Upconversion Quantum Yield and Luminescence of β -NaYF₄:Yb³⁺,Er³⁺ Nanoparticles: Influence of Environment and Dopant Concentration

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This thesis is dedicated to my daughter, Sophie Kaiser.
"Science is the process that takes us from confusion to understanding."

Brian Greene

Abstract

Lanthanide-based upconversion (UC) nanometer (nm)-sized particles (UCNPs) exhibit the unique ability to emit one higher-energy photon after the sequential absorption of two or more near-infrared (NIR) photons. This UC luminescence (UCL) makes UCNPs very attractive for the use as optical probes in the biomedical area. In this respect, the main advantages of UCNPs are the suppression of environment-based background fluorescence as well as their excitability with NIR-light allowing imaging applications in deep tissue layers of several centimeters (cm).

The future progress in the development of brightly-luminescent UCNPs requires reliable quantification methods of their optical properties to ensure comparability on an international level. For this purpose, in the first study of this thesis an integrating sphere setup (ISS), as well as corresponding relevant guidelines for the measurement conditions, has been developed enabling the absolute determination of the key parameter upconversion quantum yield Φ_{UC} with minimum uncertainty (Φ_{UC} := ratio of emitted high-energy photons to absorbed low-energy photons). The main features of the developed ISS setup present the wide tunability of P of over four orders of magnitude and a high linear detection range of over 10 orders of magnitude allowing the comprehensive characterization of the nonlinear luminescence behavior of UC materials. As the main subject of the investigations of this thesis served a UC system based on a transparent host lattice β -NaYF₄, which was doped with trivalent ytterbium ions (Yb3+), acting as absorber/antenna, and trivalent erbium ions (Er3+), acting as emitter. The suppression of radiationless deactivation processes via the particle surface presents a major challenge in the development of brightly-luminescent UCNPs. In order to quantify the luminescence quenching of UCNPs, commercial µm-sized β-NaYF₄:Yb³⁺(21%),Er³⁺(2%) particles (UC μ P), with particle diameters large enough to neglect surface effects, were comprehensively optically characterized. In this respect, a maximal absolute Φ_{UC} of 10.5% at $P=30~{\rm Wcm^{-2}}$, considering the spectral region from 360 - 900 nm, was measured for these UCµP.

New detailed insights of the UC processes of β -NaYF₄:Yb³⁺,Er³⁺ UCNPs were gained in the second and third study by the investigation of high-quality UCNPs sample series with systematical varied parameters. These UCNPs sample series were optically characterized regarding their *P*-dependent Φ_{UC} and UCL of the different Er³⁺ emission bands as well as of the decay behavior of the intensity of the Er³⁺ and Yb³⁺ emission bands. The computer-assisted simulation of these experimental data by utilizing a coupled rate equation system turned out to be a powerful tool to underpin the photophysical interpretations. A special

highlight of this advanced analysis represents the clarification of the conditions for the controversially debated bi- and triphotonic population pathways of the red-emitting $Er^{3+4}F_{9/2}$ energy level. In the following, the main results of these studies are briefly presented.

The second study deals with 23 nm-sized UCNPs with different surrounding environments, its interaction with the particle surface, and the accompanied influence on the UC processes. A main conclusion of this study points out that for UCNPs dispersed in water (H_2O), at low P, the $Er^{3+} {}^4F_{9/2}$ energy level is populated biphotonically by a nonresonant $Yb^{3+}-Er^{3+}$ energy transfer (ET) from the low energetic $Er^{3+} {}^4I_{13/2}$ energy level. Thereby, the high population density of the low Er^{3+} energy levels is favored due to the coupling of excited Er^{3+} ions near the UCNPs particle surface with the O-H vibrational modes of the H_2O molecules. However, the use of very high P of ca. 1 kWcm⁻² compensates H_2O -induced increase of nonradiative relaxation rates, and consequently, leads to moderately high Φ_{DC} of about 0.5% with a mainly triphotonic activation of the $Er^{3+} {}^4F_{9/2}$ energy level for the UCNPs dispersed in H_2O .

In the third study, the optical properties of 33 nm-sized UCNPs dispersed in toluene as function of the ion-ion distances were investigated. These distances can be directly tuned by variation of the Yb³+ and the Er³+ dopant concentrations. In this respect, it was demonstrated that the increase of the Yb³+ dopant concentration causes an enhancement of the triphotonic activation of the red-emitting $Er^{3+} ^4F_{9/2}$ energy level. This effect is induced by an increased back energy transfer from Er^{3+} to Yb³+ due to the reduction of the Er^{3+} -Yb³+-distance. Although the Φ_{UC} is reduced by a faster energy migration to the UCNP surface, the overall UCL intensity was overcompensated due to the higher number of absorbing Yb³+ ions. However, the increase of the Er^{3+} concentration resulted in different trends of the relative Er^{3+} red emission intensity. The intensity increase, observed at low P, was attributed to increased biphotonic activation by enhanced nonradiative relaxation of the green-emitting $Er^{3+} ^2H_{11/2}/^4S_{3/2}$. Contrary to that, at high P, the Er^{3+} red emission intensity diminishes when the Er^{3+} concentration increases. Consequently, a rate equation analysis revealed that this UCL behavior is a sign of a yet unknown depopulation rate.

Zusammenfassung

Lanthanid-basierte upconversion (UC) (oder aufkonvertierende) Nanopartikel (UCNPs) besitzen die spezielle Eigenschaft nach der sequentiellen Absorption von zwei oder mehreren nahinfraroten (NIR) Photonen ein höherenergetisches Photon zu emittieren. Diese UC Lumineszenz (UCL) macht UCNPs sehr attraktiv für den Einsatz als optische Nanosensoren im biomedizinischen Bereich. Hierbei sind die Hauptvorteile von UCNPs die Vermeidung von umgebungsbedingter Hintergrundfloureszenz sowie deren Anregbarkeit mit NIR-Licht, die bildgebende Verfahren in tieferen Gewebeschichten von mehreren Zentimetern (cm) erlaubt.

Der weitere Fortschritt in der Entwicklung von hell-lumineszierenden UCNPs erfordert die verlässliche Quantifizierung ihrer optischen Eigenschaften um die Vergleichbarkeit auf internationaler Ebene zu gewährleisten. Zu diesem Zweck wurde in der ersten Studie dieser Doktorarbeit ein Ulbrichtkugelaufbau (ISS) sowie relevante Richtlinien für die Messbedingungen entwickelt, die die absolute Bestimmung des Schlüsselparameters upconversion Quantenausbeute (Φ_{UC}) mit minimaler Unsicherheit ermöglichen $(\Phi_{UC} := Verhältnis von emittierten hochenergetischen Photonen zu absorbierten$ niederenergetischen Photonen). Die Hauptbesonderheiten dieses ISS Aufbaus sind die weite Durchstimmbarkeit der P über vier Größenordnungen und der große lineare Detektionsbereich von über 10 Größenordnungen, die eine umfassende Charakterisierung des nichtlinearen Lumineszenzverhaltens von UC Materialien gewährleistet. Als Hauptgegenstand dieser Arbeit diente ein UC System basierend auf einen transparenten Wirtskristall β -NaYF₄, der mit trivalenten Ytterbium-Ionen (Yb³⁺), fungieren als Antenne/Absorber, und trivalenten Erbium-Ionen (Er³⁺), fungieren als Emitter, dotiert ist. Die Vermeidung strahlungsloser Deaktivierungsprozesse über die Partikeloberfläche gilt als einer der größten Herausforderungen in der Entwicklung von hell-leuchtenden UCNPs. Zur Quantifizierung solcher Lumineszenzlöscheffekte von UCNPs wurden kommerzielle µmgroße β-NaYF₄:Yb³⁺(21%),Er³⁺(2%) Partikel (UCμP), deren Partikeldurchmesser hinreichend groß ist um Oberflächeneffekte zu vernachlässigen, umfassend optisch charakterisiert. In dieser Hinsicht, wurde für diese UC μ P eine maximale Φ_{UC} von 10.5% bei $P=30~{\rm Wcm}^{-2}$ im spektralen Bereich von 360 nm - 900 nm absolut gemessen.

Neue detaillierte Erkenntnisse über die UC Prozesse von β -NaYF₄:Yb³⁺,Er³⁺ UCNPs wurden in der zweiten und dritten Studie anhand von hochqualitativen Probenserien mit systematischen variierten Parametern erlangt. Diese UCNPs Probenserien wurden

hinsichtlich der P-abhängigen Φ_{UC} und UCL der verschiedenen Er^{3+} Emissionsbanden sowie des Abklingverhaltens der Intensität von den Er^{3+} und Yb^{3+} Emissionsbanden untersucht. Die computerunterstützte Simulation der gewonnenen experimentellen Daten mittels eines gekoppelten Ratengleichungssystems erwies sich als ein mächtiges Werkzeug für die Untermauerung der photophysikalischen Interpretationen. Ein besonderes Highlight dieser fortschrittlichen Analyse präsentiert die Klärung der Bedingungen für die kontrovers diskutierten bi- und triphotonischen Populationswege für das rot emittierende Er^{3+} $^4F_{9/2}$ Energielevel. Im Folgenden sind die Hauptergebnisse dieser Studien kurz dargestellt.

Die zweite Studie beschäftigt sich mit 23 nm-großen UCNPs in verschiedenen Partikelumgebungen, sowie dessen Interaktion mit der Partikeloberfläche und den hiermit verbundenen Einfluss auf die UC Prozesse. Eine Hauptergebnis dieser Studie zeigt auf, dass für UCNPs dispergiert in Wasser (H_2O) das rot emittierenden $Er^{3+} F_{9/2}$ Energielevel bei niedriger P hauptsächlich biphotonisch über einen nichtresonanten P_3 Energielevel bevölkert wird. Hierbei wird die hohe Population der niederenergetischen P_3 Energielevel bevölkert wird. Hierbei wird die hohe Population der niederenergetischen P_3 Energielevel durch die Kopplung von oberflächennahen angeregten P_3 Index mit den P_3 Urbrationsmoden der P_3 Moleküle begünstigt. Indessen kompensierte die Verwendung von hohen P_3 von ca. 1 KWcm-2 die P_3 die P_3 die P_3 induzierten Löschraten, wodurch für die UCNPs dispergiert in P_3 Energielevels erreicht wird.

In der dritten Studie wurden die optischen Eigenschaften von 33 nm-großen UCNPs dispergiert in Toluol in Abhängigkeit der mittleren Ionenabstände untersucht. Diese Abstände wurden über die Variation der Yb³+ und der Er³+ Dotierkonzentration durchgestimmt. In dieser Hinsicht wurde demonstriert, dass die Erhöhung der Yb³+ Konzentrationen zu einer Verstärkung der triphotonischen Aktivierung des rot emittierenden Er³+ 4F9/2 Energielevel führt. Dieser Effekt ist auf einen verstärkten Energierücktransfer vom Er³+ zum Yb³+, aufgrund der Verringerung des Er³+-Yb³+ Abstandes, zurückzuführen. Indessen führte die Erhöhung der Er³+ Konzentration zu verschiedenen Trends für die relative Intensität der roten Er³+ Emissionsbande. Die Intensitätserhöhung bei Anregung mit geringer P konnte auf die Verstärkung der nichtstrahlenden Rate des grün emittierenden Er³+ 2H11/2/4S3/2 Energielevels zurückgeführt werden. Konträr dazu verringerte sich bei Anregung mit hoher P die relative rote Er³+ Emissionsintensität mit steigender Er³+ Konzentration. Die Ratengleichungsanalyse zeigt auf, dass dieses UCL-Verhalten ein Indiz für eine bisher unbekannte Depopulationsrate ist.

List of Publications for this Cumulative Thesis

This cumulative thesis is based on the following publications, which are referred by capital roman numerals:

I. M. Kaiser, C. Würth, M. Kraft, I. Hyppänen, T. Soukka, U. Resch-Genger "Power-dependent upconversion quantum yield of NaYF₄: Yb³⁺, Er³⁺ nano-and micrometer-sized particles – measurements and simulations"

Nanoscale, **2017**, 9, 10051-10058

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II. C. Würth*, M. Kaiser*, S. Wilhelm, B. Grauel, T. Hirsch, U. Resch-Genger "Excitation power dependent population pathways and absolute quantum yields of upconversion nanoparticles in different solvents"

Nanoscale, **2017**, 9 , 4283-4294

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*equally contributed

III. M. Kaiser, C. Würth, M. Kraft, T. Soukka, U. Resch-Genger "Explaining the influence of dopant concentration and excitation power density on the luminescence and brightness of β-NaYF₄:Yb³⁺,Er³⁺ nanoparticles: Measurements and simulations"

Nanoresearch, **2019**, 12, 1871-1879

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1 Introduction

Optical probes are increasingly used in biomedical research to visualize, characterize and quantify processes on cellular und subcellular levels. Such probes enable localized measurements of pH, polarity, viscosity, and the detection of disease-related biomarkers. 1-4 The most extensively investigated and applied chromophores for the design of optical probes are fluorescent organic dyes and colloidal semiconductor nanocrystals called quantum dots (QDs).5-7 Organic dyes and QDs stand out for their excellent brightness resulting from a) a high photon absorption cross-section [cm²], which is a measure of the probability of an incoming photon being absorbed, and b) a high photoluminescence quantum yield (Φ), defined as ratio of emitted photons to absorbed photons. 5 Organic dyes exhibit unsymmetrical-shaped emission bands and a mirrored-shifted absorption band, see Figure 1.1. The emission process stems from optical transitions, which either are localized over the whole molecule or from an intramolecular charge transfer, i.e. that electron is transferred from one part to another part of the dye molecule.^{5, 8} QDs have a Gaussianshaped emission band resulting from the recombination of electron-hole pairs called excitons, which are produced by the absorption of photons with energies larger than the QDs bandgap energy. The absorptivity of QDs increases with decreasing wavelength offering a variable choice of the excitation wavelength, in contrast to organic dyes, see Figure 1.1.5 The emission and absorption properties of QDs are mainly characterized by their material composition, the particle size, and size distribution. Thereby, the decrease of the QDs emission wavelength with shrinking diameter, known as quantum size effect, arises when the QDs diameter (typically a few nanometers) is comparable or smaller than the materialspecific exciton Bohr radius equaling the distance of its electron-hole pairs. 9, 10

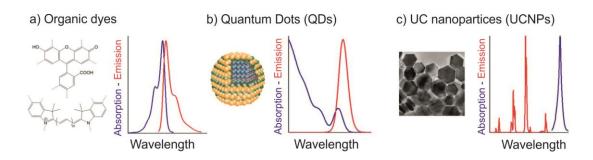


Figure 1.1 Overview of a typical structure, emission band, and absorption band of a) Organic dyes, b) QDs, and c) UCNPs. Reproduced from Würth et al.¹¹ with permission from Springer Nature.

Despite the great success of these emitters in biomedical applications, their potential is still hampered due to the lack of stable, efficient near-infrared (NIR) emissive organic dyes and QDs. 12 Therefore, the common impairments of visible (vis) light emitting optical probes, including scattering, absorption, and background fluorescence of surrounding biological matrices, are restricting their scope of applications, e.g. the use in deep tissue imaging. A further critical issue of organic dyes concern their low photostability, i.e. that they bleach after long-term irradiation, which is very problematic for single particle tracking or long-term studies in a medium or in living cells. Although QDs exhibit a high photostability, their blinking behavior, i.e. that they can spontaneously switch from being emissive to a dark state, is particularly undesirable for single particle tracking applications. Furthermore, the assessment of the toxicity of these materials for in vitro and in vivo studies is an ongoing controversial topic. In this respect, dyes have been widely characterized showing cytotoxicity levels varying from low to very high. 5 QDs, typically consisting of toxic heavy metals (e.g. Cd2+, Hg²⁺, Zn²⁺, Pb²⁺), require a very stable surface chemistry as surface cover.^{7, 13, 14} However, their nanotoxicological effects, i.e. that they may accumulate inside cells and damage them as well as a potentially high surface reactivity, are still not properly understood.

Trivalent lanthanide-(Ln3+)-based upconversion (UC) nanoparticles (UCNPs) are an emerging class of optical probes with a high potential of outperforming fluorescent dyes and QDs in many application areas. 15-31 UCNPs already demonstrated promising results as contrast agent for deep tissue super-resolution imaging and spectroscopy, 32, 33 as an optical switch to control mouse brain activity³⁴ in the optogenetics research area, and as active media for micro-cavity lasers 35 with ultra-low threshold down to 30 Wcm-2. The most efficient UCNPs materials are based on a combination of energy harvesting Ln³⁺ ions called sensitizer ions (e.g. Yb³⁺, Nd³⁺) and emitting Ln³⁺ ions called activator ions (e.g. Er³⁺,Tm³⁺,Ho³⁺) embedded in an inorganic transparent host material. ^{17, 21, 30} The parity forbidden optical 4f-4f transitions of the Ln³⁺ exhibit long luminescence lifetimes > 100 μs, and hence, UCNPs require rather low excitation power densities (P) in the 1 Wcm⁻² - 10³ Wcm⁻¹ ². Such low P can be easily provided by a low-cost continuous-wave laser diode. Moreover, the UC luminescence (UCL) of UCNPs promise a very high signal-to-noise ratio due to the circumvention of the detection of background fluorescence from surrounding biological matrices, a large wavelength shift between emission and excitation (ranging from 130 nm to 500 nm) called anti-Stokes shift and several sharp tunable emission bands with line width < 20 nm (see Figure 1.1). 17, 25 Additionally, their long luminescence lifetimes allow canceling out sources of interference by temporal discrimination called time-gated emission spectroscopy. 17, 21, 30 The NIR-light excitation of UCNPs offers a deep penetration depth in

tissue connected with low photodamage on cells compared to the vis-light excitation used for dyes and QDs. However, UCNPs exhibit rather low photon absorption cross-section compared to organic dye and QDs, which is a major challenge to overcome (strategies are addressed in the next paragraph). Further advantages of UCNPs compared to organic dyes and QDs presents the high photostability and the absence of blinking, which are important features for single particle tracking applications.³⁶ Moreover, results of toxic studies of UCNPs indicate a rather low nanotoxicology.^{17, 37, 38} However, these risks have to be more intensely investigated to allow their use inside the human body.

Nowadays, the state-of-the-art synthetic protocols allow producing high-quality UCNPs with defined crystal morphology, narrow size distributions, controlled dopant concentrations and surface chemistries as well as different particle architecture. However, due to high surface-to-volume ratio of UCNPs their maximal Φ_{UC} is still one to two orders of magnitude smaller than for bulk. A very promising strategy to avoid surface-related luminescence quenching processes consists in the passivation of the surface with an undoped optically inactive shell. Recently, report showed that 23 nm-core particles βa NaYF₄:Yb³⁺(18%),Er³⁺(2%) covered with a 22 nm-thick undoped β-NaYF₄ shell reaching a similar Φ_{UC} compared to μ m-sized UC particles.^{39, 40} Nonetheless, the Φ_{UC} of UCNPs with diameters of less than 10 nm, desired for biomedical applications, was yet not fully restored with this core-shell approach.⁴¹ Additionally, the strong absorption of water molecules (H₂O) at 980 nm, resonant to the Yb3+ absorption, and thus, drastically enhancing luminescence quenching via the UCNPs surface, 42 is limiting the use of Yb3+-based UCNPs in aqueous media. In this respect, co-doping UCNPs with Nd³⁺, excitable at ca. 800 nm, offers an alternative way to avoid absorption of H₂O, and consequently, reduced heating and luminescence quenching effects. 43, 44 However, Nd3+-based UCNPs require more complicated synthetic routes and particle design as well as exhibiting a much narrower absorption band compared to conventional Yb3+-based UCNPs. Moreover, as mentioned previously, UCNPs have a rather weak photon absorption cross-section of Ln³⁺ ions, typically 5 - 8 orders of magnitude lower than for organic dyes and QDs. Most promising strategies to enhance the absorption of UCNPs comprise a) enhancing local electromagnetic field via plasmonics effects based on metal nanostructures or photonic crystals and b) sensitization of UCNPs with organic dyes or QDs. UC luminescence enhancement in the order of three orders of magnitude have already been reported with these strategies. 45-49

The further rational design of brightly-luminescent UCNPs with industry-relevant structural properties and functionalities requires understanding of (de-)population processes and deactivation channels. This can be assessed with upconversion quantum yield (Φ_{UC})

measurements, particle absorption and decay behavior combined with theoretical investigations. In particularly, the measurement of Φ_{UC} , equaling the ratio of high energy photons emitted to low energy photons absorbed, is known to be very challenging. Due to its strong dependency on the P, absolute Φ_{UC} values can be exclusively determined reliably with a customized integrating sphere setup (ISS). This initiated an increasing number of reports on absolute Φ_{UC} measurements utilizing an ISS. $^{50-55}$ Nonetheless, the comparison of the Φ_{UC} obtained of different laboratories remains a challenging task since most reports are missing a comprehensive description of the instrument design, calibration and characterization as well as measurement procedure. Furthermore, the evaluation of reported Φ_{UC} values complicates due to differences in the excitation wavelength, missing a characterization of the excitation beam profile and the use of an insufficiently low P-range of less than two orders of magnitude. Thus, there is an urgent need for the development of standardized absolute Φ_{UC} measurements with an ISS.

1.1 Aims and outline of this cumulative thesis

The main aims of this cumulative thesis:

- (i) Designing, building-up and characterizing an ISS for absolute UC quantum yield measurements for UC materials excitable at 980 nm.
- (ii) Identifying UC population and depopulation processes for β -NaYF₄:Yb³⁺,Er³⁺ UCNPs dispersed in H₂O.
- (iii) Understanding the influence of Yb³⁺ and Er³⁺ dopant concentration on emission color of β -NaYF₄:Yb³⁺,Er³⁺ UCNPs.

In contrast to the three publications this cumulative thesis is based on, a more detailed description of the custom-built ISS is provided (see Chapter 4). Furthermore, for each publication an abbreviated version has been written (see Chapter 5,6 and 7). These parts may serve as introduction for a broader readership and describe some aspects in more detail. In the following, a brief outline of this thesis is provided:

1.1 Aims and outline of this cumulative thesis

- Chapter 2 provides fundamentals of the UC processes. In particular, the UC processes of the investigated β-NaYF₄:Yb³⁺,Er³⁺ UC system are described in detail.
- Chapter 3 presents an overview of the state of the art for each publication and introduces questions addressed by this thesis.
- Chapter 4 describes the new custom-built ISS and its characterization as well as the analysis of the obtained data. An overview of the investigated UC samples is also provided.
- Chapter 5 provides prerequisites for the measurement procedure and geometry and for the optical properties of the UC material as dispersion or powder. Furthermore, the results of Φ_{UC} measurement for a μ m-sized UC material were validated by comparison with literature data.
- Chapter 6 presents studies of UCNPs in different solvents with a focus on the optical properties of UCNPs in H_2O and D_2O . A model for the (de-)population pathways of UCNPs in H_2O was developed.
- Chapter 7 deals with the influence of the Yb³⁺ and Er³⁺ dopant concentrations on the Er³⁺ green and red emission intensities of UCNPs. The interpretations were supported with a comprehensive rate equation analysis.

2 Fundamentals

This chapter introduces the physical basics for the understanding of different UC processes. Thereafter, conditions for the composition of an efficient Ln^{3+} -based UC material including luminescent centers and host lattices are discussed. Subsequently, a focused overview of the properties of β -NaYF₄:Yb³⁺,Er³⁺ is given. This material is known to be the most efficient NIR to green light converter based on energy transfer upconversion (ETU), and hence, was comprehensively studied in this thesis. At the end of this chapter, a brief theoretical overview of the key parameter Φ_{UC} is provided.

2.1 Upconversion (UC) processes

PL is a process in which a material absorbs incident photons, exciting electrons to a higher electronic excited state, and then emits photons after the relaxation of the electrons to a lower energy state. In case of a linear absorption-emission PL process, the energy of the emitted photons do not exceed the energy of the absorbed ones (Stokes shifted emission). However, a UC process permits the combination of multiple absorbed photons resulting in the emission of higher energy photon (anti-Stokes shifted emission). The era of the UC field started in 1959, when Bloembergen proposed the idea of an infrared(IR)-detector based on the sequential absorption of two NIR photons within energy levels of an ion in a solid, called excited state absorption (ESA). Unfortunately, the lack of intense coherent laser sources at this time made it impossible to observe any effects. In 1966, Auzel extended the approach of Bloembergen by utilizing energy transfer between different ions in a solid matrix, called ETU, which could even be detected with incoherent light sources. Nowadays, ETU-based upconverters still belong to the most efficient UC materials.

Figure 2.1 shows a collection of different UC processes for the special case that exactly two absorbed photons are summed-up resulting into the emission of a higher energy photon.³⁰ ESA is the simplest UC mechanism and involves a sequential absorption of two photons *via* a real intermediate electronic energy state of a Ln³⁺-ion, see Figure 2.1 a). However, this process is not very efficient, because of the rather low absorption coefficients of suitable Ln³⁺ ions. Compared to ESA, ETU is about one hundred times more efficient.³⁰ ETU based upconverters are typically materials doped with two different Ln³⁺ called sensitizer and activator. The sensitizer ions harvest the energy, i.e. they effectively absorb incoming photons, and transfer it in sequential steps *via* nonradiative ET to the activator ion, see Figure 2.1 b). The efficiency of ETU is commonly boosted by high sensitizer concentrations allowing

fast energy migration between sensitizer ions inside the crystal lattice.⁵⁹ In some special cases, chemically identical ions can act simultaneously as activator and sensitizer ion, e.g. for singly Er³⁺-doped 1520 nm excitable upconverters used for solar cell applications.⁶⁰ The efficiency of ESA and ETU processes both rely on metastable energy levels required for the sequential absorption of photons to avoid early depopulation. This is satisfyingly fulfilled by most Ln³⁺ ions, due to their parity-forbidden f-f transitions and manifold of energy levels (see 2.3.1 Luminescent centers).

Figure 2.1 c) - e) present UC processes based on partly missing or even without any energy level called cooperative energy transfer (CET), two photon absorption (TPA) and second harmonic generation (SHG). CET is similar to ETU, but misses a real intermediate energy level for the activator ion. At the CET process, two sensitizer ions are simultaneously transferring the energy *via* a virtual energy state to the emitting activator ion. Studies on CET-based UC materials primarily focused on bulk and glasses due to their low UC efficiencies. TPA and SHG typically occur in non-lanthanide materials and have to be pumped with rather expensive lasers sources with pulse lengths in the pico- to femtosecond time scale as they require very high *P*. TPA, which is based on the simultaneous absorption of two photons without a real intermediate energy state, of fluorescent dyes molecules or QDs is frequently used in high-resolution microscopy, bioassays, and even skin cancer detection. G4-66 SHG is a UC process even without occupying any real energy level utilizing the nonlinear susceptibility of a medium. SHG is amongst other applications a popular choice for the frequency doubling of lasers with maximal energy conversion efficiency of up to 70% for optimal conditions. The process of the samples of the process of the frequency doubling of lasers with maximal energy conversion efficiency of up to 70% for optimal conditions.

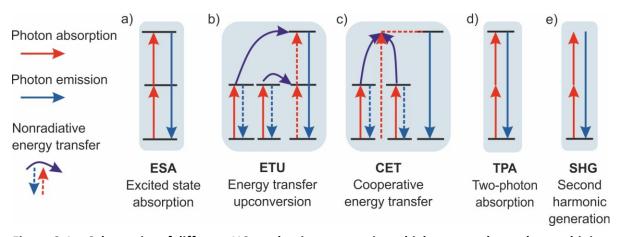


Figure 2.1 Schematics of different UC mechanism generating a high energy photon by combining two low energy photons: a) ESA; b) ETU); c) CET; d) TPA; e) SHG.

At last, a brief description of the triplet-triplet annihilation (TTA) UC process is provided. TTA-based systems have sizes of only few nanometers due to its molecular structure, and thus, are very promising candidates for biomedical applications. At the TTA process the energy of two triplet-excited dye molecules (annihilators) is combined by ET from one molecule to the other molecule, followed by the emission of a high energy photon. To enhance the absorptivity TTA molecules are typically bound to strong absorbing dye molecules (sensitizers). Thereby, the sensitizers effectively transfer the absorbed energy to the annihilators after an intersystem crossing, i.e. a sensitizer changes from an excited singlet state to a lower energetic excited triplet state. The Φ_{UC} values of such TTA-systems found to be in the order of one percentage at rather low P of several hundred mWcm⁻². Ref. The limiting factor of TTA-systems is the high sensitivity to oxygen of the involved triplet states reducing the Φ_{UC} .

2.2 Energy transfer processes between Ln³⁺ ions

In this subsection, a theoretical overview of energy transfer (ET) processes between Ln³+ is provided. Energy transfer is present when absorption and emission of a photon do not take place at the same luminescent center.³0 ET processes can be divided into radiative and nonradiative ET processes. Radiative ET usually plays a minor role for UC processes due to the small absorption coefficients of Ln³+ ions. In contrast, nonradiative ET is a key feature for ETU leading to effective migration and diffusion processes in the UC crystal lattice. Therefore in this thesis, the term ET implies nonradiative ET. Figure 2.2 depicts radiative and nonradiative ET processes for the simple case of only two ions interacting with each other. These two ions are named sensitizer and activator, where the sensitizer transfers the absorbed energy to the activator. Although donor and acceptor would be clearer terms, sensitizer and activator are intentionally used by the UC community to avoid misunderstanding with the nomenclature used for semiconductors.

Radiative ET takes place *via* an exchange of a real photon. The probability W_{SA} to transfer energy from the sensitizer to the activator for a radiative ET between two ions can be calculated in dependence of the distance R with Equation 2.1.⁷³

$$W_{\rm SA}(R) = \frac{1}{\tau_{\rm S}} \frac{\sigma_{\rm A}}{4\pi R^2} \int g_{\rm S}(\nu) g_{\rm A}(\nu) d\nu$$
 Equation 2.1

Here, τ_s is the PL lifetime of the sensitizer, σ_A the integrated absorption cross section of the activator, and the integral represents the spectral overlap between the sensitizer emission $g_s(v)$ and activator absorption $g_A(v)$. Radiative ET does not affect the PL lifetime of the donor, so that it can be experimentally distinguished from nonradiative ET processes. For more complex systems the probability of a radiative ET $W_{SA}(R)$ depends not solely on R, but also on the shape and scattering properties of the sample.

Nonradiative ET becomes relevant for short distances in the nanometer range requiring high dopant concentrations of Ln³⁺. In this case the excitation energy is transferred from the sensitizer ion to the activator without the exchange of a real photon, see Figure 2.2 b). Such nonradiative ET *via* dipole-dipole interaction was firstly described by Förster in 1946.⁷⁴ Later this theory was extended by Dexter for electric multipolar interactions leading to Equation 2.2 providing the probability for this nonradiative ET process.⁷⁵

$$W_{\rm SA}(R) = \frac{1}{\tau_c} (R_0/R)^k$$
 Equation 2.2

Here, R_0 is called the Förster radius, the distance of the two ions at which the probability of an ET equals 50%, and k is a positive integer with k=6 for dipole-dipole (Förster case), k=8 for dipole-quadrupole and k=10 for quadrupole-quadrupole interactions. The Förster radius depends on the overlap of the sensitizer emission and activator absorption, the integrated absorption cross-section of the activator ion, and the photon energy. However, for very short distances less than 1 nm the orbital overlap or Dexter exchange may become the dominant nonradiative ET process. At this process the excited electrons are "hopping" through the network. The probability W_{ET} for this process has an exponential relationship to R. The probability W_{ET} for this process has an exponential relationship to R.

$$W_{\rm ET}(R) = K^2 e^{-2R/L}$$
 Equation 2.3

Here, K is a scaling parameter with a dimension of energy and the parameter L is named "effective average Bohr radius", which accounts for the distance decay of the interaction.

A phonon-assisted nonradiative ET can take place between different ions with an energy mismatch between the energy levels of the sensitizer and activator, see Figure 2.2 c). This energy mismatch can be overpassed via production or annihilation of phonons. Small energy mismatches < 100 cm⁻¹ can be bridged by thermal phonons, and larger energy mismatches, even as high as several thousand cm⁻¹ can be bridged by optical phonons of the host lattice. In accordance to the Miyakawa-Dexter theory the probability for a phonon assisted ET $W_{PAT}(\Delta E)$ can be described with Equation 2.4.^{77, 78}

$$W_{\rm PAT}(\Delta E) = W_{\rm PAT}(0){\rm e}^{-\beta \Delta E}$$
 Equation 2.4

Here ΔE is the energy difference between the sensitizer and the activator level, $W_{PAT}(0)$ is the energy transfer probability for $\Delta E = 0$, β is a parameter characterized by the strength of electron-lattice coupling and the nature of the phonon involved.

Cross-relaxation (CR) is a special case of nonradiative ET and refers to all types of ETs between identical ions, where one ion transfers a portion of energy to another, see Figure 2.2 d). CR becomes usually a dominant process for high dopant concentration of the activator ions leading the undesired depopulation of the higher energy levels. This is also called concentration quenching.

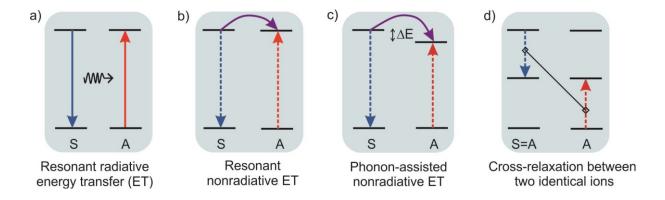


Figure 2.2 Schematics of radiative and nonradiative ET processes between two ions:

a) Resonant radiative ET; b) Resonant nonradiative ET; c) Phonon-assisted nonradiative ET; d) Cross-relaxation. Based on a figure from Auzel.³⁰

2.3 Energy transfer upconversion (ETU)-based materials

UC materials, based on ETU, consist of three components: i) host lattice, ii) sensitizer ions, and iii) activator ions. The design of efficient UC materials requires careful selection of these components.

2.3.1 Luminescent centers - sensitizer and activator ions

Ln³+ ions have excellent properties for the design of efficient UC materials due to their multiple long-lived optical f-f transitions (with the exception of Yb³+ and Ce³+ with only one optical transition). Figure 2.3 shows the energy level scheme for free Ln³+ ions with their 4f-4f transitions. For free Ln³+ ions the optical transitions between the 4f-4f energy levels are parity forbidden, i.e. that the probability for the emission of a photon is zero.⁷⁹. This rule is broken for Ln³+ ions embedded in a host lattice. Due to disturbances of the wavefunctions by the strength and symmetry of the crystal field of the host lattice, these parity forbidden transitions become weakly allowed. Because these 4f-4f optical transitions are shielded by the outer 5s²5p6 subshells,⁸⁰ these effects are small. This implies that the optical transition of the Ln³+ ions are characterized by long PL lifetimes in the microsecond to millisecond range and narrow linewidths (full width at half maximum (FWHM) typically between 10-20 nm) in the solid state.

In the case of ETU-based UC materials, the standard ingredients are Yb³⁺ as sensitizer and Er³⁺, Tm³⁺ or Ho³⁺ as activator ions. The reason for Yb³⁺ being the best choice as sensitizer is based on three facts: i) Only one excited state ($^2F_{5/2}$), ii) the high absorption cross section compared to other lanthanide ions ($1.2 \cdot 10^{-20}$ cm² ⁸¹), and iii) the $^2F_{5/2}$ energy level of Yb³⁺ (980 nm) matches energetically well with energy levels of the typical activator ions. For example the Er³⁺ 980 nm $^4I_{11/2}$ energy level is resonant to this Yb³⁺ energy level allowing efficient Yb³⁺-Er³⁺ ET. In order to increase the absorption and energy migration usually high dopant concentrations of Yb³⁺ in the range of 20% up to 98%. Ero activator ions a ladder like energy level structure with equally-spaced intermediate energy levels is required. This is fulfilled by Er³⁺, Tm³⁺ and Ho³⁺, see Figure 2.3. The dopant concentration of the activator ion is commonly in the range of 1% to avoid concentration quenching induced by cross relaxation. However, recent studies indicate that for very high *P* in the region of 1 MWcm⁻² much higher activator concentrations of up to 20% for Er³⁺ are advantageous. Sa

The energy levels of Ln^{3+} are denoted in the Russell-Saunders notation using $4f^N$ states (N = 1 to 13) of $^{2S+1}L_J$ multiplets, with S, L and J representing the total spin, orbital, and angular momenta of the N 4f electrons.⁸⁴ The splitting of the energy levels by electrostatic

interaction ^{2S+1}L is in the order of 10^4 cm⁻¹. 85 In addition, the spin-orbit interaction leads to further splitting of the levels into $^{2S+1}L_J$ with a separation of the J states by 10^3 cm⁻¹. Moreover, the crystal field of the lattice leads to an additional separation of the energy states in the order of 10^2 cm⁻¹. 85

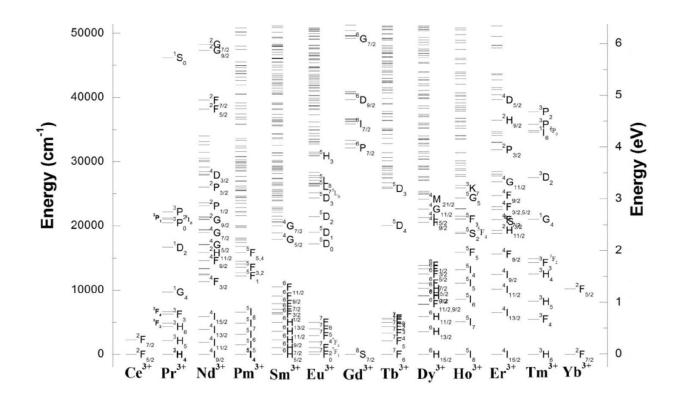


Figure 2.3 Energy level diagram for the 4f levels of Ln³+ ions. Positions of energy levels are calculated by using free ion parameters described by Walrand et al.⁸⁶ Reprinted from Mahata et al.⁸⁰ with permission from Intech.

2.3.2 Host lattice

The choice of the host lattice is essential for an efficient UC emitter. An ideal host material should meet the following four criteria:

- i) large band gap > 6 eV to be transparent for excitation and UC emission light
- ii) high tolerance for dopant concentrations of Ln³⁺ to avoid lattice defects
- iii) low phonon energies of the host lattice to minimize nonradiative deactivation
- iv) chemically and thermally inert

These criteria are met by inorganic host materials based on certain rare earth ions (Re^{3+}): Sc^{3+} , Y^{3+} , Gd^{3+} , La^{3+} or Lu^{3+} . These materials allow high dopant concentrations of the luminescent centers that substitute these Re^{3+} ions in the host lattice. Moreover the luminescent centers have similar ionic radii like all Re^{3+} ions (Ln^{3+} ions are a subgroup of the group of Re^{3+} ions). The most common host matrices used for UC materials are oxides (e.g. Y_2O_3), Y_2O_3 , Y_3O_3 , Y

Among these materials, the fluoride-based NaReF₄ have been used for decades as nm-sized host lattices due to their low phonon energy (quantized energy portions of lattice vibrations), excellent chemical stability, and thermal stability. ⁹³⁻⁹⁵ NaReF₄ has two different crystal phase structures, the cubic (α –NaReF₄) and the hexagonal phase structure (β –NaReF₄), see Figure 2.4.

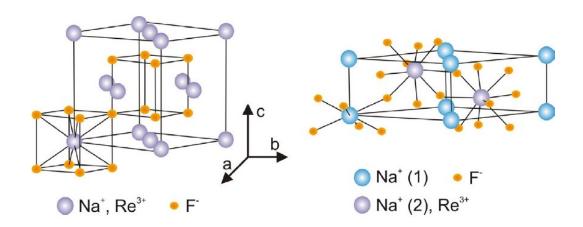


Figure 2.4 Schematic presentation of the crystal structure of NaReF₄ in the cubic and hexagonal- phase. a) The cubic phase is characterized by an equal number of F⁻ cubes containing cations and vacancies; b) the hexagonal phase shows an ordered array of F⁻ ions with two kind of cation sites, where one is occupied by Na⁺ and the other one is randomly occupied by either Na⁺ and Re³⁺. Reprinted from Wang et al. 96 with permission from Nature Publishing Group.

It is well established that the family of β –NaReF₄ is much more efficient as host lattice for UC materials than α –NaReF₄, due to their lower phonon energies of about 350 cm⁻¹ and a lower symmetry of the crystal field.^{97, 98} These lower phonon energies result in a smaller probability of phonon-based nonradiative deactivation. The probability of such a

deactivation process, also called multiphonon relaxation (MPR), depends exponentially on the number of phonons with an energy $E_{\rm phonon}$ needed to bridge the energy gap ΔE between the final and the initial energy level and can be calculated with Equation 2.5.^{99, 100}

$$W_{\rm MPR}(p) = \beta e^{-\alpha p}$$
 Equation 2.5

Here, α and β depend on the specific properties of the host lattice and $p = \Delta E/E_{phonon}$ equals the number of phonons required for the nonradiative deactivation. MPR is the dominant nonradiative process for perfectly crystalline materials. However, for nm-sized particles the nonradiative deactivation is mostly governed by lattice defects or molecules near the surface or inside the particle volume.^{59, 101} In particular, this is critical for ET-based UC systems, where the ET can be effectively transferred *via* energy migration to quenching centers. A recent study showed that surface-related quenching and the number of defect centers can be strongly minimized with a thick inactive shell and slow particle growth rates.³⁹

2.4 Upconversion luminescence (UCL) of β-NaYF₄:Yb³⁺,Er³⁺ crystals

The UC system β -NaYF₄:Yb³⁺,Er³⁺ combines the previously discussed criteria for host lattice and luminescent centers for an efficient ETU-based UC material. This material is intensively studied since the mid-1960s and well known for its high UC efficiency. Typical dopant concentrations range from 17-20% for the Yb³⁺-ion and 1-2% for the Er³⁺-ion dopant concentrations aiming on an intense Er³⁺ green UCL.^{21, 97} ESA processes involving the Er³⁺ energy levels can be neglected, due to the 10-fold higher absorbance of Yb³⁺ compared to Er³⁺. Figure 2.5 shows a typical UCL spectrum of β -NaYF₄:Yb³⁺,Er³⁺, which is dominated by the Er³⁺ emission bands ranging from 370-870 nm. The strongest UC emission bands are centered at 380 nm, 410 nm, 520 nm and 540 nm, 655 nm, 810 nm, and 850 nm. All Er³⁺ emission bands are characterized by a substructure of several sharp emission peaks, caused by a splitting from the crystal field of the host lattice. The associated emissive Er³⁺ energy levels can be populated *via* two or more energy transfers from Yb³⁺ to Er³⁺ and in combination with internal nonradiative relaxation processes, see Figure 2.6.

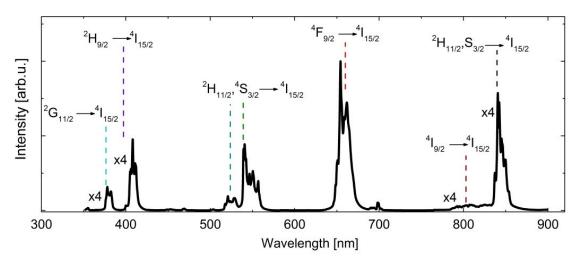


Figure 2.5 UCL emission spectrum of μm-sized β-NaYF₄:Yb³⁺(21%),Er³⁺(2%) UC particles (UCμP_{BAM}) excited at 976.4 nm at a P of 20 Wcm⁻². The typical Er³⁺ emission bands are centered at 380 nm (4 G_{11/2} \rightarrow 4 I_{15/2}), 410 nm (2 H_{9/2} \rightarrow 4 I_{15/2}), 520 nm/540 nm (2 H_{11/2}, 4 S_{3/2} \rightarrow 4 I_{15/2}), 655 nm (4 F_{9/2} \rightarrow 4 I_{15/2}), 810 nm(4 I_{9/2} \rightarrow 4 I_{15/2}), and 850 nm (2 H_{11/2}, 4 S_{3/2} \rightarrow 4 I_{15/2}). For a better visualization the UV and NIR bands were multiplied by a factor of 4. The Er³⁺ green and red emission bands account for > 80% of the overall UCL intensity.

The Er³+ green emission bands (${}^2H_{11/2}$, ${}^4S_{3/2} \rightarrow {}^4I_{15/2}$) centered at 520 nm and 540 nm and Er³+ NIR 850 nm emission band (${}^2H_{11/2}$, ${}^4S_{3/2} \rightarrow {}^4I_{13/2}$) centered at 850 nm are activated *via* two Yb³+-Er³+ ETs. The first ET populates the Er³+ ${}^4I_{11/2}$ energy level, followed by a second ET populating the Er³+ ${}^4F_{7/2}$ energy level. Due to the high nonradiative rate for Er³+ ${}^4F_{7/2}$ energy level, the excited electron relaxes mainly to the directly lower lying ${}^2H_{11/2}$, ${}^4S_{3/2}$ energy levels. Subsequently, an optical transition can take place either to the ground state ${}^4I_{15/2}$ or the first excited state ${}^4I_{13/2}$ of Er³+ resulting in the green 520/540 nm or NIR 850 nm emission, respectively.

The Er³⁺ NIR 810 nm emission band (${}^4I_{9/2} \rightarrow {}^4I_{15/2}$) centered at 810 nm can be populated *via* two different biphotonic pathways. The first population pathway is *via* CR from the biphotonically activated ${}^2H_{11/2}$, ${}^4S_{3/2}$ to ${}^4I_{9/2}$ and the second *via* Er³⁺-Er³⁺ ET in case of a high population of the Er³⁺ ${}^4I_{13/2}$ energy level (${}^4I_{13/2} + {}^4I_{13/2} \rightarrow {}^4I_{9/2}$).

The Er³+ ultraviolet (UV) emission band (${}^4G_{11/2} \rightarrow {}^4I_{15/2}$) centered at 380 nm and the Er³+ purple emission band (${}^2H_{9/2} \rightarrow {}^4I_{15/2}$) centered at 410 nm are populated by a triphotonic UC process, i.e. that three sequential Yb³+-Er³+ ETs are involved. The first two ET steps are identical as described above for the biphotonically activated green-emitting ${}^2H_{11/2}$, ${}^4S_{3/2}$ energy levels. The third ET excites the electron to the ${}^2G_{7/2}$ energy level, followed by nonradiative relaxation to the ${}^4G_{11/2}$ and ${}^2H_{9/2}$ energy levels. The optical transition to the ground state ${}^4I_{15/2}$ then results in the UV or purple emission, respectively.

The Er³⁺ red emission band (${}^4F_{9/2} \rightarrow {}^4I_{15/2}$) centered at 655 nm has the most complex nature of all emission bands as it can be activated via different biphotonic and triphotonic pathways. A comprehensive overview of the conditions for bi- or triphotonic activation, which was intensively debated within the last years, is given in the introduction of Publication III. In the following a brief version of this overview is presented. The photonic nature of the Er³⁺ red emission band depends on the crystal phase, size, particle architecture, environment and excitation power density (P). 23, 26, 29, 40, 42, 55, 96, 102-113 The triphotonic population pathway suggested from the Berry group in 2014¹⁰² includes an Er³⁺-Yb³⁺ back energy transfer (BET) from the triphotonically activated UV emissive ⁴G_{11/2} energy state to the red emissive ${}^4F_{9/2}$ energy state. The biphotonic activation of the Er $^{3+}$ ${}^4F_{9/2}$ energy level can occur via two different population routes. The first biphotonic population pathway directly feeds the ⁴F_{9/2} energy level via nonradiative relaxation from the biphotonically activated ⁴S_{3/2} energy levels. Therefore, an energy gap in the order of 3000 cm⁻¹ to 3200 cm⁻² ¹ has to be overpassed. In the case of UCNPs capped with oleic acid molecules the stretching vibrational modes of the CH₂-groups (2800 cm⁻¹ to 2950 cm⁻¹) are perfectly capable to bridge this energy gap. 114 The second biphotonic pathway occurs via the 1520 nm 41_{13/2} energy level of Er³⁺, which can be populated by nonradiative relaxation from one ET activated Er³⁺ 980 nm ⁴I_{11/2} energy level. Due to an energy mismatch of 1000 cm⁻¹ to the Er³⁺ red emissive ⁴F_{9/2} , this ET needs assistance by phonons and requires population of the ⁴I_{13/2} energy level.

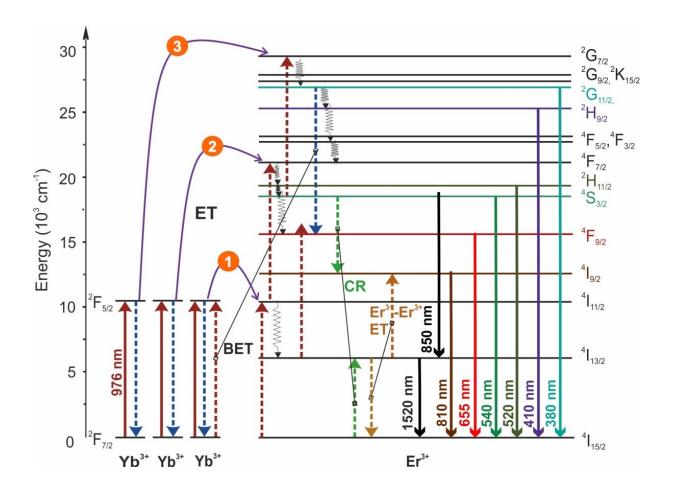


Figure 2.6 Term scheme illustrating the relevant population pathways for the different Er³+ emission bands for Yb³, Er³+ based UC materials. Solid arrows represent excitation or de-excitation of an energy state by absorption or emission of a photon; the dashed arrows represents ET processes including Yb³+,Er³+ ET, Er³+-Yb³+ BET, CR between two Er³+ ions and Er³+-Er³+ ET; curved arrows presents nonradiative relaxation processes by MPR. It should be noted that for simplification the splitting of energy levels by the crystal field of the host lattice is not included in the term scheme.

2.5 Upconversion photoluminescence quantum yield (Φ_{UC})

The Φ (also often referred as "internal quantum yield"), presents the ratio of the number of emitted (N_{em}) to the number of absorbed absorbed photons (N_{abs}), see Equation 2.6.^{11, 21, 115, 116} This key parameter characterizes the photon conversion performance of photoluminescent materials.

$$\Phi = N_{\rm em}/N_{\rm abs}$$
 Equation 2.6

In the case of the Φ_{UC} , only emitted photons with a higher energy than the energy of the absorbed ones are considered ($E_{em} > E_{abs}$), see Equation 2.7.^{11, 53, 102, 115, 117}

$$\Phi_{\rm UC} = N_{\rm em}/N_{\rm abs}$$
, for $E_{\rm em} > E_{\rm abs}$ Equation 2.7

As the UC process is a nonlinear process, a higher excitation power density (P) generally leads to an increase of Φ_{UC} . In the case of a biphotonic UC process, the UCL intensity (I_{UC}) is proportional to P^2 , and consequently, Φ_{UC} increases linearly with P. However, due to the energy conservation law, the P-dependent increase of Φ_{UC} has to level off at higher P, and thus Φ_{UC} is limited by 100%/n, where n is the photonic order of the UC process. The saturation behavior of the P-dependent Φ_{UC} ($\Phi_{UC}(P)$) for an ideal biphotonic emitter was studied by the Andersson-Engels group. They assumed a two-level emitter system and subsequently derived a formula for $\Phi_{UC}(P)$ for this simple UC system, see Figure 2.7 a) and Equation 2.8. The saturation is a process of the P-dependent and P-dependent P-dependent and P-dep

$$\phi_{\rm UC}(P) = \phi_{\rm UC,sat}/(\frac{P_{\rm balance}}{P} + 1)$$
 Equation 2.8

Here, $\Phi_{UC,sat}$ is the saturated upconversion quantum yield, where $\Phi_{UC}(P)$ converges at high P, and $P_{balance}$ is the balancing excitation power density, where $\Phi_{UC}(P_{balance})$ equals $\Phi_{UC,sat}/2$.

Figure 2.7 b) shows the typical shape of the Φ_{UC} curve for such an ideal biphotonic emitter, calculated with Equation 2.8. This formula is only valid for purely biphotonic UC processes like the NIR 800 nm emission band of Tm^{3+} , Yb^{3+} and the green 520/540 nm of Yb^{3+} , Er^{3+} based UC systems. ^{40, 119, 120} In contrast, this formula is not suitable to describe the shape of $\Phi_{UC}(P)$ for the red emission of Yb^{3+} , Er^{3+} or Yb^{3+} , Er^{3+} as here the population processes are more complicated. ^{40, 55, 102, 121}

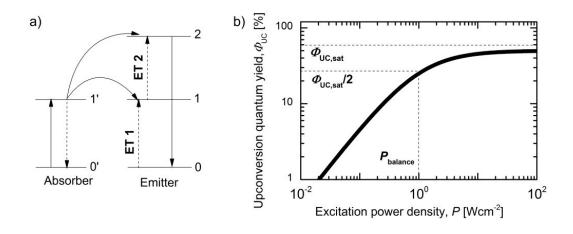


Figure 2.7 *P*-dependent Φ_{UC} behavior of a two-level emitter ETU system: a) Term scheme of the two-level emitter system; b) $\Phi_{UC}(P)$ curve calculated for $\Phi_{UC,sat} = 50\%$ and $P_{balance} = 1 \text{ Wcm}^{-2}$ with Equation 2.8.¹¹⁹

The simulation of $\Phi_{UC}(P)$ for more complex UC systems can be achieved by using a rate equation system consisting of a set of coupled differential equations, see Equation 2.9.¹²²

$$\frac{dN_i}{dt} = \sum$$
 populating rates $-\sum$ depopulating rates Equation 2.9

Thereby, the population densities (N_i) of the most significant 4f energy levels of the Ln³⁺ ions and their interactions have to be considered. The interactions are represented by populating and depopulating rates of the energy states including radiative rates and nonradiative rates like ET, CR, and MPR.A very decent approach was demonstrated by the Berry group for β -NaYF₄:Yb³⁺,Er³⁺ μ m-sized and nm-sized UC systems with a rate equation system including a set of nine energy levels for Er³⁺. ¹⁰², ¹²³ This rate equation system was later also used in Publication II and III to draw and underpin interpretations of the influence on UC mechanisms for β -NaYF₄:Yb³⁺,Er³⁺ UCNPs by modeling the P-dependent intensities of the Er³⁺ emission bands.

3 Reviews of Literature and Open Questions

In this chapter a review of the literature for each publication is presented. Furthermore, open questions are formulated, which are addressed in the chapters 4, 5, 6 and 7.

3.1. Absolute upconversion photoluminescence quantum yield (Φ_{UC})

Recently, absolute Φ_{UC} measurements have raised great interest in the UC community as a quantitative comparison tool to identify strategies to improve the brightness of UCNPs. 124 Attempts to measure the Φ_{UC} relative against a Φ_{UC} standard or to determine enhancement factors with a spectrofluorometer¹²⁵⁻¹²⁷ have shown limited reliability. This is related to the fact that there is no reliable $\Phi_{\sf UC}$ standard available, and additionally, due to often different scattering properties and different P-dependencies of $\Phi_{
m UC}$ for UC materials. 40 Therefore, the Φ_{UC} can be only measured absolutely with a spectrally calibrated ISS done.⁵¹, $^{52,\,54,\,55,\,128\cdot132}$ Although, the increasing number of reports on ISS the comparability of the $arPhi_{
m UC}$ results obtained with these setups is still limited by the often missing description of the instrument design and instrument characterization. For example, there is no report how the influence of the excitation beam profile (BP), i.e. the spatial distribution of P, is considered. Additionally, often reports provide only a single Φ_{UC} value or the Φ_{UC} has been measured only in a small P-range of less than two orders of magnitude, which is insufficient for monitoring saturation dynamics of the UC process. Up to the publication date of Publication I (June 2017) the highest measured Φ_{UC} value for Yb³⁺,Er³⁺-based systems was 7.8 % for a β-NaYF₄:Yb³⁺(20%),Er³⁺(2%) μm-sized UC powder at *P* of 22 Wcm⁻².⁵⁴

Open questions

- What are the prerequisites for the design and characterization of an ISS for reliable Φ_{UC} measurements? (see 4.2.1 Integrating sphere)
- What are the prerequisites on the optical properties of the sample, measurement procedure, and measurement geometry to obtain Φ_{UC} values with minimum uncertainty? (see 5.1. Measurement conditions for $\Phi_{UC}(P)$)

3.2. Solvent-dependent luminescence quenching

UCNPs are generally more prone to surface quenching due to their much higher surfaceto-volume ratio and shorter distance between particle center and surface compared to larger-sized particles. Thus, Ln³⁺ ions of UCNPs can transfer energy effectively to the surface region, where the excited ions can be deactivated nonradiatively by impurities or crystal defects on the surface, as well as vibrational modes of O-H and C-H bands of solvent and ligand molecules. 110, 112, 113, 133-136 Particularly, the interest in UCNPs for aqueous environments for life sciences applications has triggered an increasing number of reports studying solvent quenching mechanism in water (H₂O). 42, 107, 133, 137-139 Arppe et al. showed in a joint publication in 2015 that Yb³⁺ ions play a major role for the migration of energy to the Therefore, studied bare surface. they rod-shaped ca. 35 nm-sized β-NaYF₄:Yb³⁺(17%),Er³⁺(2%) UCNPs dispersed in H₂O and deuterated water (D₂O), showing a decrease of the Er³⁺ green ($^2H_{11/2}$, $^4S_{3/2} \rightarrow ^4I_{15/2}$) and Er³⁺ red ($^4F_{9/2} \rightarrow ^4I_{15/2}$) UCL intensity of 99.9% for UCNPs in H₂O compared to D₂O for the UCNPs in H₂O.¹⁴⁰ Wilhelm et al. reported in a joint publication green-to-red ratios (Igreen/Ired) of NaYF4:Yb3+(17%),Er3+(2%) 23 nm-sized UCNPs in dependence of different solvents and different surface ligands. 138 This report did not include a model for the change of population and depopulation processes in H₂O.

Open questions

- How to measure Φ_{UC} in strongly absorbent solvents like H₂O? (see 5.1. Measurement conditions for $\Phi_{UC}(P)$, Heating of the UCNP sample and solvent)
- What is the influence on the Φ_{UC} of surface protected UCNPs dispersed in different solvents? (see 6.1 Solvent-dependent quantum yield (Φ_{UC}))
 - How are the UCL bands and de- and population processes of these UCNPs affected by different solvents? (see 6.2 Emission color of UCNPs dispersed in D_2O and H_2O)

3.3. Dopant concentration-dependent color tuning

The dopant ion concentration of Yb³+,Er³+-based UCNPs has direct influence on their optical properties like Φ_{UC} and emission color. An increase of the Yb³+ concentration leads to an increase of the intensity of the Er³+ red emission band ($^4F_{9/2} \rightarrow ^4I_{15/2}$) at the expenses of the Er³+ green emission band ($^2H_{11/2}$, $^4S_{3/2} \rightarrow ^4I_{15/2}$) as frequently reported.^{82, 104-106, 141-143} However, different explanations are given for this observation. For the variation of Er³+ concentration, ambiguous or even opposite trends of the Er³+ green-to-red intensity ratio (I_{green}/I_{red}) have been observed.^{141, 142, 144-146}. Additionally, the evaluation of these data is difficult as the dopant concentration-dependent I_{green}/I_{red} is also affected by P, and most publications provide the I_{green}/I_{red} only for single P values. To clarify the mechanisms influenced by the dopant concentration, a comprehensive rate equation analysis using P-dependent spectral intensities for a broad P-range and lifetime data of the different emission bands for P-NaYF4:Yb³+,Er³+ is necessary. This has not been reported for varying the Yb³+ and Er³+ concentration.

Open questions

- How do the Yb³⁺ and the Er³⁺ dopant concentration influence the *P*-dependent UCL? (see 7.1 Dopant concentration-dependent upconversion luminescence (UCL))
- How do the rate constants change by variations of the Yb³⁺ and the Er³⁺ dopant concentration? (see 7.2 Influence of the dopant concentrations on rate equation constants)

4. Samples and Experimental Methods

This chapter provides an overview of the investigated UC samples and a description of the custom-built ISS with its optical and opto-electronic components and ratiometric characterization, as well as the analysis of the data. At the end of the chapter, the commercial setup for lifetime measurements is described.

4.1. Overview of investigated β-NaYF₄:Yb³⁺,Er³⁺ samples

In Table 4.1 all investigated β -NaYF₄:Yb³⁺,Er³⁺ UC particles are listed. For Publication I commercial 3 μ m-sized β -NaYF₄:Yb³⁺(21%),Er³⁺(2%) UC particles (PTIR 550) obtained from *Phosphor Technology Ltd.* were studied. ¹⁴⁷ The UCNPs were synthesized by project partners using different chemicals for the building blocks of β -NaYF₄:Yb³⁺,Er³⁺ crystals. These reagents were mixed and heated up to initialize formation of the UCNPs. To produce high quality UCPNs with pure phase structure well-developed protocols with exact timing for each heating step, high purity reagents, and optical growth control are required. ¹³⁸

UCNPs samples synthesized by Emilia Palo from the Prof. Soukka group, University of Turku, Finland

Emilia Palo from the group of Tero Soukka from Finland synthesized the UCNPs, for Publication I and Publication III, using a protocol from Ylihärsilä et. al.¹⁴⁸ with small modifications (see Table 4.1). For Publication I, oleic acid (OA)-capped 25 nm-sized β-NaYF₄: Yb³⁺(17%) ,Er³⁺(3%) UCNPs were synthesized. For Publication III, two dopant concentration series with different Yb³⁺ and Er³⁺ concentrations were synthesized (see Table 4.1). The structural analysis was mainly performed by the Soukka group. This included X-ray diffraction (XRD) measurements to verify the hexagonal crystal phase and transmission electron microscopy (TEM) measurements to determine the size and size distribution of the UCNPs. Moreover, measurements with a inductively coupled plasma optical emission spectrometer (ICP-OES) were performed by M.Sc. Melissa Monks from the *Federal Institute for Material Research and Testing* (BAM) in order to determine the dopant Er³⁺ and Yb³⁺ content of the studied UCNP samples of Publication III.

UCNPs samples synthesized by Stefan Wilhelm from the Dr. Hirsch group, University of Regensbur g, Germany

For Publication II, OA-capped 23 nm-sized β-NaYF₄:Yb³⁺(19%),Er³⁺(2%) UCNPs were synthesized by Stefan Wilhelm, from the group of Dr. Hirsch from the University of Regensburg, using a newly developed synthesis protocol ¹³⁸ based on a protocol from Li. et al in 2008 ¹⁴⁹. This new protocol yields high quality UCNPs with a small size distribution in one large batch of about 2 grams. From this batch, surface modified UCNPs samples were produced by functionalizing these UCNPs with different surface passivating and stabilizing molecules to ensure dispersibility in different solvents.¹³⁸ Two different surface modification strategies were used to enable the transfer of the UCNPs to polar solvents like H₂O, D₂O and DMF. The first strategy involves a complete exchange of the original OA molecules with citrate and BF₄-. In the second strategy OA-capped UCNPs are shelled with a second layer of ligands molecules using DSPE, which are molecules soluble in polar and nonpolar solvents, see Table 4.1. The Hirsch group also performed the structural analysis with TEM and XRD measurements. Additionally, they did ICP-OES and dynamic light scattering (DLS) measurements to determine dopant concentrations and solvodynamic particle diameters, respectively.

Table 4.1 Investigated β-NaYF₄:Yb³⁺,Er³⁺ UC samples with respective average particle diameter, the Yb³⁺ and the Er³⁺ dopant content, passivating ligands and solvent.

β-NaYF ₄ :Yb ³⁺ ,Er ³⁺ Samples	Particle diameter	Yb ³⁺ concentration ^a	Er ³⁺ concentration ^a	Solvent	Ligand
Publication I					
UCμP _{BAM} powder	3 μm	21.4%	2.2%	-	-
UCNPs powder	25 nm	17%	3%	-	Oleic acid
UCNPs dispersion	25 nm	17%	3%	Toluene	Oleic acid
Publication II					
UCNPs dispersion	23 nm	19.3%	2.3%	Cyclohexane	Oleic acid
UCNPs dispersion	23 nm	19.3%	2.3%	DMF ^b	BF ₄
UCNPs dispersion ^c	23 nm	19.3%	2.3%	D_2O	Citrate
UCNPs dispersion ^c	23 nm	19.3%	2.3%	H ₂ O	Citrate
UCNPs dispersion	23 nm	19.3%	2.3%	D_2O	DSPE ^d
UCNPs dispersion	23 nm	19.3%	2.3%	H₂O	DSPEd
Publication III					
UCNPs dispersion	33 nm	14.4%	0.9%	Toluene	Oleic acid
UCNPs dispersion	33 nm	14.4%	2.1%	Toluene	Oleic acid
UCNPs dispersion	33 nm	14.4%	3.1%	Toluene	Oleic acid
UCNPs dispersion	33 nm	14.4%	3.8%	Toluene	Oleic acid
UCNPs dispersion	33 nm	10.6%	3.1%	Toluene	Oleic acid
UCNPs dispersion	33 nm	17.1%	3.1%	Toluene	Oleic acid
UCNPs dispersion	33 nm	20.5%	3.1%	Toluene	Oleic acid

Er³⁺ and Yb³⁺ substitute for the Y³⁺ ions of the β-NaYF₄ matrix; sum of dopant concentration of Er³⁺,Yb³⁺ and Y³⁺ equals 100%.

4.2 Optical characterization methods

In this section the custom-built new integration sphere setup for absolute measurement of Φ_{UC} and UCL is presented, as well as, the commercial setup used for time resolved PL measurements.

b N,N-dimethylformamide

these samples are not discussed in this thesis

d 1,2-distearoyl-sn-glycero-3-phospho-ethanolamine-N-[methoxy-(poly-ethylene glycol)-2000] (ammonium salt) (DSPE)

4.2.1 Integrating sphere setup (ISS)

This subsection comprehensively describes the components and characterization of the custom-built ISS. Moreover, the analysis of *P*-dependent UCL leading to information about the photonic order and (de)population processes of the UC process is presented.

4.2.1.1 Components and characterization of the ISS

Figure 4.1 shows the ISS for absolute P-dependent Φ_{UC} measurements, which I extended from an existing ISS built by Dr. Christian Würth from the BAM. ^{116, 150} A detailed description of the ISS is also provided in the Supporting Information (SI) of Publication I. This ISS consists of an excitation channel, where the excitation light is generated and guided to the sample, and a detection channel to record the emitted UCL and excitation light in absolute spectral photon fluxes ($s^{-1}m^{-2}nm^{-1}$). The heart of the ISS is the integrating sphere. Its interior is covered with a diffusive white reflective coating guarantying that after multiple reflections the incident light is evenly distributed at any point of the inner surface. Consequently, the measurements of photon fluxes with an ISS do not require any consideration of the direction of propagation, the beam geometry and polarization of the excitation and emitted light.

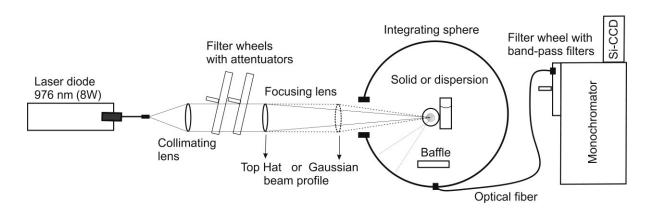


Figure 4.1 Schematic presentation of the ISS for the absolute determination of the $\Phi_{UC}(P)$ for UC powder and UC dispersions. The setup is divided into an excitation channel including the 8W 976 nm laser diode, filter wheels with attenuators with known transmission for controlled attenuation of the excitation light and focusing optics, and a detection channel including the integration sphere coupled into a filter wheel (equipped with band pass filters), monochromator, and a Si-CCD detector. The sample was mounted in the center of the integrating sphere. Reprinted from Kaiser et al.⁴⁰ with permission from The Royal Society of Chemistry.

Excitation channel

The excitation source is a special customized optical fiber-coupled laser diode (from *Roithner Lasertechnik GmbH*) fulfilling the requirements for the absolute measurement of $\Phi_{UC}(P)$ for NaYF₄:Yb³⁺,Er³⁺ UC crystals. These requirements include a high power of 8 W for characterizing the nonlinearity of the UC process, an adjusted emission wavelength of 976.4 nm that is resonant with the ground state absorption of Yb³⁺ (${}^{2}F_{7/2} \rightarrow {}^{2}F_{5/2}$) and a high power stability < 0.1 % (see Figure 4.2) as prerequisite for minimum fluctuations of the absolute Φ_{UC} measurements.

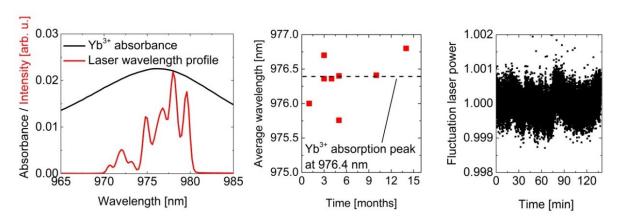


Figure 4.2 Characterization results of the 976 nm laser diode used for Φ_{UC} measurements: a) Wavelength-dependent absorbance of Yb³⁺ (976.4 nm, $^2F_{7/2} \rightarrow ^2F_{5/2}$) (black line) and wavelength profile of the 976 nm laser diode with a FWHM of about 5 nm (red line), which was adjusted to be resonant to the Yb³⁺ ground state absorption; b) Intensity averaged wavelength of the laser beam wavelength with a stability < 0.4 nm over a period of over 1 year; c) Laser power with a stability < 0.1 % for over 120 minutes. Reprinted from Kaiser et al.⁴⁰ with permission from The Royal Society of Chemistry.

The laser light from the 200 μ m-sized output slit of the optical multimode-fiber was collimated and then focused on the sample. In order to investigate excitation beam profile (BP)-dependent effects on \mathcal{O}_{UC} , two different excitation geometries were realized by using focal lenses with focal lengths of 500 mm and 125 mm to obtain a nearly homogenous Top Hat BP (TH_{exp}) and an inhomogeneous Gaussian BP (Gauss_{exp}), respectively (see Figure 4.3). The *P* tuning range of the TH_{exp} is from 0.25 Wcm⁻² to 410 Wcm⁻² and for the Gauss_{exp} from 2.5 Wcm⁻² to 3400 Wcm⁻², thus, allowing to tune *P* over four orders of magnitude. To guarantee maximal stability of *P*, the laser power was attenuated by two automated filter wheels equipped with reflective neutral density filters, which were placed between the collimating and focusing optics. To avoid damage of the laser diode by directly back-reflected

laser light, these filter wheels were tilted by a small angle. Furthermore, for safety reasons the reflected laser light was guided into a beam dump. In order to control the filter wheels, the monochromator, and to readout the detector, a Labview program was written with the help of the former bachelor student Nils Handelmann from the BAM.

Beam profile characterization

In order to verify BP-dependent effects on the $\Phi_{UC}(P)$, the two realized excitation geometries TH_{exp} and $Gauss_{exp}$ (see Figure 4.3 a)) were systematically characterized with a beam profiler (*Newport LBP2*; pixel size: 8 x 9.3 µm). Therefore, cross-sections perpendicular to the beam propagation were recorded in 1 mm steps for a path length of 10 mm illustrated in Figure 4.3 b). The average beam diameter was defined at a 4 % intensity drop of the peak intensity of the in direction of the laser beam propagation averaged intensity (see Figure 4.3 c)) to ensure comparable Φ_{UC} for a biphotonic emitter at the same averaged *P* values. This resulted in beam diameters of 1.4 mm and 0.55 mm for TH_{exp} and $Gauss_{exp}$, respectively.

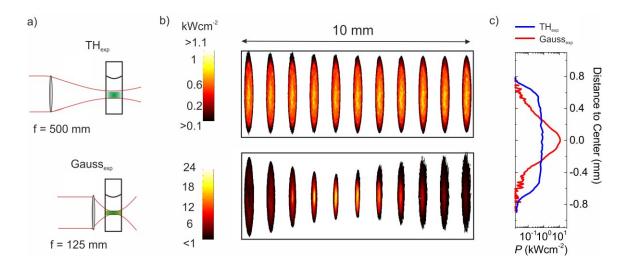


Figure 4.3 Experimentally realized illumination geometries used for the investigation of the influence of the beam shape on the Φ_{UC}(*P*) measurements. a) Schematic of the realization of TH_{exp} BP (Top) and Gauss_{exp} BP (Bottom) by the use of focal lenses with 500 mm and 125 mm focal length, respectively; b) Cross-sections of the *P*-distribution of TH_{exp} (Top) and Gauss_{exp} (Bottom) determined experimentally in 1 mm steps for a path length of 10 mm; c) Averaged *P*-distribution in the propagation direction of the laser beam for TH_{exp} and Gauss_{exp} for a laser power of 8 W. Reprinted from Kaiser et al.⁴⁰ with permission from The Royal Society of Chemistry.

Detection channel

The first part of the detection channel is a BaSO₄ inner-coated integrating sphere (from *Labsphere*) with a diameter of 15 cm, reflection over 97% in the vis to NIR spectral region), in which the sample was center-mounted with a BaSO₄ coated sample holder. At the bottom of the integrating sphere an optical fiber was attached guiding the light into a monochromator (Shamrock 303i, grating with blaze angle of 500 nm, spectral resolution 0.5 nm, *Andor Technology PLC*) equipped with a filter wheel and a silicon-based charged-coupled-device (Si-CCD) detector (Andor iDus CCD DU420-BRDD, 1024 x 256 Pixel with a pixel width of 26 μ m, *Andor Technology PLC*). A BaSO₄-coated baffle was placed inside the integrating sphere above the optical fiber to ensure that only diffusely reflected light was detected. The filter wheel was equipped with five different bandpass filters to suppress overexposure by stray light of the intense laser light and to cover the complete wavelength region from 370 nm - 900 nm for the UCL detection. The laser light was recorded with a calibrated absorptive neutral density filter with an attenuation factor of 5,600.

Calibration of the detection channel

The wavelength calibration and spectral calibration of the detection channel was performed according to previously reported protocols. 12, 150 Therefore, a correction function for the wavelength scale was created by using a Hg-Ar discharge lamp from Ocean Optics (HR4000CG-UV-NIR). This included the recording of the spectral atomic lines with the ISS and comparison with the atomic spectra database from the National Institute of Standard and Technology (NIST). This correction function was then used for all further measurements. The optical response of the detection system was calibrated with a spectral radiance transfer standard, which wavelength dependence ($L_{\lambda}(\lambda)$) was calibrated against a black body radiator by The National Metrology Institute of Germany (PTB). This spectral radiance transfer standard consists of a halogen lamp mounted inside an integrating sphere, to guarantee a diffuse spectral radiance. In order to determine the spectral instrument response functions, the spectrum of the spectral radiance standard was recorded from 350 nm - 1050 nm for all different bandpass filters used in the detection filter wheel. The obtained instrument response functions were then multiplied with λ/hc_0 to obtain correction functions in units of spectral photon flux s⁻¹m-²nm⁻¹. Additionally, an intensity correction function for different illumination times of the Si-CCD detector has been recorded The validation of the instrument response functions was performed by recording and comparing the intensity corrected spectra of the certified spectral emission standards F003 - F005 from the BAM. 151 In order to control the filter wheels, the monochromator, and to readout the detector, a Labview

program was written with the help of the former bachelor student Nils Handelmann from the BAM. Moreover, this program performed an automatic correction of the wavelength scale and the spectral sensitivity of the readout data of the detector. Furthermore, the complete analysis of the data, see *4.2.1.2 Data analysis*, was automatized with an additionally self-written Labview program.

Sample cells

All sample cells consist of high-quality quartz suprasil (QS) glass with an optical transmission > 80% from 200 nm - 2500 nm. Small sample volumes have been used to suppress indirect excitation and reabsorption effects (see 5.1. Measurement conditions for $\Phi_{UC}(P)$). The UCNP dispersion samples were filled into 10 mm (inner length) x 4 mm (inner thickness) quartz cuvettes purchased from Hellma GmbH. As a blank sample, i.e. reference, a cuvette filled only with the pure solvent was used. The powder samples were pressed into 5 mm (inner diameter) x 0.1 mm (inner thickness) custom-designed round quartz cuvettes produced by Hellma GmbH. In this case, an empty round cuvette was used as a blank sample. Identical measurement positions were ensured using the reflection with a Helium-Neon (He-Ne)-laser.

4.2.1.2 Data analysis

In this subsection the analysis of the data obtained with the ISS is described. First, the calculation of the P-dependent Φ_{UC} is presented. Second, the further analysis of the P-dependent UC emission spectra is detailed, which reveal additional information in respect photonic order and the (de-)population pathways regarding the different UC emission bands.

Calculation of the P-dependent upconversion quantum yield ($\Phi_{UC}(P)$)

The *P*-dependent Φ_{UC} curve can be calculated as the quotient of the emitted and the absorbed photon flux, see Equation 4.1. This calculation requires the measurement of a) the *P*-dependent spectrally-corrected sample spectrum $(I_{sample}(\lambda, P))$ and b) the *P*-dependent spectrally-corrected blank sample spectrum $(I_{blank}(\lambda, P))$.

$$\Phi_{UC}(P) = \frac{\text{emitted UC photon flux}}{\text{absorbed photon flux}} = \frac{\int_{350 \text{ nm}}^{900 \text{ nm}} I(\lambda, P) d\lambda}{\int_{968 \text{ nm}}^{982 \text{ nm}} I_{\text{abs}}(\lambda, P) d\lambda} = \frac{\int_{350 \text{ nm}}^{900 \text{ nm}} I_{\text{sample}}(\lambda, P) - T I_{\text{blank}}(\lambda, P) d\lambda}{\int_{968 \text{ nm}}^{982 \text{ nm}} I_{\text{sample}}(\lambda, P) d\lambda} = \frac{\int_{350 \text{ nm}}^{900 \text{ nm}} I_{\text{sample}}(\lambda, P) - T I_{\text{blank}}(\lambda, P) d\lambda}{\int_{968 \text{ nm}}^{982 \text{ nm}} I_{\text{blank}}(\lambda, P) d\lambda}$$

$$\text{Equation 4.1}$$

The *P*-dependent spectral photon flux ($I(\lambda,P)$), needed for the calculation of the emitted UC photon flux, was determined with a stray light correction from the excitation light of $I_{\text{sample}}(\lambda,P)$. Therefore, $I_{\text{sample}}(\lambda,P)$ was subtracted by $I_{\text{blank}}(\lambda,P)$ multiplied with a factor T to resulting to $I(\lambda,P)$. Thereby, the factor T considers scattering of the narrow band intense excitation light inside the detection monochromator. This can result in artefacts in the emission spectra, and subsequently, affects the calculated Φ_{UC} . Particularly, for weakly emissive UC samples this stray light correction has to be considered. The emitted absolute photon flux, $I(\lambda,P)$ was then calculated by integration from 350 nm - 900 nm, thereby including all UC emission bands of Er^{3+} and omitting the sensitizer Yb³⁺ luminescence above 900 nm. The absorbed absolute photon flux was obtained by integrating the absorbed photon flux $I_{\text{abs}}(\lambda,P)$, equaling the difference of $I_{\text{blank}}(\lambda,P)$ and $I_{\text{sample}}(\lambda,P)$, over the wavelength range of the excitation peak, here 968 nm - 982 nm (see Figure 4.2 a)).

Analysis of the *P*-dependent spectral photon flux $I(\lambda,P)$

Figure 4.4 a) presents a set of $I(\lambda,P)$ for a respective UC sample for a broad P-range. This set of $I(\lambda,P)$ can be used to obtain the following parameters providing information on of the UC processes for the different Er^{3+} emission bands: i) P-dependent integrated spectral photon fluxes $(I_{\Delta\lambda}(P))$, ii) P-dependent relative spectral photon fluxes $(I_{rel,\Delta\lambda}(P))$, iii) P-dependent green-to-red ratios $(I_{green}/I_{red}(P))$, and iv) P-dependent slope factors (n(P)) (see Figure 4.4 b), c), d) and e)). In the following these parameters are defined.

P-dependent integral spectral photon flux $(I_{\Delta\lambda}(P))$

The *P*-dependent integral spectral photon flux ($I_{\Delta\lambda}(P)$) was determined for each Er³⁺ emission band by integrating $I(\lambda,P)$ over the integration intervals presented in Table 4.2. The $I_{\Delta\lambda}(P)$ of the Er³⁺ green, red and purple emission bands are presented for a respective UC sample in Figure 4.4 b).

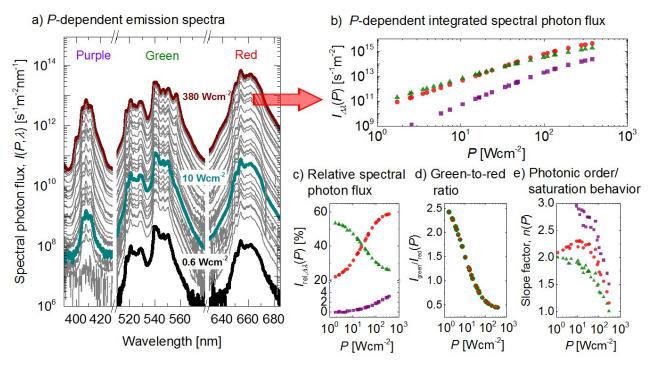


Figure 4.4 Parameters derived from the *P*-dependent spectral photon flux ($I(\lambda,P)$): a) $I(\lambda,P)$ of Er³⁺ purple (${}^2H_{9/2} \rightarrow {}^4I_{15/2}$), Er³⁺ green (${}^2H_{11/2}, {}^4S_{3/2} \rightarrow {}^4I_{15/2}$) and Er³⁺ red (${}^4F_{9/2} \rightarrow {}^4I_{15/2}$) emission bands with *P* varying from 0.6 Wcm⁻² to 380 Wcm⁻² for a representative UCNPs sample; b) *P*-dependent integral spectral photon flux ($I_{\lambda}(P)$); c) *P*-dependent relative spectral photon flux ($I_{rel,\Delta\lambda}(P)$) presenting changes in the population density of the respective emissive states; d) *P*-dependent green-to red ratio ($I_{green}/I_{red}(P)$); e) *P*-dependent slope factors (I_{P}) revealing the photonic order and saturation behavior of the UC process.

Table 4.2 Intervals for integration of the intensity of the Er³⁺ UC emission bands.

$\Delta\lambda$, color/center wavelength	Electronic transition	Integration interval	
UV/380 nm	$^{4}G_{11/2} \rightarrow ^{4}I_{15/2}$	372 nm - 394 nm	
Purple/410 nm	$^{2}H_{9/2} \rightarrow ^{4}I_{15/2}$	394 nm - 430 nm	
Green/520nm,540 nm	$^{2}H_{11/2}$, $^{4}S_{3/2} \rightarrow ^{4}I_{15/2}$	510 nm - 570 nm	
Red/655 nm	${}^{4}F_{9/2} \rightarrow {}^{4}I_{15/2}$	630 nm - 685 nm	
NIR 1/810 nm	$^{4}I_{9/2} \rightarrow ^{4}I_{15/2}$	783 nm - 833 nm	
NIR 2/850 nm	$^{2}H_{11/2}$, $^{4}S_{3/2} \rightarrow ^{4}I_{13/2}$	833 nm - 880 nm	
Overall UCL intensity		370 nm - 900 nm	

P-dependent relative spectral emission intensity ($I_{rel,\Delta\lambda}(P)$)

The *P*-dependent relative spectral emission intensity ($I_{rel,\Delta\lambda}(P)$) is defined by the quotient of $I_{\Delta\lambda}(P)$ to the overall UC photon flux ($I_{\lambda all}(P)$), see Equation 4.2.

$$I_{\text{rel},\Delta\lambda}(P) = \frac{I_{\Delta\lambda}(P)}{I_{\lambda\text{all}}(P)} = \frac{\int_{\lambda_{\text{em,low}}}^{\lambda_{\text{em,low}}} I(\lambda,P) d\lambda}{\int_{350 \text{ nm}}^{900 \text{ nm}} I(\lambda,P) d\lambda}$$
 Equation 4.2

Here, $\lambda_{\rm em,up}$ and $\lambda_{\rm em,low}$ present the wavelengths of the upper and lower bounds used for the integration of the UC emission bands, see Table 4.2. The $I_{\rm rel,}\Delta\lambda(P)$ of the Er³⁺ green, red and purple emission bands are presented in Figure 4.4 c).

 $I_{\text{rel},\Delta\lambda}(P)$ provides information about the population density of the respective energy levels in dependence of P. The shape of $I_{\text{rel},\Delta\lambda}(P)$ was used to optimize rate equation constants in Publication II and III. Moreover, the crossing point of the Er³⁺ green and red intensity, see Figure 4.4 c), was recently used as indication point for the UC efficiency for β -NaYF₄:Yb³⁺,Er³⁺ UC systems with similar Yb³⁺ and Er³⁺ dopant concentrations.³⁹

It should be noted, to avoid misunderstandings, that in Publications I and II $I_{\text{rel},\Delta\lambda}(P)$ referred to $I_{\Delta\lambda}(P)$. The nomenclature defined in this thesis is in accordance to the nomenclature used in Publication III.

Green-to-red ratio (I_{green}/I_{red})

 $I_{\text{green}}/I_{\text{red}}(P)$ is the P-dependent ratio of the integral photon flux of the Er³⁺ green (520,540 nm, $^2H_{11/2}$, $^4S_{3/2} \rightarrow ^4I_{15/2}$) to Er³⁺ red (655 nm, $^4F_{7/2} \rightarrow ^4I_{15/2}$) emission intensity, see Equation 4.3 and Figure 4.4 d).

$$I_{\text{green}}/I_{\text{red}}(P) = \frac{\int_{510 \text{ nm}}^{570 \text{ nm}} I(\lambda, P) d\lambda}{\int_{630 \text{ nm}}^{685 \text{ nm}} I(\lambda, P) d\lambda}$$
 Equation 4.3

This quantity is commonly used by the UC community to compare the properties of UC samples with similar material composition and size. A high $I_{green}/I_{red}(P)$ for Yb^{3+} , Er^{3+} -based upconverters can account for low nonradiative rates as the red emissive energy level can be directly fed from the $Er^{3+} H_{11/2}$, $^4S_{3/2}$ energy level via nonradiative relaxation (see Figure 2.7).

Slope factor (n(P))

The P-dependent slope factor (n(P)) represents the nonlinear increase of the emission intensity with P (see Equation 4.4 and Equation 4.5). At low P, n(P) typically presents the photonic order of the UC process. For example, in case of a biphotonic process n(P) equals two at low P. Saturation of the UC process at higher P results in a decrease of n(P). For the calculation of n(P) energetically neighboring data points $I(P_1), I(P_2)$ were used, see Figure 4.4 b) and Figure 4.4 e) and Equation 4.6.

$$I(P) \propto P^{n(P)}$$
 Equation 4.4

$$n(P) = \frac{d \log (I(P))}{d \log (P)}$$
 Equation 4.5

$$n(P) = \frac{\log(I(P_2)) - \log(I(P_1))}{\log(P_2) - \log(P_1)}$$
 Equation 4.6

An energy level can also be populated by a mixture of processes with different photonic orders like the Er^{3+} red emissive ${}^4F_{9/2}$ energy level, which can be populated via bi- and triphotonic processes, see Figure 2.6. Such mixtures of different photonic population processes can be characterized via the shape of n(P) of the Er^{3+} red emission band, see Figure 4.4 e).

4.2.2 Time-resolved photoluminescence spectroscopy

The lifetime curves of the emissive energy levels, characterized by their rise and decay behaviors (see Figure 4.5), provide information about the temporal evolution of the population and depopulation processes. Measurements of the lifetime curves were performed with a commercial *Edinburgh Instruments* spectrofluorometer FSP-920. This setup was equipped with an electrically-pulsed 1 W 978 nm laser diode (950 μ s long square pulse) to excite the sensitizer Yb³⁺ ions for the determination of the UCL lifetime curves for the Er³⁺ 380 nm, 410 nm, 520/540 nm, 655 nm, 810 nm and the downshifted (DS) lifetime curve of the 1520 nm emission bands. Additionally, DS luminescence lifetimes were determined by directly exciting the Er³⁺ ions with an electrically pulsed xenon lamp (pulse width *ca.* 1 μ s) combined with an excitation double monochromator to select the respective excitation wavelength. This provided the DS luminescence curves of the Er³⁺ 410, nm, 520/540 nm, 655 nm emission bands. Moreover, the DS lifetime of Yb³⁺ 976 nm emission

band was determined with excitation wavelength of 940 nm. The detection channel was also equipped with a double monochromator to select the respective emission wavelength for detection. The luminescence lifetime curves were then recorded with a red extended photomultiplier tube (PMT) (R2658P) for the UV-NIR region, and a nitrogen cooled NIR PMT (R5509P) for the Er³⁺ 1520 nm emission band. Distortion by stray light or undesired intensity of other emission bands were canceled out by using different band pass filters.

In order to estimate the influence of the long-pulse excitation of the above described setup, I also performed short-pulse measurements with a setup of the work group of Prof. Dr. Hoffmann from the *Technical University of Berlin* for the UC samples investigated in Publication I. Here, luminescence lifetime curves were obtained with short pulse excitation with pulse lengths of *ca.* 10 ns and an excitation wavelength of 940 nm using a dye (IR140) laser pumped with an excimer laser, see Figure 4.6. The deviations of the derived decay times for short- and long-pulse excitation using a differential method, see Supporting Information of Publication I, were < 10%.

It should be noted that the determined UC lifetimes are always results of combinations of decay and ET processes.¹⁵² Furthermore, DS decay lifetimes can be affected by CR processes.¹⁵² This underlines the fact that only with aid of a comprehensive rate equation analysis, taking into account the significant interaction between the Yb³⁺ and Er³⁺ energy levels, the population and depopulation dynamics can be described accurately as done in Publications II and III.

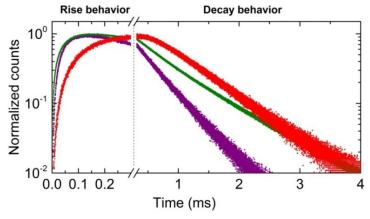


Figure 4.5 Temporal rise and decay behavior of μm-sized β-NaYF₄:Yb³⁺(21%),Er³⁺(2%) UC particles for the Er³⁺ purple at 410 nm (${}^2H_{9/2} \rightarrow {}^4I_{15/2}$), Er³⁺ green 520 nm/540 nm (${}^2H_{11/2}, {}^4S_{3/2} \rightarrow {}^4I_{15/2}$) and 655 nm (${}^4F_{9/2} \rightarrow {}^4I_{15/2}$) emission bands. The lifetime curves were obtained using a dye (IR140) laser pumped with an excimer laser with short pulse excitation of 10 ns with an excitation wavelength of 940 nm with a setup from the work group of Prof. Dr. Hoffmann from the *Technical University of Berlin* (TU-Berlin).

5 Measurement Strategies for the *P*-dependent $\Phi_{UC}(\Phi_{UC}(P))$

In this chapter prerequisites and challenges for absolute P-dependent Φ_{UC} measurements with the custom-built ISS (see 4.2.1 Integrating sphere setup (ISS)) are presented for β -NaYF₄:Yb³⁺,Er³⁺ UC crystals. The first part of this chapter addresses stringent requirements for the optical properties of the sample, the measurement geometry, and measurement procedure. In the second part, the $\Phi_{UC}(P)$ values of μ m-sized β -NaYF₄:Yb³⁺(21%),Er³⁺(2%) UC particles (UC μ P) are validated by comparison to experimental and theoretical literature data.

5.1. Measurement conditions for $\Phi_{UC}(P)$

In this section, the prerequisites and measurement strategies of the Φ_{UC} measurement are detailed. Disregarding these recommendations can lead to a high fluctuation, systematic underestimation or overestimation of the measured Φ_{UC} values. The following instructions present a guideline for accurate P-dependent Φ_{UC} measurements with minimum uncertainty.

Absorption of the UC sample

An accurate Φ_{UC} measurement can be only guaranteed for a UC sample with a suitable absorption. This parameter can be tuned by either varying the particle concentration or the optical path length. The lower measuring limit of the UC sample absorption sensitively depends on the stability of the excitation source and on the reproducibility of measurement conditions of the sample and the blank. Whereas, the upper measuring limit depends on the optical path length and nonlinearity of the UC material.

Figure 5.1 shows that with the use of a high-stability excitation source of 0.1%, as utilized in this work, see Figure 4.2 c), a reasonable Φ_{UC} determination requires UC samples with an absorption larger than 1% to minimize fluctuation of a single Φ_{UC} measurement down to 10%. However, systematic errors regarding the positioning, the preparation and the optical properties of the sample and the blank must also be considered. Although identical positioning can be ensured with the reflection of a He-Ne laser, small slight deviations of the reflectance of the quartz cuvettes from about 0.1% - 0.4% showed to be a crucial uncertainty source at low UC sample absorptions. Therefore, a **lower limit of 5%** for the absorption of the UC sample is recommended to minimize these systematic errors. In case of UCNPs

dispersions with very low particle concentrations, this recommendation may not be fulfilled. However, the use of the same quartz cuvette for UC sample and blank has still proven to be moderately precise for absorption measurements down to 2%.

A strong absorptive UC sample can lead attenuation of the laser beam inside the sample volume resulting to inhomogeneous spatial emission for these nonlinear emitters. To minimize this effect, an **upper limit of about 20%**, for the absorption of the UC sample with an optical path length of 10 mm is recommended.

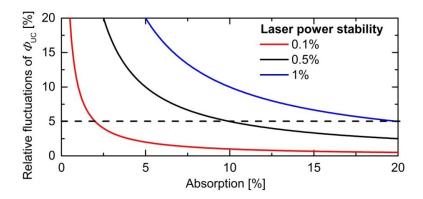


Figure 5.1 Relative fluctuations for a single Φ_{UC} measurement in dependence of the fraction of absorbed light for different laser power stabilities. The 976 nm laser diode used for the absolute Φ_{UC} measurements in this thesis has a power stability < 0.1 %. The relative fluctuations of the Φ_{UC} measurements were calculated from the quotient of the relative laser power stability to the relative absorption of the UC sample.

Scattering of the UC sample

Another common issue, for Φ_{UC} measurements, results from the scattering of the excitation light by the UC sample. Particularly, scattering can critically influence the Φ_{UC} measurements for i) dispersions with UCNPs sizes larger than 50 nm or agglomerated UCNPs and ii) for powder samples. Additionally, scattering of the excitation light by UCNPs dispersion can lead to a lowered P inside the sample volume, and hence, results in underestimated Φ_{UC} values. Therefore, it was ensured for all investigated UCNPs dispersions that no detectable scattering occurs at the excitation wavelength, see Figure 5.2.

Light scattering by powder samples can lead either to a decrease of P, due to a diffusion of the incident light, or even to an increase of P by multiple reflections, dependent on the preparation of the powder sample. In order to minimize internal scattering effects, the powder samples were prepared in thin cell cuvettes having an inner thickness of 100 μ m. In addition, the sample holder, placed in the center of the integrating sphere, was tilted by 30° to avoid that diffuse back-scattered excitation light leave the integrating sphere via the

entrance port. This excitation geometry has the advantage that there is no need for a nonabsorbing reference sample with similar scattering properties compared to the powder sample, since most of the reflected incident light is collected with the integrating sphere.

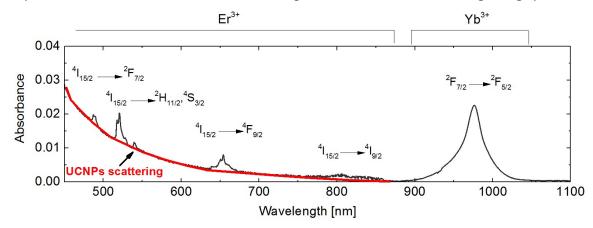


Figure 5.2 Absorbance spectra of 25 nm-sized UCNPs dispersed in toluene taken from Publication I. This shows the absence of scattering at the excitation wavelength of 976 nm for this UCNPs dispersion, ensuring no effect on the measured Φ_{UC} . Reprinted from Kaiser et al.⁴⁰ with permission from The Royal Society of Chemistry.

Yb³⁺ 980 nm emission (${}^{4}F_{5/2} \rightarrow {}^{4}F_{7/2}$)

Figure 5.3 a) schematically shows the distortion of the measured absorption, due to the overlap of emitted light from the sensitizing Yb³⁺ ion with excitation light. This issue was addressed by subtracting the emitted intensity of Yb³⁺ from the recorded intensity.¹³² Therefore, the real shape of the Yb³⁺ 980 nm emission band was measured separately with an excitation wavelength < 950 nm and then normalized to the distorted recorded sample spectrum. Fischer et al. reported deviations of up to 40% for Φ_{UC} of core-shell UCNPs between Yb³⁺ emission-corrected and -uncorrected values.¹³² For the UCNPs studied in this work, no Yb³⁺ 980 nm emission-correction was necessary, as the Yb³⁺ emission intensity is weak for these unshelled systems.⁴⁰ In case of the μ m-sized β -NaYF₄:Yb³⁺,Er³⁺ particles (UC μ P), investigated in Publication I, a Yb³⁺ 980 nm emission-correction was required. The relative deviations of Yb³⁺ emission-corrected to -uncorrected absorption as well as associated Φ_{UC} values additionally show a *P* dependence with 14% at a *P* of 0.2 Wcm⁻² down to 3% for a *P* of 100 Wcm⁻², see Figure 5.3 b) and c).

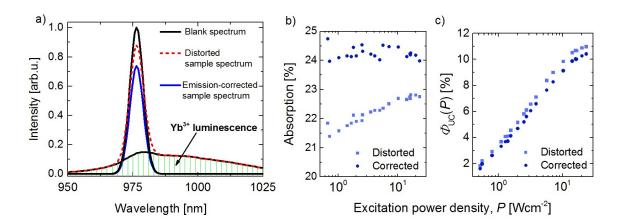


Figure 5.3 Emission-correction method of the $\Phi_{UC}(P)$ for the intensity of the Yb³+ 980 nm emission band (${}^4F_{5/2} \Rightarrow {}^4F_{7/2}$). a) Schematic of the Yb³+ emission-correction method including the Yb³+ 980 nm emission band (black line, area under curve in shaded green), blank spectrum (black line), distorted sample spectrum (dashed red line) and Yb³+ 980 nm emission-corrected sample spectrum (blue line); b) Yb³+ emission-corrected absorption values and c) Yb³+ emission-corrected $\Phi_{UC}(P)$ values; of the UCµP sample studied in Publication I. Reprinted from Kaiser et al.⁴⁰ with permission from The Royal Society of Chemistry.

Reabsorption and indirect excitation

Multiple diffuse reflections inside the integrating sphere can lead to *reabsorption* of the emitted light and *indirect absorption* of the excitation light by the UC sample. Thereby, reabsorption can reduce the emitted photon flux, whereas indirect excitation is associated with a very low P, both leading to underestimated Φ_{UC} values. These values can be corrected with a method developed by MacDougall et al., who extended the widely known reabsorption-correction for linear emitters developed from de Mello. However, these effects can be easily circumvented by minimizing the sample volume compared to the integrating sphere volume, making the events of reabsorption and indirect excitation negligible, and thus, a correction obsolete. Suitable sample cell sizes can be determined by measuring Φ_{UC} for a series of sample cells with varied volumes.

Excitation beam profile (BP)

Linear emitters like fluorescent dyes and QDs do not need any consideration of the P-inhomogeneity of the BP. However, UCL depends strongly on P. Consequently, an inhomogeneous BP results in a distribution of $\Phi_{UC}(P)$ for these nonlinear emitters.

These BP-dependent effects on the $\Phi_{UC}(P)$ were quantified by a calculation using the experimentally realized BPs (see Figure 4.3). For this pupose, the formula of the Anderson

Engels group for the $\Phi_{UC}(P)$ of an ideal biphotonic emitter (see Equation 2.8) was extended by including the spatial P-distribution of the BP of the illuminated sample volume (detailed in Publication I). Figure 5.4 shows the P-dependent deviation of the $\Phi_{UC}(P)$ for the experimentally realized TH_{exp} and $Gauss_{exp}$ shaped BPs (see Figure 4.3) compared to an ideal Top Hat (TH_{ideal}) shaped BP, i.e. a perfectly homogenous BP. At low P, The deviations of $\Phi_{UC}(P)$ for $Gauss_{exp}$ and TH_{exp} account to about 45% and 10%, respectively. This underlines the need for a homogeneous Top Hat BP for accurate $\Phi_{UC}(P)$ measurement.

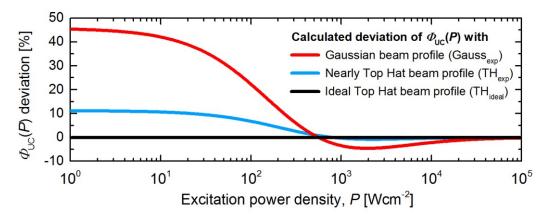


Figure 5.4 Calculated deviation of $\Phi_{UC}(P)$ for experimentally realized beam profiles. Here, the experimentally realized TH_{exp} and Gauss_{exp} BPs (see Figure 4.3) were compared to a TH_{ideal} BP. At low P, the deviation of the simulated $\Phi_{UC}(P)$ values obtained with the TH_{exp} and Gauss_{exp} BPs accounts for up to ca. 10% and 45%, respectively. Reproduced from Kaiser et al.⁴⁰ with permission from The Royal Society of Chemistry.

Heating effects at high laser power

The excitation of the UC sample with high laser powers is accompanied by an increase of its temperature. This results in a decrease of the measured $\Phi_{UC}(P)$ due to thermal deactivation of the emissive Er^{3+} energy levels. The heating of the UC sample depends on the measurement time, laser power, and beam diameter, as well as on the UC sample specific properties like size, absorptivity, crystalline quality, and the optical and thermal properties of its environment (e.g. absorption and heat capacity of the solvent). By using a short measurement time of less than 30 s, the thermally induced intensity loss was limited to maximally 30 % at the highest P used for the investigated UC samples studied. In particular, powder samples found to be quite prone to degradation for $P > 130 \, \text{Wcm}^{-2}$ for the used beam diameters of sizes in the mm-range. Instead, smaller beam diameters in the μ m-range used typically in microscopic studies allows excitation of UC powders with very high P of more than $10^6 \, \text{Wcm}^{-2}$. UCNPs dispersions generally have a better heat transfer than powders

allowing excitation with laser powers of more than 8 W (used for the here investigated UCNPs dispersions) without harming the samples.

The property of UCNPs to act as nanothermometers allows monitoring the temperature increase for different P. Therefore, the temperature-dependent intensity-ratio $I_{520\text{nm}}/I_{540\text{nm}}(T)$ of the thermally coupled Er^{3+} 520 nm ($^2\text{H}_{11/2} \rightarrow ^4\text{I}_{15/2}$) and 540 nm ($^4\text{S}_{3/2} \rightarrow ^4\text{I}_{15/2}$) emission bands is utilized, see Figure 5.5 a). $^{156,\ 157}$ Note that the $I_{520\text{nm}}/I_{540\text{nm}}(T)$ depends on many parameters, like e.g. size, solvent and dopant concentration of the UCNPs and has to be calibrated separately. Exemplarily, Figure 5.5 b) represent the temperature of UCNPs dispersed in water (H_2O) and heavy water (D_2O) for different P, excited with a Gauss_{exp} BP for under 30 s, with an optical path length of 10 mm and a sample volume of about 1 ml. The drastic increase of the temperature from 25 °C to 60 °C of UCNPs dispersed in H_2O , at P=1000 Wcm⁻², mainly results from the high absorption of H_2O of 40% at 976 nm for the chosen 10 mm optical path length. This underlines the critical influence of absorbing solvents on the actual temperature of the sample for Φ_{UC} measurements. In contrast, for UCNPs dispersed in D_2O the temperature increase is much less pronounced with D_2O at D_2O the temperature increase is much less pronounced with D_2O at D_2O does not absorb at the excitation light.

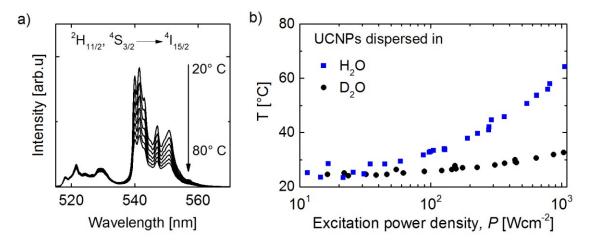


Figure 5.5 Monitoring of the temperature increase of UCNPs by laser excitation during the $\Phi_{Uc}(P)$ measurement: a) Temperature-dependent Er^{3+} 520 nm to Er^{3+} 540 nm emission intensity ratio $I_{520nm}/I_{540nm}(T)$ from 20° C - 80° C separately recorded and used for calibration; b) P-dependent temperature increase for DSPE-capped UCNPs dispersed in H_2O (blue) and D_2O (black), for standard Φ_{UC} measurement conditions described in the text, calculated with the $I_{520nm}/I_{540nm}(T)$. Reproduced from Würth, Kaiser et al. 55 with permission from The Royal Society of Chemistry.

Φ_{UC} measurement in H₂O at high P

Accurate Φ_{UC} of measurements for UCNPs dispersed in H₂O are very challenging, since H₂O has a high absorption coefficient at the excitation wavelength of 976 nm. This results in heating up of the UCNPs sample for high P (see Figure 5.5 b)) resulting in a decrease in intensity. Moreover, the absorption coefficient of H₂O increases with increasing temperature, which requires more specific measurement conditions for high P to avoid overestimation of the measured Φ_{UC} . This implies that blank and sample spectrum must be recorded with the same illumination times to match their temperatures. Additionally, the absorption of the UCNPs dispersion should be kept low at ca. 5-10%, so that nonradiative deactivation processes minimally contribute to the increase of the temperature. In addition, the high absorption of H₂O leads to a lowered average P and to an increase of the inhomogeneity of the BP. As discussed earlier in this section, an inhomogeneous BP leads to overestimated Φ_{UC} values.

5.2 Validation of measured $\Phi_{UC}(P)$

The aim of this section is to validate the measured $\Phi_{UC}(P)$ data by comparing with literature data obtained from different laboratories. For this purpose, high-quality bulk systems with similar dopant concentrations are the best choice due to their low nonradiative rates, and hence, well-comparable optical properties. As a representative bulk material, commercially available 3 μ m-sized β -NaYF₄:Yb³⁺(21%),Er³⁺(2%) UC particles (**UC** μ P_{BAM}) (from *Phosphor Technologies*)¹⁴⁷ were chosen. Reliable reports of Φ_{UC} measurements of bulk β -NaYF₄:Yb³⁺,Er³⁺ are rare since these measurements are challenging, as intensively discussed in the previous sections of this thesis. Reported Φ_{UC} data usually includes the intensity data of a limited number or even only for a single UC emission band. Furthermore, most works provide Φ_{UC} data for small *P*-range < 2 orders of magnitude or even only a single *P*, which is not enough to fully characterize the saturation behavior of the UC processes. Therefore, this work provided a comprehensive characterization of the Φ_{UC} of **UC** μ P_{BAM} regarding the full spectral region from 360 - 900 nm, considering all UC emission bands with a significant contribution to the overall UCL, recorded for a wide *P* range of over three orders of magnitude with a well-characterized TH-shaped excitation beam profile.

Three reports were used for the comparison of the measured $\Phi_{UC}(P)$ of $UC\mu P_{BAM}$. The first report from Page et al. (1997) provides the $\Phi_{UC}(P)$ of the Er^{3+} green emission band $(\Phi_{UC,green}(P))$ for μ m-sized β -NaYF₄:Yb³⁺,Er³⁺ powder ($UC\mu P_{Page}$). However, the dopant concentration for $UC\mu P_{Page}$ was not specified. In the second report from the Van Veggel group, the maximal $\Phi_{UC,green}$ for β -NaYF₄:Yb³⁺(20%),Er³⁺(2%) particles with sizes >> 100 nm ($UC\mu P_{VanVeggel}$) was determined. Both of these reports used an ISS for the absolute Φ_{UC} determination. The third report from the Berry group in 2014 provides simulated values of the spectral $\Phi_{UC}(P)$'s of different Er^{3+} emission bands of μ m-sized rod-shaped β -NaYF₄:Yb³⁺(18%),Er³⁺(2%) crystallites ($UC\mu P_{Berry}$) (from Lorad Chemical Corp.). 102

5.2.1 Validation by comparison with measured results from Page et al.

In 1997, Page et al. reported on the absolute determination of UC conversion efficiencies for single emission bands for different green-, red- and blue-emitting UC phosphors. Thereby, they provided a detailed description of the spectral calibration of their ISS and measurement procedure. For UC μ P_{Page} they determined the Φ _{UC,green}(P) for a broad P-range of over three orders of magnitude.

Figure 5.7 a) displays the $\Phi_{UC,green}(P)$ curves for $UC\mu P_{BAM}$ and $UC\mu P_{Page}$ for a P-range from 0.2 Wcm⁻² to 130 Wcm⁻². The excellent agreement of these $\Phi_{UC,green}(P)$ curves with similar increase and identical maximal $\Phi_{UC,green}$ of 2.4% at P=20 Wcm⁻² implies comparability of these materials, i.e. crystal phase, crystal quality and Yb³⁺ and the Er³⁺ dopant concentrations. Both $\Phi_{UC,green}(P)$ curves starts to decrease for P>20 Wcm⁻², which may be attributed to a competition of triphotonic to biphotonic processes at high P. In this respect, thermal effects as reason for this decrease were excluded, see Publication I.

Figure 5.7 b) presents the normalized spectrally-corrected emission spectra of the Er³⁺ green and Er³⁺ red emission bands for UC μ P_{BAM} to UC μ P_{Page} at a *P* of ca. 20 Wcm⁻². The good match of these normalized spectra further supports the assumption that these materials are comparable. Moreover, the domination of the Er³⁺ red emission intensity, known for its triphotonic activation for μ m-sized crystals,¹⁰² strengthen the hypothesis of triphotonic processes being responsible for the fall of the Φ _{UC,green}(*P*) at high *P*.

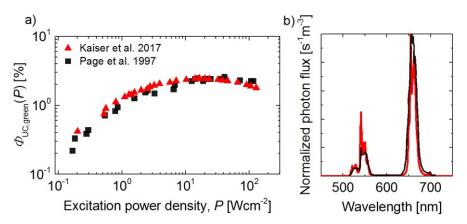


Figure 5.7 Validation of absolutely measured $Φ_{UC}(P)$ of the Er³⁺ green emission band ($Φ_{UC,green}(P)$) for UCμP_{BAM} (red color) to UCμP_{Page} (black color) from Page et al. ⁵²: a) $Φ_{UC,green}(P)$ curves for UCμP_{BAM} and UCμP_{Page} displayed from 0.2 Wcm⁻² to 130 Wcm⁻² b) Normalized spectrally-corrected emission spectra for UCμP_{BAM} excited with P = 20 Wcm⁻² and UCμP_{Page} excited with a P in the Wcm⁻²-range. The data from Page was digitized and values were transformed from energy to photon numbers. Reprinted from Kaiser et al. ⁴⁰ with permission from The Royal Society of Chemistry.

5.2.2 Validation by comparison with measured results from the Van Veggel group

The Van Veggel Group published in 2010, a strategy for the absolute measurement of the Φ_{UC} for UCNPs using a commercially available spectrophotometer (FLS 920 from *Edinburgh Instruments*) combined with a 980 nm laser diode and an integrating sphere.⁵¹ This work is an important milestone in the UCNPs research area as the performance of UCNPs was previously only compared relatively. In order to validate their Φ_{UC} results, they compared the maximal $\Phi_{UC,green}$ value of 3 % for UC μ P_{VanVeggel} with the $\Phi_{UC,green}$ value of 2.4% from UC μ P_{Page}. This value also matches well with the $\Phi_{UC,green}$ value of 2.4% measured for UC μ P_{BAM}.

5.2.3 Validation by comparison with simulated results from the Berry group

In 2014, the Berry group simulated the spectral $\Phi_{UC}(P)$'s of the Er³⁺ green ($\Phi_{UC,green}(P)$), Er³⁺ red ($\Phi_{UC,red}(P)$) and Er³⁺ purple ($\Phi_{UC,purple}(P)$) emission bands. These simulation results were based on a comprehensive rate equation analysis utilizing measured P-dependent intensity and UCL lifetime data of the Er³⁺ purple, green and red emission bands. This analysis was based on a new rate equation model revisiting the back then outdated model of population processes for the different Er³⁺ energy levels of β -NaYF₄:Yb³⁺,Er³⁺.

Figure 5.6 a) - d) shows the $\mathcal{O}_{\text{UC,preen}}(P)$, $\mathcal{O}_{\text{UC,pred}}(P)$, $\mathcal{O}_{\text{UC,purple}}(P)$, and their sum ($\mathcal{O}_{\text{UC,vis}}(P)$) for UCµP_{BAM} and UCµP_{Berry}. The $\mathcal{O}_{\text{UC,vis}}(P)$ and $\mathcal{O}_{\text{UC,green}}(P)$ curves have similar shape for both samples with relative deviation < 40%. Taking into account the complexity of the population processes for the multitude Er^{3+} levels, these deviations between experimental and simulated data can be considered as relatively small. However, the $\mathcal{O}_{\text{UC,red}}(P)$ and $\mathcal{O}_{\text{UC,purple}}(P)$ show high deviations by factors of two and five, respectively. In particular for $\mathcal{O}_{\text{UC,purple}}(P)$, the shape also strongly differs between UCµP_{Berry} and UCµP_{BAM}. Moreover, the $I_{\text{green}}/I_{\text{red}}$ ratio strongly deviates by a factor of ca. 3 between UCµP_{Berry} and UCµP_{BAM}. However, the $I_{\text{green}}/I_{\text{red}}$ ratio determined for UCµP_{BAM} are supported by the values of UCµP_{Page}, see 5.2.1 Validation by comparison with measured results from Page et al.. Further, the comparability of UCµP_{Berry} and UCµP_{BAM} is underlined by similar luminescent decay kinetic of the different UC emission bands, see Figure 5.6 e).

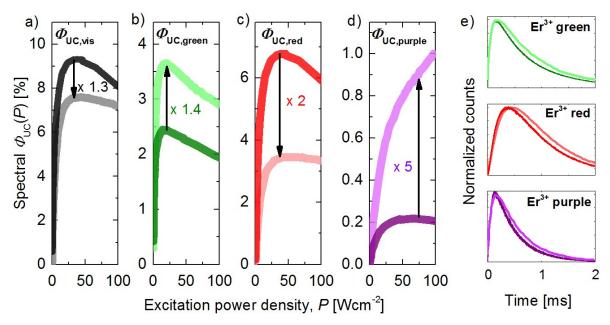


Figure 5.6 Validation of $\Phi_{UC}(P)$'s of experimental results for UC μ P_{BAM} (dark colored lines) by the simulation results for UC μ P_{Berry} (light colored lines) from the Berry group:

a)-d) P-dependent spectral Φ_{UC} of the Er³⁺ green ($\Phi_{UC,green}(P)$), red ($\Phi_{UC,red}(P)$) and purple Er³⁺ emission bands ($\Phi_{UC,purple}(P)$), as well as the vis $\Phi_{UC}(\Phi_{UC,vis}(P))$, equaling the sum of $\Phi_{UC,green}(P)$, $\Phi_{UC,red}(P)$ and $\Phi_{UC,purple}(P)$) for UC μ P_{BAM} and UC μ P_{Berry}; e) Comparison of lifetime curves for UC μ P_{BAM} (λ_{exc} = 940 nm, pulse energy of ca. 1 mJcm⁻²) compared to UC μ P_{Berry} (λ_{exc} = 943 nm, pulse energy of 66 mJcm⁻²) revealing similar (de-)population dynamics for UC μ P_{BAM} and UC μ P_{Berry}. Reproduced from Kaiser et al.⁴⁰ with permission from The Royal Society of Chemistry.

5.3. Conclusions of chapter 5

A guideline for P-dependent Φ_{UC} measurement with the newly custom-built ISS has been developed. Special emphasis was given to the challenges and requirements on the optical properties of the sample and measurement geometry for accurate $\Phi_{UC}(P)$ measurements. Overall, this underlines the need for careful consideration of these conditions to obtain P-dependent Φ_{UC} values with high precision. In particular, the choice of the BP showed to be crucial factor for an accurate determination of the $\Phi_{UC}(P)$ underlining the need for a homogenous Top Hat BP.

The obtained $\Phi_{UC}(P)$ data for a commercial μ m-sized β -NaYF₄:Yb³⁺,Er³⁺ (UC μ P_{BAM}) was validated with literature data from Page et al.⁵², the Van Veggel group⁵¹, and the Berry group¹⁰³. The independently measured maximal $\Phi_{UC,green}(P)$ of UC μ P_{BAM} and UC μ P_{Page} with 2.4% as well as UC μ P_{VanVeggel} with 3% underlines the quality of spectral calibration and suitability of the measurement strategies of the newly-developed ISS. The theoretical results for UC μ P_{Berry} of the $\Phi_{UC,vis}(P)$ and $\Phi_{UC,green}(P)$ showed deviations up to 40% compared to the measured values for UC μ P_{BAM}. Taken into account the complexity of the UC processes these deviations can be considered as relatively small. However, the high deviations of UC μ P_{Berry} and UC μ P_{BAM} for the $\Phi_{UC,red}(P)$ and $\Phi_{UC,purple}(P)$ by factors two and five, respectively, suggest that a further optimization of their rate equation model for β -NaYF₄:Yb³⁺,Er³⁺ μ m-sized UC systems is needed.

These studies laid the ground for the following quantitative optical characterization of β -NaYF₄:Yb³⁺,Er³⁺ UCNPs in dependence of the solvent and dopant concentration.

6 Solvent-Dependent Optical Properties of β–NaYF₄:Yb³⁺,Er³⁺ UCNPs

This chapter deals with the influence of particle microenvironment and thereby, also surface functionalization on the UCL and Φ_{UC} of 23 nm-sized β –NaYF₄:Yb³⁺(19%),Er³⁺(2%) UCNPs. In the first part, the impact of vibrational modes of different solvents on the Φ_{UC} is discussed. In the second part, a special emphasis was dedicated to the study of the population mechanisms of the Yb³⁺ and Er³⁺ energy levels of UCNPs dispersed in water (H₂O) and heavy water (D₂O). In contrast to H₂O, the vibrational modes of D₂O are shifted to lower frequencies, and therefore, lead less likely to the depopulation of the Yb³⁺ and Er³⁺ energy levels. This resulted to a model of the (de-)population processes for the participating Er³⁺ energy levels for UCNPs dispersed in H₂O and UCNPs dispersed in solvents without UCL quenching OH-bonds.

6.1 Solvent-dependent upconversion quantum yield (Φ_{UC})

In this section, the influence of different solvents on the $\Phi_{UC}(P)$ of identical UCNPs passivated with different surface passivating molecules was studied. Different surface chemistry, i.e. different ligands, were required to render the UCNPs dispersible in different solvents. The UCL and $\Phi_{UC}(P)$'s of the following four samples were recorded: i) oleate-capped UCNPs dispersed in cyclohexane, ii) BF₄-capped UCNPs dispersed in DMF and iii) DSPE-capped UCNPs dispersed in H₂O and iv) DSPE-capped UCNPs dispersed D₂O (see *4.1 Overview of the investigated β-NaYF₄:Yb³⁺,Er³⁺ samples)*. The passivating surface molecules have two tasks: i) to control particle colloidal stability by electrostatic or steric effects to prevent agglomeration of the UCNPs and ii) to minimize the particle surface area accessible to the solvent molecules aiming to reduce solvent-induced UCL quenching. The complete suppression of solvent-induced UCL quenching requires a tight ligand shell preventing any penetration of solvent molecules, which was not completely achieved here.

Figure 6.1 a) presents the P-dependent Φ_{UC} of the investigated UCNPs. The DSPE-capped UCNPs dispersed in D_2O showed the best performance with a Φ_{UC} of about 1.1% at $P=800~{\rm Wcm}^{-2}$. In comparison, the UCNPs dispersed in DMF, cyclohexane and H_2O perform with 90%, 75% and 40% of this Φ_{UC} value at 800 Wcm⁻², respectively. Figure 6.1 b) displays the absorbance spectra of the solvents near the spectral region of the Yb³⁺ 980 nm emission band (${}^2F_{5/2} \rightarrow {}^2F_{7/2}$) This demonstrates a correlation for the reduction of the Φ_{UC} by

the different solvents with the overlap of the absorbance spectra of the solvent and the Yb³⁺ 980 nm emission band. This overlap provides a measure for the probability of the nonradiative deactivation of near-surface excited Yb³⁺ ions *via* ET to the solvent molecules. The strong absorption of the vibrational overtone O-H-mode at 980 nm of H₂O can be identified as the main factor for the diminishing of the Φ_{UC} by a factor of 2.5 – 3 for the DSPEcapped UCNPs in H₂O compared to the UCNPs in D₂O. This is also in accordance with the reduction of the Yb³⁺ 980 nm emission lifetime from 160 μ s to 40 μ s, see Publication II. The fact that Arppe et al. reported for 38 nm-sized bare UCNPs dispersed in H₂O a UCL quenching of 99.9% of the Er³⁺ green and red UCL compared to the same UCNPs dispersed in D₂O underlines the partial repulsion of H₂O molecules from the surface of the 23 nm-sized DSPEcapped UCNPs.

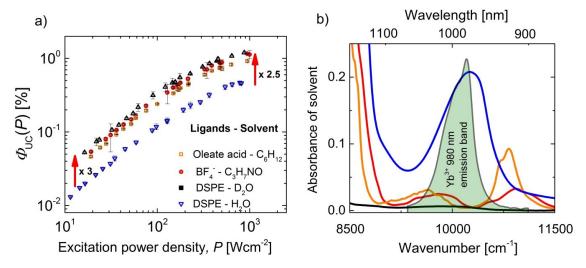


Figure 6.1 Surface quenching of 23 nm-sized β -NaYF₄:Yb³+(19%),Er³+(2%) UCNPs by solvent molecules. a) Measured $\Phi_{UC}(P)$ curves for UCNPs in different solvents: Oleate-capped UCNPs in cyclohexane (C₆H₁₂) (orange squares), BF₄-capped UCNPs in DMF (red circles) as well as DSPE-capped UCNPs in heavy water (D₂O) (black triangles) and water (H₂O) (blue triangles); b) Absorbance spectra of the solvents - same color code as in a) - and the Yb³+ 980 nm emission band (${}^2F_{5/2} \rightarrow {}^2F_{7/2}$); Reproduced from Würth, Kaiser et al.⁵⁵ with permission from The Royal Society of Chemistry.

6.2 Emission color of UCNPs dispersed in D₂O and H₂O

Subsequently, the differences in population processes at different P of the emissive Er^{3+} energy levels of the DSPE-capped UCNPs dispersed in H_2O and D_2O were investigated. Figure 6.2 a) shows the normalized UCL spectra of these UCNPs for a low P value of 16 Wcm⁻² and a high P value of 1000 Wcm⁻². The spectra display the wavelength region from 500 nm -

900 nm including the Er³⁺ green at 520/540 nm (${}^{2}H_{11/2}$, ${}^{4}S_{3/2} \rightarrow {}^{4}I_{15/2}$), Er³⁺ red at 655nm (${}^{4}F_{9/2}$ \rightarrow ⁴I_{15/2}), Er³⁺ 810 nm (⁴I_{9/2} \rightarrow ⁴I_{15/2}), and Er³⁺ 850 nm (²H_{11/2}, ⁴S_{3/2} \rightarrow ⁴I_{13/2}) emission bands. At $P = 16 \text{ Wcm}^{-2}$, an enhanced relative intensity of the Er³⁺ red and 810 nm emission bands can be observed for the UCNPs dispersed in H₂O compared to D₂O. Thereby, the relative intensities of the Er3+ green and Er3+ 850 nm emission bands, both originating from the ⁴S_{3/2}, ²H_{11/2} level are strongly reduced. This reveals that the population dynamics, at low P, differ strongly for UCNPs dispersed in H₂O and D₂O. Contrarily, at a high P of 1000 Wcm⁻², the normalized UCL spectra are nearly identical in both solvents. Thus, the population dynamics for the emissive Er³⁺ energy levels are assumed to be comparable. These observations are supported by the P-dependent behavior of I_{green}/I_{red} ($I_{green}/I_{red}(P)$), see Figure 6.2 b). For UCNPs in D₂O, $I_{green}/I_{red}(P)$ decreases for increasing P and converge to a constant value of about 0.4. In case of the UCNPs in H_2O , the $I_{green}/I_{red}(P)$ starts at a 5 times lower value, increases to a maximal value of about 0.7 and converges to the same $I_{green}/I_{red}(P)$ value of about 0.4 as observed for the UCPNs in D2O. This difference of relative spectral intensities of the UCL emission bands originate from the strong vibrational modes of H₂O located between 3300 cm⁻¹ to 3700 cm⁻¹ and their respective energy levels of Er3+ involved in the population of the emissive states. 159 In the case of D₂O, these vibrational modes are shifted to lower energies due to the higher mass of deuterium compared to the hydrogen atoms, and thus, are not resonant anymore with the energy gaps between the Er3+ energy levels, see Figure 6.3.

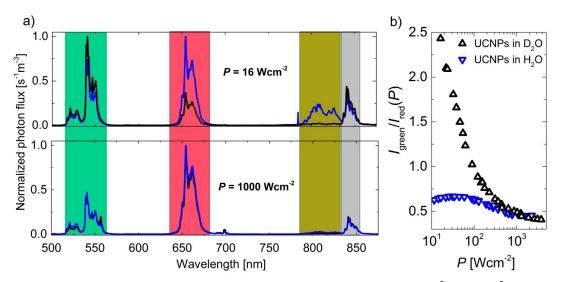


Figure 6.2 a) Normalized emission spectra of 23 nm-sized β-NaYF₄:Yb³⁺(19%),Er³⁺(2%) UCNPs dispersed in D₂O (black line) and H₂O (blue line) for low P value of 16 Wcm⁻² (top) and a high P of 1000 Wcm⁻² (bottom); b) P-dependent I_{green}/I_{red} for UCNPs in D₂O (black symbols) and H₂O (blue symbols). Reproduced from Würth, Kaiser et al. ⁵⁵ with permission from The Royal Society of Chemistry.

The energy scheme displayed in Figure 6.3 highlights the proposed dominant population pathways of the different UCL bands of the UCNPs in H2O. This model accounts for the efficient coupling of O-H vibrations to the Er³⁺ ⁴S_{3/2}/²H_{11/2} and ⁴I_{11/2} energy levels, enhancing nonradiative decay rates to the next lower energy level. At low P, the nonradiative decay of the Er³⁺ 980 nm ⁴l_{11/2} energy level results in a high population of the Er³⁺ 1520 nm ⁴l_{13/2} energy level for UCNPs dispersed in H₂O. This favors the population from the Er^{3+ 4}I_{13/2} energy level to the Er³⁺ red emissive ${}^4F_{9/2}$ energy level via Yb³⁺ to Er³⁺ ET (${}^4I_{13/2} \rightarrow {}^4F_{9/2}$) at low P, which agrees with the observed increase of $I_{green}/I_{red}(P)$ in the low P-region. Additionally, the high population of the Er^{3+ 4}I_{13/2} energy levels is also responsible for the high intensity of the 810 nm emission band induced by the Er³+-Er³+ energy transfer process 4 I_{13/2} + 4 I_{13/2} \rightarrow 4 I_{9/2}. 160 In contrast, for the UCNPs dispersed in D₂O and organic solvents, at low P, the Er³⁺ red emissive ⁴F_{9/2} level is mainly populated *via* direct nonradiative relaxation from the Er³⁺ green emissive ${}^{4}S_{3/2}/{}^{2}H_{11/2}$ level, see Figure 2.6. At high P, UCNPs showed solvent-independent population of the Er³⁺ red emissive ⁴F_{9/2} energy level, by a triphotonic process via the ⁴G_{11/2} energy level (see Figure 6.3, Right Panel), due to the compensation of solvent-induced quenching rates by the high P.

These conclusions of the (de-)population of UCNPs in H_2O and D_2O were underpinned with results from a rate equation analysis using the *P*-dependent emission spectra and luminescence lifetimes of Er^{3+} 520/540 nm and 655 nm, as well as the Yb³⁺ 980 nm emission bands, detailed in Publication II.

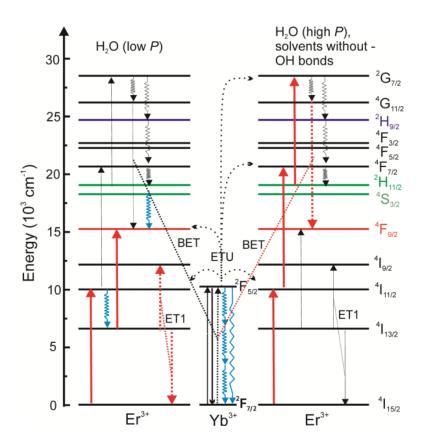


Figure 6.3 Energy level diagram for Yb³⁺-Er³⁺ interactions for UCNPs in H₂O at low *P* (left) and high *P* (right); red arrows show dominant population pathways; for high *P* the dominant population pathways of UCNPs in H₂O closely match with the photophysics of UCNPs in organic solvents and D₂O; Blue arrows: indicate nonradiative deactivation by O–H vibrations of H₂O. The arrow length represents the energy of the vibrational mode. ET: energy transfer, ETU: energy transfer upconversion, and BET: back energy transfer. Reprinted from Würth, Kaiser et al. ⁵⁵ with permission from The Royal Society of Chemistry.

6.3 Conclusions of chapter 6

In summary, the influence of vibrational modes of solvent molecules on the $\Phi_{UC}(P)$ and UCL was assessed for β -NaYF₄:Yb³⁺,Er³⁺ UCPNs with different surface modifications produced from the same UCNPs batch. The optical absorption of the solvent vibrational modes at 980 nm shows to be directly connected to nonradiative deactivation of near-surface excited Yb³⁺ ions. In this respect, H₂O with its highly absorptive vibrational modes, is acting as a strong luminescent quencher for the UCL, as also previously reported. The protection of the surface of DSPE-capped UCNPs from H₂O molecules is underlined by a reduction of UCL by only about 60% in H₂O compared to D₂O. This is a strong improvement compared to the UCL intensity loss of 99.9% of bare UCNPs dispersed in H₂O reported by Arppe et al. et al. 42, which have a about five times higher particle volume compared to the here investigated UCNPs.

With the aid of UCL emission spectra and $I_{green}/I_{red}(P)$ ratios, the (de-)population dynamics for the Er^{3+} energy levels of UCNPs in H_2O were identified. At low P, the fundamental vibrational modes of H_2O from 3300-3700 cm⁻¹ lead to high population of the $Er^{3+} I_{13/2}$ energy level, favoring the population pathway from the $I_{13/2}$ energy level to the red emissive $I_{13/2}$ energy level, favoring the population pathway from the $I_{13/2}$ energy level to the red emissive $I_{13/2}$ energy level to the red emissive are compensated resulting in identical (de-)population dynamics, and thus, identical emission color for UCNPs dispersed in I_{12} to I_{12} to I_{12} interpretations for the (de-)population dynamics in different solvents were supported by results of a rate equation analysis performed with a model from the Berry group I_{10} , which was detailed in Publication II. These methods for the refined understanding of the UC processes of UCNPs in I_{12} owere a valuable starting point for the investigations of the change of UC mechanisms by the variation of the dopant concentrations, which is discussed in the next chapter.

7 Dopant Concentration-Dependent Optical Properties of β-NaYF₄:Yb³⁺,Er³⁺ UCNPs

This chapter presents a study on the effect of varying the Yb $^{3+}$ and Er $^{3+}$ dopant concentrations on the emission colors of β -NaYF $_4$:Yb $^{3+}$,Er $^{3+}$ UCNPs with similar size. The P-dependent UCL data were measured with the ISS, and used as input data for a rate equation model containing all relevant Er $^{3+}$ levels to identify dopant concentration-dependent parameters. This combination of experimental data and theoretical analysis provided deep insights into the underlying UC mechanisms.

7.1 Dopant concentration-dependent upconversion luminescence (UCL)

In this section, the experimental results of the *P*-dependent Er^{3+} green and red emission intensity as a function of the Yb³⁺ and Er^{3+} dopant concentration of β -NaYF₄:Yb³⁺, Er^{3+} UCNPs are discussed. The UCL was recorded over a wide *P*-range (> two orders of magnitude) for two dopant concentration series, namely the Yb³⁺ and the Er^{3+} series. The **Yb³⁺ series** includes four samples with Yb³⁺ sensitizer varied concentration from 11% - 21% and a constant Er^{3+} concentration of 3%. The **Er³⁺ series** consisted of four samples, with a Er^{3+} activator concentration varied from 1% - 4% and a constant Yb³⁺ concentration of 14%. The sample containing 14% Yb³⁺ and 3% Er^{3+} is part of both series. All UCNPs samples were similarly sized to about 33 nm, and were capped with oleic-acid molecules to provide dispersibility in toluene. This ensures that the observed effects are solely caused by the variation of dopant concentrations. A list including both concentration series is provided in Table 4.1.

Two different P values were chosen as corner points for the comparison of the UCL of the Yb³⁺ and Er³⁺ series. The first P value of $P_{unsat} = 1.8 \text{ Wcm}^{-2}$ represents the unsaturated P-region for these UCNPs: The population of the Er³⁺ activator ion energy levels induced by the Yb³⁺ sensitizer ion via ET is low, and thus, the events of BET from Er³⁺ to Yb³⁺ is less pronounced. In this case, the biphotonic Er³⁺ green emission intensity increases with P^2 . For higher P, the increase of the Er³⁺ green emission intensity levels off due to saturation. The second value $P_{sat} = 380 \text{ Wcm}^{-2}$ marks the saturated P-region, where the intensity of the Er³⁺ green emission band increases linearly with P. In this case, the Er³⁺ energy levels are highly populated so that the contribution of BET from Er³⁺ to Yb³⁺ is relevant.

Figure 7.1 a) and c) presents the $I_{green}/I_{red}(P)$ for the Yb³⁺ and Er³⁺ series. The $I_{green}/I_{red}(P)$ converges to constant maximum and minimum values for decreasing and increasing P. Coincidentally, the $I_{green}/I_{red}(P)$ ratios reach minimum and maximum values at P_{sat} and P_{unsat} for the investigated UCNPs. Analyzing the dopant concentration-dependent trend of the $I_{green}/I_{red}(P)$ shows that with increasing Yb³⁺ concentration, the maximum value at low P is unaffected, while the minimum value at high P decreases with increasing Yb³⁺ concentration. Therefore the P-dependent color tuning range increases for increasing Yb³⁺, see Figure 7.1 a). In contrast, for increasing Er³⁺ concentration the maximum value at low P decreases, while the minimum value at high P only shows a slight increase. Subsequently, the P-dependent color tuning range can be increased with decreasing Er³⁺ concentration, see Figure 7.1 c).

Figure 7.1 b) and d) show the dopant concentration-dependent normalized UCL spectra at P_{unsat} and P_{sat} for the Yb³⁺ and Er³⁺ series. In order to highlight changes of the population of the respective energy levels, the intensity values of the UCL spectra were normalized to the overall UCL intensity $(I_{\lambda all}(P))$ (integration interval from 370 nm to 900 nm). Subsequently, the P-dependent intensities of the UCL spectra are relative contributions $I_{\rm rel}(\lambda,P)$. Further, the P-dependent relative intensities of the Er³⁺ green and red emission bands are referred to as $I_{\text{rel,green}}(P)$ and $I_{\text{rel,red}}(P)$, respectively. The UCL spectra of the Yb³⁺ series at P_{unsat} are identical, underlining that the population dynamics of the respective Er³⁺ energy levels are independent of Yb3+ concentration for this low P, see Figure 7.1 b). This reveals very good comparability of crystallinity (low defect density), crystal size, surface chemistry, and spatial Er^{3+} dopant ion distribution for this dopant concentration series, as $I_{rel,green}(P)$ and $I_{rel,red}(P)$ are known to be sensitive to these parameters by affecting the rate constants of the Er3+ ions. 40, 41, 55, 138, 161-163 Contrary, at P_{sat} , $I_{\text{rel,red}}(P)$ increases by a factor of 1.5 and $I_{\text{rel,green}}(P)$ drops by a factor of 2.3 for the Yb3+ concentration varied from 11% to 21%. An increase of I_{rel,red} for increasing Yb³⁺ concentration has been already reported in the literature, but the concrete mechanism was still under discussion.^{82, 104-106, 141-143}

The rate equation analysis, more detailed in the following section, showed that this enhancement originates from an enhancement of the triphotonic population pathway to the red emissive $Er^{3+} {}^4F_{9/2}$ energy level fed by the triphotonically activated $Er^{3+} {}^4G_{11/2}$ energy level *via* a BET to Yb³⁺, see Figure 2.6.

For the Er³⁺ series, at P_{unsat} , $I_{\text{rel,red}}(P)$ enhances and $I_{\text{rel,green}}(P)$ decreases by factors of 1.3 and 1.2 for Er³⁺ concentration varied from 1% to 4%, see Figure 7.2 d). This can be ascribed to a higher number of surface or near-surface Er³⁺ ions, which enhance the probability of a nonradiative relaxation from the Er³⁺ green emissive level ${}^2H_{11/2}$, ${}^4S_{3/2}$ to the Er³⁺ red emissive level ${}^4F_{9/2}$ (see Figure 2.6), which was found to be the main process for the activation of the Er³⁺ red emission band for these 33 nm-sized UCNPs dispersed in toluene at low P. Contrarily, at P_{sat} , $I_{\text{rel,red}}(P)$ decreases and $I_{\text{rel,green}}(P)$ increases each by a factor of only 1.1 for increasing Er³⁺ concentration. These opposite trends for the Er³⁺ series at P_{sat} and P_{unsat} explain the ambiguous spectral trends reported in the literature and underline the need of considering a broad $P_{\text{-range}}$.

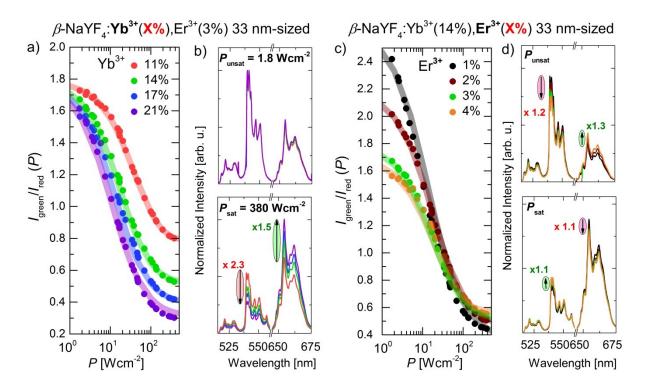


Figure 7.1 Dopant concentration and *P*-dependent spectral properties of 33 nm-sized oleate-capped β-NaYF₄:Yb³⁺,Er³⁺ UCNPs in toluene. a), b): Yb³⁺ series with a Yb³⁺ concentration varied between 11% to 21% at a constant Er³⁺ concentration of 3%; c), d): Er³⁺ series with a varied Er³⁺ concentration of 1% to 4% and a constant Yb³⁺ concentration of 14%; a), c) *P*-dependent measured (symbols) and simulated (lines) I_{green}/I_{red} values; b),d) UCL spectra from 515 nm to 675 nm for P_{unsat} of 1.8 Wcm⁻² and P_{sat} of 380 Wcm⁻²; all spectra were normalized to the total UCL intensity integrated from 370 nm to 900 nm. Reprinted from Kaiser et al.¹⁶⁴ with permission from *Tsinghua University Press*.

7.2 Influence of the dopant concentrations on rate equation constants

To support the interpretation of the dopant concentration effects, a comprehensive rate equation analysis of the UCNPs samples of the investigated Yb $^{3+}$ and Er $^{3+}$ series was performed. The experimentally determined P-dependent intensity values and slope factors (n(P)) of the Er $^{3+}$ red and green emission bands as well as the luminescence decay curves of the Er $^{3+}$ green, Er $^{3+}$ red, and Yb $^{3+}$ 980 nm emission bands were used as fitting parameters. The rate equation model, including all relevant Er $^{3+}$ energy levels and interactions, was adapted from the Berry group. $^{102, 123}$ Despite the success of this rate equation model from the Berry group in describing P-dependent UCL of UCNPs, 123 a couple of rate constants had to be changed drastically (by up to two orders of magnitude) for this work. To model the data of the Yb $^{3+}$ and Er $^{3+}$ series simplifications were made, e.g. the obtained rate constants were averaged over the complete particle volume. In contrast, Hossan et al. assumed an extra dark layer for such core-only particles without an inactive shell layer. 123 Comprehensive descriptions of the optimization of the rate constants, simplifications, and fitting procedures are provided in Publication III.

After optimizing the rate constants for the UCNPs sample containing 14% Yb³⁺ and 3% Er³⁺, only three parameters for the Yb³⁺ series and four parameters for the Er³⁺ series had to varied, including the respective density of dopant ions. The good match of the fitted data with the experimental data, including intensity and lifetime data of the Er³⁺ green and Er³⁺ red emission bands, see Figure 7.1 a) and c) and Publication III, supports the correct identification of the affected relevant parameters responsible for the dopant concentration-dependent color change of the UCNPs.

Figure 7.2 provides an overview of the rate constants affected by the variation of the Yb³+ and Er³+ concentration. For both series, an increase of the nonradiative Yb³+ 980 nm rate constant (k_{Yb_NR}) with increasing dopant concentration was observed. Thereby, k_{Yb_NR} was directly assessed from the measured decay behavior of the Yb³+ 980 nm luminescence. The increase of k_{Yb_NR} can be explained by a higher amount of near-surface Yb³+ and Er³+ ions that can be deactivated by oleic-acid ligand molecules (C-H vibrational modes between 2700 cm¹ and 2950 cm⁻¹)¹¹⁴ or by the surrounding toluene solvent molecules (vibrational modes at 8000 cm⁻¹)⁴0. k_{Yb_NR} was found to be more sensitive to changes of the Er³+ concentration, since the energy gap to the next lower lying level for the Er³+ 980 nm ⁴ $l_{11/2}$ is three times smaller compared to the Yb³+ 980 nm ⁴ $l_{5/2}$ energy level. In addition, energy migration to the particle surface is enhanced for higher dopant concentrations due to the shorter ion-ion distances. The rate equation analysis showed that the change of k_{Yb_NR} does only affect the

saturation behavior of the population of the Er³⁺ activator energy levels, but has no effect on the emission color of the UCNPs. Although the increase of k_{Yb_NR} also had a negative influence on the Φ_{UC} , it was shown that the increase of absorbing Yb³⁺ concentration overcompensates this effect with respect to UC particle brightness, which is defined as product of the Φ_{UC} , cross section [cm²] of an Yb³⁺ ion in a NaYF₄ matrix, and the number of absorbing Yb³⁺ ion inside the UCNP, see Publication III.

For the Yb³⁺ series, all ET and BET rates were found to be enhanced simultaneously by 220% for a Yb³⁺ concentration from 11% to 21%, equaling a decrease of the average Yb³⁺-Yb³⁺ and Yb³⁺-Er³⁺ distances by *ca.* 20% from 0.9 nm to 0.7 nm. The trend of the ion distance-dependent ET and BET rates shows rather an exponential behavior, which may indicate that the energy transfer between Yb³⁺ and Er³⁺ originates from electron hopping (Dexter ET), see 2.2 Energy transfer processes between Ln^{3+} . This underlines the possibility to investigate the physical nature of nonradiative ET for Yb³⁺,Er³⁺-based UC systems with a rate equation analysis. Moreover, the rate equation analysis revealed that for increasing Yb³⁺ concentration, a pronounced BET from triphotonically activated ${}^4G_{7/2}$ to the red emissive ${}^4F_{9/2}$ is responsible for the increase of the Er³⁺ red emission intensity at P_{sat} , see Figure 7.1 b). In the case of the Er³⁺ series, ET and BET are not affected. This is in accordance with a self-developed Monte Carlo simulation showing that for these relatively small Er³⁺ concentrations the average distance of Er³⁺ ions to the nearest Yb³⁺ ion changes only minimally (see Supporting Information of Publication III).

The observed color change for increasing Er^{3+} concentration results mainly from two processes. As discussed previously, at low P, the enhanced nonradiative rate from the ${}^2H_{11/2}$, ${}^4S_{3/2}$ to the red emissive ${}^4F_{9/2}$ energy level (k_{NR6}) by higher surface coupling results in an enhancement of the Er^{3+} red emission intensity, see middle Panel of Figure 7.3 b). At high P, the decrease of $I_{rel,red}(P)$ is assigned to a yet unknown Er^{3+} - Er^{3+} ET leading to a depopulation of the Er^{3+} red emissive ${}^4F_{9/2}$ energy level. This was simulated with a simplified approach by using an additional factor for the Yb^{3+} - Er^{3+} ET rate k_{ET5-8} , which also depopulates the Er^{3+} $^4F_{9/2}$ energy level. Therefore, the trend for the simulated change of Er^{3+} - Er^{3+} ET, see Figure 2.7 b) lower Panel, may not correspond to the real change of the Er^{3+} - Er^{3+} ET rate. In order to further optimize the rate equation system, four new possible candidates for the missing Er^{3+} - Er^{3+} ET rate constant have been suggested, which were detailed and discussed in Publication III.

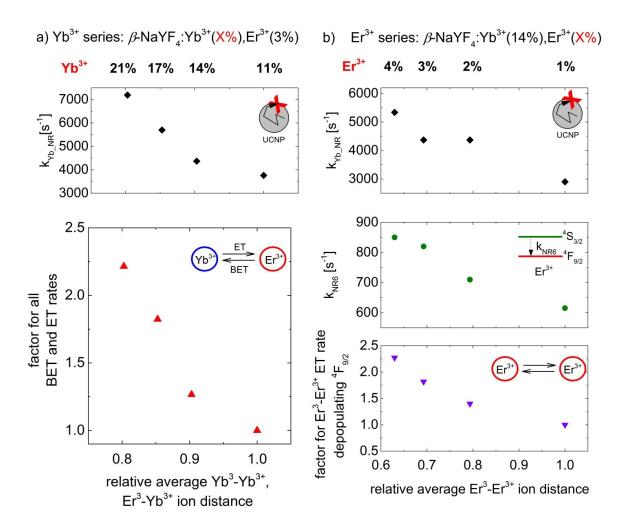


Figure 7.2 Dependence of the rate constants on the dopant concentration for varied parameters of the rate equation analysis: a) Yb³⁺ series and b) Er³⁺ series; In the case of the variation of the Yb³⁺ concentration, the nonradiative rate of the Yb³⁺ 980 nm ²F_{5/2} energy level and the ET and BET rates are influenced. For the Er³⁺ concentration, the nonradiative rate of the Yb³⁺ 980 nm ²F_{5/2} energy level, the nonradiative relaxation from the ²H_{11/2}, ⁴S_{3/2} to the red emissive level ⁴F_{9/2}, and a not yet identified Er³⁺-Er³⁺ ET rate is influenced. For the Er³⁺-Er³⁺ ET a simplified approach was used by varying an Yb³⁺-Er³⁺ ET depopulating the ⁴F_{9/2} energy level to simulate the enhancement of the red emission band. In Publication III, four possible candidates for the missing Er³⁺-Er³⁺ were discussed. Reproduced from Kaiser et al.¹⁶⁴ with permission from Tsinghua University Press.

7.3 Conclusions of chapter 7

In summary, the tuning of the UCL color of 33 nm-sized β -NaYF₄:Yb³⁺,Er³⁺ UCNPs by the variation of the Yb³⁺ and Er³⁺ dopant concentrations was studied for a P-range of over two orders of magnitude. Combining experimental results and theoretical investigations with a nine-level rate equation model from the Berry group, including the most significant energy levels of Er³⁺,^{102, 123} allowed to make reliable statements about the influence of the dopant concentrations on interactions between the ions and the particle surface.

The variation of the Yb³⁺ concentration from 11% to 21%, equaling a reduction in Er^{3+} -Yb³⁺ distance of 20%, resulted in an increase of 220 % for all Yb³⁺ to Er^{3+} ET rates and Er^{3+} to Yb³⁺ BET rates. This indirect measure of the change of ET processes opens the possibility to study the physical nature of nonradiative ET processes. The corresponding increase of the relative intensity of the Er^{3+} red emission band ($I_{rel,red}(P)$) at high P was attributed to an enhanced triphotonic activation of the red emissive Er^{3+} $^4F_{9/2}$ energy level the Er^{3+} $^4G_{11/2}$ to the red emissive $^4F_{9/2}$ energy level.

For the Er³⁺ series, different trends for $I_{\text{rel},\text{red}}(P)$ at low P and high P were determined. At low P, the increase of the number of Er³⁺ ions leads to an enhanced biphotonic activation of the Er³⁺ red emission band via direct relaxation from the green emissive ${}^2H_{11/2}$, ${}^4S_{3/2}$ to ${}^4F_{9/2}$ energy levels, induced by an increased surface coupling of the Er³⁺ ions. At high P, a not yet identified Er³⁺-Er³⁺ rate was found to be responsible for the decrease of $I_{\text{rel},\text{red}}(P)$. Moreover, these results reveal that the modulation range of the emission color by P can be extended by increasing the Yb³⁺ and/or decreasing the Er³⁺ concentration. Overall, this highlights the possibility of predicting the emission color for β -NaYF₄:Yb³⁺,Er³⁺ UCNPs by rate equation modelling.

8 General Conclusions for Publications

In summary, an ISS has been designed, validated and utilized for the accurate absolute determination of P-dependent Φ_{UC} and UCL spectra for different β -NaYF₄:Yb³⁺,Er³⁺ UC systems. The obtained $\Phi_{UC}(P)$ and UCL results for β -NaYF₄:Yb³⁺, Er³⁺ UCNPs in different solvents and with varied Yb³⁺ and Er³⁺ dopant concentrations, in combination with theoretical modelling, allowed substantial extensions of the understanding of surface quenching and color tuning.

Publication I summarized the challenges and requirements of the custom-built ISS for $\Phi_{\text{UC}}(P)$ measurements for 976 nm-excitable UC materials. This includes prerequisites on setup components and new calibration strategies. Moreover, measurement conditions for powders and dispersed UC samples were derived with respect to measurement geometry, experimental procedures, and optical properties of the UC material. This report can be used as a protocol to perform $\Phi_{\text{UC}}(P)$ measurements with minimum uncertainty in the order of about 10 %. The maximum Φ_{UC} of 10.5% at $P=30~\text{Wcm}^{-2}$ for a commercial β -NaYF₄:Yb³⁺(21%),Er³⁺(2%) 3 μ m-sized UC particles (UC μ P) is nowadays being frequently referenced as highest value for Yb³⁺,Er³⁺-based systems, underlining the overall accepted accuracy of this new ISS.

In Publication II, identical UCNPs dispersed in organic solvents, water (H_2O) and heavy water (D_2O) functionalized with different surface ligands were systematically studied. Thereby the Φ_{UC} was found to be mainly determined by the interaction of near-surface Yb^{3+} ions with the vibrational modes of the solvent and ligand molecules. In this respect, H_2O leads to the highest quenching rate from the UCNPs surface due to its high frequency O-H vibrational modes resonant to the Yb^{3+} 980 nm energy level. For organic solvents and D_2O , the probability of energy transfer is less likely, since their C-H and O-D vibrational modes are non-resonant to the Yb^{3+} 980 nm energy level as their frequencies are much lower. Instead, the lower-frequency vibrational modes of these solvents in combination with the low frequency vibrational modes of ligand molecules at ca. 3000 cm⁻¹ clearly affected the emission color of the UCNPs. Thereby, these vibrational modes can effectively bridge the energy gaps between the multitudes of Er^{3+} energy levels ranging from 1000 cm⁻¹ to 3500 cm⁻¹. Furthermore, the absolute UCL measurements, in combination with a rate equation analysis, enabled the development of a new (de-)population model explaining the change in emission color of UCNPs in H_2O .

8 General Conclusions for Publications

In Publication III, the influence of the dopant concentration on emission color and particle brightness for a set of 33 nm-sized oleate-capped UCNPs dispersed in toluene was assessed for Yb³⁺ and Er³⁺ dopant concentrations varied from 11 - 21% and 1 - 4%, respectively. The experimental results, including emission intensities and slope factors of the Er³⁺ green and red emission bands as well as luminescence lifetimes, were used for the optimization of the rate constants of a rate equation model considering all significant Er³⁺ energy levels. As a result an increase of the Yb³⁺ concentration was found to lead to enhanced triphotonic activation and an increase in Er³⁺ concentration to an enhanced biphotonic activation of the red emissive $^4F_{9/2}$ energy level. Moreover, the results indicate that the particle brightness of these UCNPs can be further improved with higher Yb³⁺ and lower Er³⁺ concentrations. This enabled an estimate of trends for the ultimate goal to find optimized dopant concentrations for maximal particle brightness in dependence of UCNPs size and particle architecture. Overall, a refined understanding of UC processes of β -NaYF₄:Yb³⁺,Er³⁺ UCNPs was gained by using quantitative measurements with the new custom-built ISS in combination with a comprehensive rate equation analysis.

List of Abbreviations

BET back energy transfer

BP beam profile

CCD charge-coupled device

CET cooperative energy transfer

CR cross-relaxation

DMF N,N-dimethylformamide, solvent for UCNPs

DS downshifted

DSPE 1,2-distearoyl-sn-glycero-3-phospho-ethanolamine-N-[methoxy-

(poly-ethylene glycol)-2000] (ammonium salt)

E_{em} energy of emitted photonsE_{abs} energy of absorbed photons

Er³⁺ trivalent erbium ions
ESA excited state absorption

ET energy transfer

ETU energy transfer upconversion full width at half maximum

Gauss_{exp} experimentally realized Gaussian beam profile

Ho³⁺ trivalent holmium ions

 $I_{\text{sample}}(\lambda, P)$ P-dependent spectrally-corrected sample spectrum $I_{\text{blank}}(\lambda, P)$ P-dependent spectrally-corrected blank spectrum

 $I(\lambda,P)$ P-dependent spectral photon flux

 $I(\lambda, P)$ for certain UC emission band or color

 $I_{\lambda all}(P)$ $I(\lambda, P)$ including all UC emission bands

 $I_{\text{rel}}(\lambda, P)$ $I(\lambda, P)$ relative to $I_{\lambda all}(P)$

 $I_{\text{rel},\Delta\lambda}(P)$ $I(\lambda,P)$ for certain UC emission band or color relative to $I_{\lambda all}(P)$

I_{green}/I_{red}(P) green-to-red intensity ratio
 ISS integrating sphere setup
 Ln³⁺ trivalent lanthanide ions
 Nd³⁺ trivalent neodymium ions
 NIR near-infrared spectral region

1

n(P) P-dependent slope factor
 N_{em} number of emitted photons
 N_{abs} number of absorbed photons

OA Oleic acid, ligand molecules for UCNPs

P excitation power density

P_{balance} balancing excitation power density

P_{BP} excitation power for certain beam profile

PL photoluminescence

Re³⁺ trivalent rare earth ions

Si-CCD silicon-based charged-coupled-device

SHG second harmonic generation

TEM transmission electron microscopy

TH_{exp} experimentally realized Top Hat beam profile

THideal ideal Top Hat beam profile

Tm³+ trivalent thulium ions

TPA two-photon absorption

TTA triplet-triplet annihilation

UC upconversion

UCL upconversion luminescence

UCNPs upconversion nm-sized particlesUCμP upconversion μm-sized particles

UV ultraviolet spectral region
 vis visible spectral region
 Yb³+ trivalent ytterbium ion

 α -NaYF₄ sodium yttrium tetraflouride in the cubic crystal phase

 β -NaYF₄ sodium yttrium tetraflouride in the hexagonal crystal phase

photoluminescence quantum yield

 $\Phi_{
m UC}$ upconversion photoluminescence quantum yield

 $\Phi_{UC}(P)$ P-dependent Φ_{UC} $\Phi_{UC,sat}$ saturated Φ_{UC} $\Phi_{UC,max}$ maximum Φ_{UC}

 $\Phi_{UC,green}(P)$ P-dependent Φ_{UC} of Er^{3+} green emission band $\Phi_{UC,red}(P)$ P-dependent Φ_{UC} of Er^{3+} red emission band $\Phi_{UC,purple}(P)$ P-dependent Φ_{UC} of Er^{3+} purple emission band $\Phi_{UC,vis}(P)$ P-dependent Φ_{UC} for visible spectral region

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List of Publications

First-author publication related to the upconversion research field referred in the text with capital roman numerals

P1. <u>M. Kaiser</u>, C. Würth, M. Kraft, I. Hyppänen, T. Soukka, U. Resch-Genger "Power-dependent upconversion quantum yield of NaYF₄: Yb³⁺, Er³⁺ nano-and micrometer-sized particles – measurements and simulations"

Nanoscale, 2017, 9, 10051-10058

DOI: 10.1039/C7NR02449E

P2. C. Würth*, <u>M. Kaiser*</u>, S. Wilhelm, B. Grauel, T. Hirsch, U. Resch-Genger "Excitation power dependent population pathways and absolute quantum yields of upconversion nanoparticles in different solvents"

Nanoscale, **2017**, 9 , 4283-4294

DOI: 10.1039/C7NR00092H

*equally contributed

P3. M. Kaiser, C. Würth, M. Kraft, T. Soukka, U. Resch-Genger

"Explaining the influence of dopant concentration and excitation power density on the luminescence and brightness of β -NaYF₄:Yb³⁺,Er³⁺ nanoparticles:

Measurements and simulations"

Nanoresearch, 2019, 12, 1871-1879

First-author publication unrelated to the upconversion research field

P4. M. Müller*, M. Kaiser*, G.M. Stachowski, U. Resch-Genger, N. Gaponik, A. Eychmüller

"Photoluminescence Quantum Yield and Matrix-Induced Luminescence Enhancement of Colloidal Quantum Dots Embedded in Ionic Crystals"

Chemistry of Materials, **2014**, 26 (10), 3231-3237

DOI: 10.1021/cm5009043 *equally contributed

Co-authored publications related to the upconversion research field

P5. S. Wilhelm, <u>M. Kaiser</u>, C. Würth, J. Heiland, C. Carrillo-Carrion, V. Muhr, O.S. Wolfbeis, W.J. Parak, U. Resch-Genger, T. Hirsch

"Water dispersible upconverting nanoparticles: effects of surface modification on their luminescence and colloidal stability"

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P6. R. Arppe, I. Hyppänen, N. Perälä, R.Peltomaa, <u>M. Kaiser</u>, C. Würth, S. Christ, U. Resch-Genger, M. Schäferling, T. Soukka

"Quenching of the upconversion luminescence of NaYF₄: Yb³⁺, Er³⁺ and NaYF₄: Yb³⁺, Tm³⁺ nanophosphors by water: the role of the sensitizer Yb³⁺ in non-radiative relaxation"

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P7. A. Pilch, C. Würth, M. Kaiser, D. Wawrzyńczyk, M. Kurnatowska, S. Arabasz, K. Prorok, M. Samoć, W. Strek, U. Resch-Genger, A. Bednarkiewicz "Shaping Luminescent Properties of Yb³⁺ and Ho³⁺ Co-Doped Upconverting

Core—Shell β-NaYF₄ Nanoparticles by Dopant Distribution and Spacing"

Small, **2017**, 13, 1701635

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Co-authored publications unrelated to upconversion research field

P8. C. Würth, D. Geißler, T. Behnke, <u>M. Kaiser</u>, U. Resch-Genger

"Critical review of the determination of photoluminescence quantum yields of luminescent reporters"

Analytical and bioanalytical chemistry, **2015**, 407, 59-78 DOI: 10.1007/s00216-014-8130-z

P9. S. Hatami, C. Würth, <u>M. Kaiser</u>, S. Leubner, S. Gabriel, L. Bahrig, V. Lesnyak, J. Pauli, N. Gaponik, A. Eychmüller, U. Resch-Genger

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P11. M.R. Wagner, G. Callsen, J.S. Reparaz, J.-H. Schulze, R. Kirste, M. Cobet, I.A. Ostapenko, S. Rodt, C. Nenstiel, <u>M. Kaiser</u>, A. Hoffmann, A.V. Rodina, M.R.Phillips, S.Lautenschläger, S. Eisermann, B.K. Meyer

"Bound excitons in ZnO: Structural defect complexes versus shallow impurity centers"

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List of publications

P12. B.K. Meyer, J. Sann, S. Eisermann, S. Lautenschlaeger, M.R. Wagner, <u>M Kaiser</u>, G. Callsen, J.S. Reparaz, A. Hoffmann

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MRS Proceedings, **2009**, 1202, 1202-105-21

DOI: 10.1557/PROC-1202-I05-21

Conference Talks and Posters

This list concludes all first-author conference talks and poster presentations.

- C1. <u>M. Kaiser</u>, C. Würth, I. Hyppänen, E. Palo, T. Soukka, U. Resch-Genger "Optical conversion efficiency of up-conversion nanoparticles as new class of luminescent reporters" *ICL 2014*, Wroclaw, Poland, **2014**
- C2. <u>M. Kaiser</u>, C. Würth, I. Hypännen, E. Palo, T. Soukka, U. Resch-Genger "Upconversion quantum yields of rare earth doped nanoparticles dependent on dopant concentration" 78th DPG Spring Meeting 2014, Dresden, Germany, 2014
- C3. <u>M Kaiser</u>, C. Würth, U. Resch-Genger, I. Hyppänen, T. Soukka "Integration Sphere Setup for the Absolute Determination of Upconversion Quantum Yields of Lanthanide Doped Nanoparticles" iNOW2013 International Nano Optoelectronics Workshop, Cargese, France, 2013
- C4. <u>M. Kaiser</u>, S. Wilhelm, C. Würth, J. Heiland, O.S. Wolfbeis, T.Hirsch, U.Resch-Genger "Luminescence Properties of Upconverting NaYF₄(Yb,Er) Nanoparticles in water and heavy water" 13th Conference on Methods and Applications of Fluorescence (MAF 13), Genoa, Italy, 2013
- C5. <u>M. Kaiser</u>, C.Würth, U.Resch-Genger, I.Hyppänen, T.Soukka "Upconversion nanoparticles as new class of fluorescent reporters Tools to characterize their signal-relevant optical properties" *Biosensor Symposium*, Berlin, Germany, 2013
- C6. <u>M. Kaiser</u>, C. Würth, U. Resch-Genger, I. Hyppänen, T. Soukka **"Tools for the characterization of the signal-relevant properties of upconversion nanoparticles as new class of fluorescent reporters"** *ANAKON***, Essen, Germany, 2013**
- C7. <u>M. Kaiser</u>, J. Heiland, M. Kraft, C. Würth, S. Wilhelm, V. Muhr, N. Leibl, O. S. Wolfbeis, T. Hirsch, U. Resch-Genger "Optical characterization of lanthanide doped up-converting nanoparticles" *BAM Wissensbörse*, Berlin, Germany, 2013
- C8. <u>M. Kaiser</u>, C. Würth, U. Resch-Genger, I. Hyppänen, T. Soukka "Absolute Photoluminescence Quantum Yield of Hexagonal NaYF₄:Er³⁺,Yb³⁺ Upconversion Nanoparticles" 77th DPG Spring Meeting 2013, Regensburg, Germany, 2013
- C9. <u>M. Kaiser</u>, C. Würth, M. Vorsthove, T. Felbeck, U. Kynast, U. Resch-Genger "Luminescence Properties of Cer-dotierten Yttrium-Aluminium-Garnet (YAG:Ce) Nanoparticles Absolute Quantum Yields and Influence of Particle Size" Nano-Additive, Berlin, Germany, 2012

- C10. M. Kaiser, C. Würth, T. Felbeck, M. Vorsthove, U. Kynast, U. Resch-Genger "Influence of the Particle Size on the Optical Properties of YAG:Ce"

 Photochemie Tagung, Postdam, Germany, 2012
- C11. M. Kaiser, U. Resch-Genger, C. Würth, U. Kynast, T. Felbeck, M. Vorsthove "Luminescence Properties of Cer-dotierten Yttrium-Aluminium-Garnet (YAG:Ce) Nanoparticles" Upcore Joint Seminar 2012, Regensburg, Germany, 2012
- C12. M. Kaiser, U. Resch-Genger, C. Würth, U. Kynast, T. Felbeck, M. Vorsthove "Luminescence Properties of Cer-dotierten Yttrium-Aluminium-Garnet (YAG:Ce) Nanoparticles Absolute Quantum Yields and Influence of Particle Size" 76th DPG Spring Meeting 2012, Berlin, Germany, 2012
- C13. <u>M. Kaiser</u>, M. R. Wagner, G. Callsen, A. Hoffmann, S. Lautenschläger, S. Eisermann, B. K. Meyer "Excitons and their excitation channels in a-plane and c-plane ZnO" 74th DPG Spring Meeting 2010, Regensburg, Germany, 2010

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Contributions to the Manuscripts

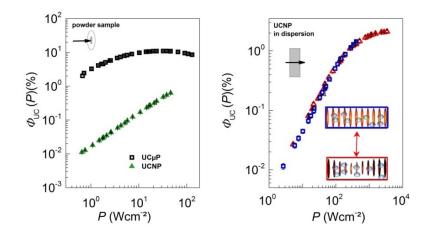
Publication I

Manuscript Title:

Power-dependent upconversion quantum yield of NaYF₄:Yb³⁺,Er³⁺ nano- and micrometersized particles – Measurements and simulations

List of Authors:

Martin Kaiser, Christian Würth, Marco Kraft, Iko Hyppänen, Tero Soukka, Ute Resch-Genger



Substantial contributions:

A newly custom-built integrating sphere setup was designed, built up and calibrated. I performed all measurements including excitation power density-dependent absolute upconversion luminescence (UCL), upconversion quantum yield (Φ_{UC}), and lifetime measurements, did the analysis of the data and created all graphics. New approaches for the analysis of the data as P-dependent slope factor and relative contribution were developed. I experimentally realized and characterized the Top Hat and Gaussian beam profiles. I validated the experimental results with respect to the influence of the beam profile on Φ_{UC} with theoretical simulations by using the formula of the Andersson-Engels group for ideal biphotonic converters. I wrote the manuscript in close corporation with Dr. Ute Resch-Genger and Dr. Christian Würth.

Publication II

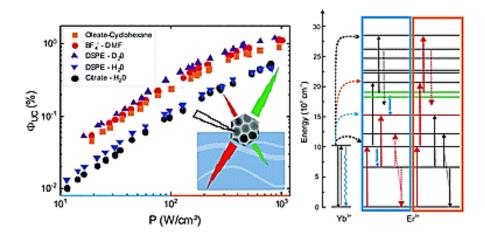
Manuscript Title:

Excitation power dependent population pathways and absolute quantum yields of upconversion nanoparticles in different solvents

List of Authors:

Christian Würth*, <u>Martin Kaiser</u>*, Stefan Wilhelm, Bettina Grauel, Thomas Hirsch, Ute Resch-Genger

*Equally contributed



Substantial contributions:

I performed all measurements including excitation power density-dependent absolute upconversion luminescence (UCL), upconversion quantum yield (Φ_{UC}), and lifetime measurements. I developed a Φ_{UC} measurement strategy for upconversion nanoparticles (UCNPs) dispersed in water and did the analysis of all Φ_{UC} and lifetime data and created many of the graphics of the manuscript. Interpretations of the luminescence quenching and change in the luminescence emission band ratios were concluded with Dr. Christian Würth. I proposed that the increasing green-to-red intensity ratio at low P is an argument for the population of the red emissive ${}^4F_{9/2}$ energy level via ${}^4I_{13/2}$ level. I drafted the paper in close corporation with Dr. Christian Würth and Dr. Ute Resch-Genger.

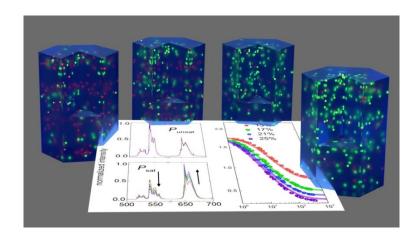
Publication III

Manuscript Title:

Explaining the influence of dopant concentration and excitation power density on the luminescence and brightness of β -NaYF₄:Yb³⁺,Er³⁺ nanoparticles: Measurements and simulations

List of Authors:

Martin Kaiser, Christian Würth, Marco Kraft, Tero Soukka, Ute Resch-Genger



Substantial contributions:

I guided Marco Kraft at measuring the excitation power density-dependent absolute upconversion luminescence (UCL) and upconversion quantum yield (Φ_{UC}) for the differently doped β -NaYF₄:Yb³⁺,Er³⁺ nanoparticles. I performed all luminescence lifetime measurements, did the analysis of all data, and performed a rate equation analysis with a 9-level Er³⁺ rate equation model. I did the photophysical interpretation, which was discussed with Dr. Christian Würth. I wrote the manuscript in close corporation with Dr. Ute Resch-Genger.