composite design resulting equation.

Table 4 showed the input variables, their limits, and their goals settings. Different solutions can be attained in a desirability-based approach. The ones with high desirability are selected. Figure 9 shows the results of the desirability function approach.



Figure 9. Optimization plot of multiple responses under the defined parameters: (**A**) initial concentration, (**B**) time, (**C**) duty cycle (DC), and (**D**) voltage.

| Response | Goal | Lower | Target | Upper | Weight | Importance |
|----------|---------|-------|--------|-------|--------|------------|
| L_{C3} | Maximum | 8.41 | 14.46 | 14.46 | 1 | 1 |
| L_{C2} | Maximum | 4.35 | 10.05 | 10.05 | 1 | 1 |
| L_{C1} | Maximum | 2.32 | 5.89 | 5.89 | 1 | 1 |
| De (%) | Maximum | 24.56 | 83.57 | 83.57 | 1 | 1 |

Table 4. Goals and parameter range for optimization of deposition levels.

The optimum levels under the defined parameters were found to be a concentration of 16.34 g/L of TiO_2 , a deposition time of 150 s, a duty cycle of 90%, and an applied voltage of 4 V. At optimum conditions, the model predicted an 82.75% decolorization efficiency of AB-10B. The lack-of-fit was greater than 0.05 in all four responses, implying that there was no difference between the experimental and predicted models.

The use of low voltage (4 V) was very common in the aqueous system [54]. However, water electrolysis might occur. No gas was developed on the surface of the electrode because the deposition time (150 s) was too small. If the process is well controlled, there will be an adequate formation of cracks on the surface of the deposited film. This will be justified by SEM images. The electrophoresis mobility was affected by the initial concentration [39]. Consequently, a concentration of 16 g/L was sufficient to have a uniform TiO_2 film at low voltage and in a short deposition time.

At a low duty cycle, the T_{ON} time was too short. The alternative relationship between T_{ON} and T_{OFF} considerably determined the development of the TiO₂ film. Besra et al. [55] believed that the gas progress in the case of pulse voltage electrolysis was an active process where the gas generation on the electrode surface changed after each T_{OFF} and may be different from previous and following ones. Such phenomena will produce separate micro- and nanosized bubbles homogeneously dispersed.

The coalescence of small bubbles was energetically unfavorable, and their incorporation in the deposit produced no macro-bubbles. The 90% duty cycle was nearly collinear with the constant voltage because the long duty cycle approached a constant voltage deposition [56].

We reproduced the deposition of TiO_2 film in optimum operation levels four times (Table 5) to approve the validity of the suggested models.

| | Optimal Responses | Exp 1 | Exp 2 | Exp 3 | Exp 4 |
|----------|--------------------------|-------|-------|-------|-------|
| L_{C1} | 5.9 | 5.2 | 4.8 | 5.3 | 5 |
| L_{C2} | 12.5 | 10.5 | 11.3 | 12 | 11.8 |
| L_{C3} | 16.7 | 14.3 | 14 | 14.8 | 14.5 |
| De (%) | 82.75 | 75.3 | 78 | 86 | 81 |

Table 5. Validation of the optimum conditions.

The average decolorization of AB-10B from the confirmation experiments was 80.075%, which agrees with the projected value of 82.75%. Therefore, the optimum point determined by RSM was effectively confirmed.

Figure 10a shows the SEM micrographs of the nanocrystalline TiO_2 films deposited in optimization conditions. Decreasing the voltage further produced a film that was too thin. The adhesion and contact forces between particles were better. As a result, we obtained a smooth and crack-free film.

Too high electric-field intensity when applying a pulsed EPD can create an extremely high deposition rate such that the TiO_2 particles will not be deposited at appropriate positions, resulting in poor thickness homogeneity.

Additionally, Figure 10b shows that TiO_2 thin films consist of small aggregated particles, resulting in a highly porous volume. RSM helped to optimize the photocatalytic decolorization of AB-10B with TiO_2 films and provide good adhesion properties of the prepared film using pulsed EPD.



Figure 10. SEM images of the TiO₂/ CL at optimum operating conditions: (a) Scale of 100 μ m, and (b) Scale of 20 μ m.

4. Conclusions

The photocatalytic performance of TiO₂ films has been greatly improved due to the microstructure of the surface. In this work, a novel supported photocatalyst was prepared by TiO₂ deposition using pulsed EPD. The effects of different parameters (deposition time, applied voltage, initial concentration, and duty cycle) on the thin film properties were investigated. The statistical design was used for the multi-response optimization of TiO₂ thin films. We tried to maximize the photocatalytic efficiency (*De* (%)) and the adhesion properties (L_{C1} , L_{C2} , and L_{C3}) of TiO₂ films. An empirical relationship between the response and independent variables was established. The experimental design of the responses according to the four independent variables was modeled by polynomial equations with a probability of 5%. Both the correlation coefficient (R²) and the adjusted correlation coefficient (R²adj) values were good, which meant that the proposed quadratic models matched the experimental data very well.

The optimum levels were found to be a concentration of 16.34 g/L of TiO_2 , a deposition time of 150 s, a duty cycle of 90%, and an applied voltage of 4 V. The degradation efficiency reached 82.75% after maximum desired responses.

In this study, the projected multi-objective optimization approach proved to be a powerful methodology. Enhanced modeling steps help to obtain the best correlation. It offers to scientific researchers and industrials a supportive multi-objective optimization technique in case of various combinations of input.

Author Contributions: Conceptualization, N.B. and A.B.Y.; Methodology, N.B., A.B.Y., M.A.R. and M.A.-A.; Formal analysis, N.B., M.A.R., M.A.-A. and A.B.Y.; Investigation, N.B., A.B.Y. and M.A.R; Resources, N.B., M.A.R, M.A.-A. and A.B.Y.; Writing—original draft preparation, N.B., A.B.Y. and L.B.; Writing—review and editing, N.B., M.A.R, M.A.-A. and A.B.Y.; Project administration, N.B., A.B.Y., M.A.-A. and L.B. All authors have read and agreed to the published version of the manuscript.

Funding: This research received no external funding.

Acknowledgments: The authors would like to thank the German Jordanian University represented by the Deanship of Graduate Studies and Scientific Research for supporting this work.

Conflicts of Interest: The authors declare no conflict of interest.

References

- Gao, F.; Sherwood, P. Photoelectron spectroscopic studies of the formation of hydroxyapatite films on 316L. Surf. Interface Anal. 2012, 44, 1587–1600. [CrossRef]
- Choy, C.; Shan, X.; Hou, K. Corrosion resistance of TiO₂ films grown on stainless steel by atomic layer Deposition. *Surf. Coat. Technol.* 2008, 202, 2399–2402.

- 3. Krishna, D.; Chena, Y.; Sun, Z. Magnetron sputtered TiO2 films on a stainless steel substrate: Selective rutile phase formation and its tribological and anti-corrosion performance. *Thin Solid Films* **2011**, *519*, 4860–4864. [CrossRef]
- de Sousa, R.R.M.; de Araújo, F.O.; da Costa, J.A.P.; Nishimoto, A.; Viana, B.C.; Alves, C., Jr. Deposition of TiO₂ film on duplex stainless steel substrate using the cathodic cage plasma technique. *Mater. Res.* 2016, 19, 1207–1212. [CrossRef]
- 5. Pawar, N.; Bhargava, A.; Dayal, S.; Kumar, C. Synthesis and characterization of pure anatase phase nanocrystalline TiO₂ thin film by magnetron sputtering method. In Proceedings of the AIP Conference Proceedings, Bikaner, India, 30–31 October 2015; AIP Publishing: Rajasthan, India, 2016; Volume 1728, p. 020561.
- Chen, W.; Koshy, P.; Sorrell, C. Effects of film topology and contamination as a function of thickness on the photo-induced hydrophilicity of transparent TiO₂ thin films deposited on glass substrates by spin coating. *J. Mater. Sci.* 2016, *51*, 2465–2480. [CrossRef]
- Vanaraja, M.; Muthukrishnan, K.; Boomadevi, S.; Karn, R.; Singh, V.; Singh, P.; Pandiyan, K. Dip coated TiO₂ nanostructured thin film: Synthesis and application. *Phase Transit. A Multi. J.* 2016, *89*, 107–114. [CrossRef]
- Atyaoui, A.; Bousselmi, L.; Cachet, H.; Peng, P.; Sutter, E. Influence of geometric and electronic characteristics of TiO2 electrodes with nanotubular array on their photocatalytic efficiencies. *J. Photochem. Photobiol. A Chem.* 2011, 224, 71–79. [CrossRef]
- 9. Atyaoui, A.; Cachet, H.; Sutter, E.; Bousselmi, L. Effect of the anodization voltage on the dimensions and photoactivity of titania nanotubes arrays. *Surf. Interface Anal.* **2013**, *45*, 1751–1759. [CrossRef]
- Li, T.; Ding, D. Ni/Si-codoped TiO₂ nanostructure photoanode for enhanced photoelectrochemical water splitting. *Materials* 2019, 112, 4102. [CrossRef]
- 11. Zhou, M.; Roualdès, S.; Zhao, J.; Autès, V.; Ayral, A. Nanocrystalline TiO₂ thin film prepared by low-temperature plasma-enhanced chemical vapor deposition for photocatalytic applications. *Thin Solid Films* **2015**, *589*, 770–777. [CrossRef]
- 12. Wang, N.; Wang, J.; Zheng, F.; Wu, Y.; Hou, B. A comparison of diamond-like carbon films properties obtained by plasma enhanced chemical vapor deposition and electro-deposition. *Mater. Sci. Forum* **2016**, *852*, 1029–1033. [CrossRef]
- 13. Sasinska, A.; Singh, T.; Wang, S.; Mathur, S.; Kraehnert, R. Enhanced photocatalytic performance in atomic layer deposition grown TiO₂ thin films via hydrogen plasma treatment. *J. Vac. Sci. Technol.* **2015**, *33*, 01A152. [CrossRef]
- 14. Eshaghi, A.; Aghaei, A. Determination of optical properties in nanostructured TiO₂ thin film fabricated by electron beam physical vapor deposition. *J. Opt. Technol.* **2016**, *83*, 26–29. [CrossRef]
- Ruzybayev, A.I.; Yassitepe, E.; Karim, A.; Shah, S.I.; Bhatti, A. Phase transformations in the pulsed laser deposition grown TiO₂ thin films as a consequence of O2 partial pressure and Nd doping. *J. Phys. Chem. C* 2015, *119*, 11578–11587.
- 16. Purcell-Milton, F.; Curutchet, A.; Gun'ko, Y. Electrophoretic deposition of quantum dots and characterisation of composites. *Materials* **2019**, *12*, 4089. [CrossRef]
- 17. Fiołek, A.; Zimowski, S.; Kopia, A.; Łukaszczyk, A.; Moskalewicz, T. Electrophoretic Co-deposition of polyetheretherketone and graphite particles: Microstructure, electrochemical corrosion resistance, and coating adhesion to a titanium alloy. *Materials* **2020**, *13*, 3251. [CrossRef]
- Pascall, A.J.; Mora, J.; Jackson, J.; Kuntz, J. Light directed electrophoretic deposition for additive manufacturing: Spatially localized deposition control with photoconductive counter electrodes. *Key Eng. Mater.* 2015, 654, 261–267. [CrossRef]
- 19. Lin, C.K.; Yang, T.; Feng, Y.; Tsung, T.; Su, C. Characterization of elecrophoretically deposited nanocrystalline titanium dioxide films. *Surf. Coat. Technol.* **2006**, *200*, 3184. [CrossRef]
- 20. Santillán, M.; Membrives, F.; Quaranta, N.; Boccaccini, A. Characterization of TiO₂ nanoparticle suspensions for electrophoretic deposition. *J. Nanoparticle Res.* **2008**, *10*, 787. [CrossRef]
- 21. Lebrette, S.; Pagnoux, C.; Abélard, P. Stability of aqueous TiO₂ suspensions: Influence of ethanol. *J. Colloid Interface Sci.* **2004**, *280*, 400. [CrossRef]
- 22. Hanaor, D.; Michelazzi, M.; Veronesi, P.; Leonelli, C.; Romagnoli, M.; Sorrell, C. Anodic aqueous electrophoretic deposition of titanium dioxide using carboxylic acids as dispersing agents. *J. Eur. Ceram. Soc.* **2011**, *31*, 1041–1047. [CrossRef]

- 23. Besra, L.; Uchikoshi, T.; Suzuki, T.; Sakka, Y. Application of constant current pulse to suppress bubble incorporation and control deposit morphology during aqueous electrophoretic deposition (EPD). *J. Eur. Ceram. Soc.* **2009**, *29*, 1837–1845. [CrossRef]
- 24. Tang, F.; Uchikoshi, T.; Ozawa, K.; Sakka, Y. Effect of polyethylenimine on the dispersion and electrophoretic deposition of nano-sized titania aqueous suspensions. *J. Eur. Ceram. Soc.* **2006**, *26*, 1555–1560. [CrossRef]
- 25. Nurhayati, E.; Yang, H.; Chen, C.; Liu, C.; Juang, Y.; Huang, C.; Hu, C. Electro-photocatalytic fenton decolorization of orange G using mesoporous TiO₂/stainless steel mesh photo-electrode prepared by the sol-gel dip-coating method. *Int. J. Electrochem. Sci.* **2016**, *11*, 3615. [CrossRef]
- 26. Barbana, N.; Youssef, A.B.; Dhiflaoui, H.; Bousselmi, L. Preparation and characterization of photocatalytic TiO₂ films on functionalized stainless steel. *J. Mater. Sci.* **2018**, *53*, 3341–3364. [CrossRef]
- 27. Toh, A.; Cai, R.; Butler, D. The influence of surface topography on the photocatalytic activity of electrophoretically deposited titanium dioxide thin films. *Wear* **2009**, *266*, 585–588. [CrossRef]
- Khataee, A.R.; Fathinia, M.; Aber, S.; Zarei, M. Optimization of photocatalytic treatment of dye solution on supported TiO2nanoparticles by central compositedesign: Intermediates identification. *J. Hazard. Mater.* 2010, 181, 886–897. [CrossRef]
- 29. Alim, M.; Lee, J.; Akoh, C.; Choi, M.; Jeon, M.; Shin, J.; Lee, K. Enzymatic transesterification of fractionated rice bran oil with conjugated linoleic acid: Optimization by response surface methodology. *LWT Food Sci. Technol.* **2008**, *41*, 764–770. [CrossRef]
- 30. Vaez, M.; Moghaddam, A.; Alijani, S. Optimization and modeling of photocatalytic degradation of azo dye using a response surface methodology (RSM) based on the central composite design with immobilized titania nanoparticles. *Ind. Eng. Chem. Res.* **2012**, *51*, 4199–4207. [CrossRef]
- 31. Shafiq, N.; Kumar, R.; Zahid, M.; Tufail, R. Effects of modified metakaolin using nano-silica on the mechanical properties and durability of concrete. *Materials* **2019**, *12*, 2291. [CrossRef]
- 32. Derringer, G. Balancing act: Optimizing a product's properties. Qual. Prog. 1994, 27, 51–58.
- 33. Bansal, P.; Verma, A.; Mehta, C.; Singla, J.; Toor, A. Assessment of integrated binary process by coupling photocatalysis and photo-Fenton for the removal of cephalexin from aqueous solution. *J. Mater. Sci.* **2018**, *53*, 7326–7343. [CrossRef]
- 34. Seo, T.; Weon, J. Influence of weathering and substrate roughness on the interfacial adhesion of acrylic coating based on an increasing load scratch test. *J. Mater. Sci.* **2012**, *47*, 2234–2240. [CrossRef]
- 35. Hare, B.A.; Moyse, A.; Sue, H. Analysis of scratch-induced damages in multi-layer packaging film systems. *J. Mater. Sci.* **2012**, 47, 1389–1398. [CrossRef]
- Vaz, F.; Machado, P.; Rebouta, L.; Cerqueira, P.; Goudeau, P.; Rivière, J.; Alves, E.; Pischow, K.; Rijk, J. Mechanical characterization of reactively magnetron-sputtered TiN films. *Surf. Coat. Technol.* 2003, 174–175, 375–382. [CrossRef]
- 37. Oliver, W.; Pharr, G. An improved technique for determining hardness and elastic modulus using load and displacement sensing indentation experiments. *J. Mater. Res.* **1992**, *7*, 1564. [CrossRef]
- 38. Box, G.E.P.; Wilson, K.B. On the experimental attainment of optimum conditions. *J. R. Stat. Soc. Ser. B* **1951**, 13, 1–45. [CrossRef]
- 39. Hamaker, H.; Verwey, E. The role of the forces between the particles in electrodeposition and other phenomena. *Trans. Faraday Soc.* **1940**, *36*, 180–185. [CrossRef]
- 40. Nyongesa, F.; Aduda, B. Electrophoretic deposition of titanium dioxide thin films for photocatalytic water purification systems. *Adv. Mater.* **2017**, *6*, 31–37. [CrossRef]
- 41. Mohammadi, M.; Vossoughi, M.; Feilizadeh, M.; Rashtchian, D.; Moradi, S.; Alemzadeh, I. Effects of electrophoretic deposition parameters on the photocatalytic activity of TiO₂ films: Optimization by response surface methodology. *Colloids Surf. A Physicochem. Eng. Asp.* **2014**, 452, 1–8. [CrossRef]
- 42. Bezerra, M.; Santelli, R.; Oliveira, E.; Villar, L.; Escaleira, L. Response surface methodology (RSM) as a tool for optimization in analytical chemistry. *Talanta* **2008**, *76*, 965–977. [CrossRef] [PubMed]
- 43. Adeli, H.; Zein, S.; Tan, S.; Ahmad, A. Optimization of the mechanical strength properties of Poly (Llactide)/multiwalled carbon nanotube scaffolds using response surface methodology. *Nano Brief Rep. Rev.* **2011**, *6*, 113–122.
- 44. Fu, J.; Zhao, Y.; Wu, Q. Optimising photoelectrocatalytic oxidation of fulvic acid using response surface methodology. *J. Hazard. Mater.* **2007**, 144, 499–505. [CrossRef] [PubMed]
- 45. Benehkohal, N.P. Innovations in Electrophoretic Deposition of Nanotitania-Based Photoanodes for Use in Dye-Sensitized Solar Cells. Ph.D. Thesis, McGill University, Montreal, QC, Canada, 2013.

- Chiou, C.; Juang, R. Photocatalytic degradation of phenol in aqueous solutions by Pr-doped TiO₂ nanoparticles. *J. Hazard. Mater.* 2007, 149, 1–7. [CrossRef] [PubMed]
- 47. Noorsuhana, M.Y.; Hidayah, A.A.N. Electrophoretic deposition of titanium dioxide (TiO₂) nanoparticles on ceramic membrane. *Adv. Mater. Res.* **2014**, 974, 20–25. [CrossRef]
- 48. Uchikoshi, T.; Ozawa, K.; Hatton, B.; Sakka, Y. Dense bubble-free ceramic deposits from aqueous suspensions by electrophoretic deposition. *J. Mater. Res.* **2001**, *16*, 321. [CrossRef]
- 49. Besra, L.; Liu, M. A review on fundamentals and applications of electrophoretic deposition (EPD). *Prog. Mater. Sci.* 2007, 52, 1–61. [CrossRef]
- 50. Abdullah, H.; Sorrell, C.C. TiO2 thick films by electrophoretic deposition. J. Aust. Ceram. Soc. 2008, 44, 12–16.
- Chang, H.; Su, H.; Chen, W.; Huang, K.D.; Chien, S.; Chen, S.; Chen, C. Fabrication of multilayer TiO₂ thin films for dye-sensitized solar cellswith high conversion efficiency by electrophoresis deposition. *Sol. Energy* 2010, *84*, 130–136. [CrossRef]
- 52. Zhang, Y.; Li, Q. Synthesis and characterization of Fe-doped TiO2films by elec-trophoretic method and its photocatalytic activity toward methyl orange. *Solid State Sci.* **2013**, *16*, 16–20. [CrossRef]
- 53. Benyounis, K.; Olabi, A.; Hashmi, M. Multi-response optimization of CO₂ laser welding process of austenitic stainless steel. *Opt. Laser Technol.* **2008**, 40, 76–87. [CrossRef]
- 54. Dzepina, B.; Sigalas, I.; Herrmann, M.; Nilen, R. The aqueous electrophoretic deposition (EPD) of diamond-diamond laminates. *Int. J. Refract. Met. Hard Mater.* **2013**, *36*, 126–129. [CrossRef]
- 55. Besra, L.; Uchikoshi, T.; Suzuki, T.; Sakka, Y. Bubble-free aqueous electrophoretic deposition (EPD) by pulse-potential application. *J. Am. Ceram. Soc.* **2008**, *91*, 3154. [CrossRef]
- 56. Palmer, T. Investigation of Electrophoretic Deposition as a Fabrication Technique for High Performance Composites. Master's Thesis, Massachusetts Institute of Technology, Cambridge, MA, USA, 2009.

Publisher's Note: MDPI stays neutral with regard to jurisdictional claims in published maps and institutional affiliations.



© 2020 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (http://creativecommons.org/licenses/by/4.0/).